Preparation and physicochemical properties of zeolitic imidazolate framework-8 (ZIF-8)/rice husk derived graphene (GRHA) nanohybrid composites

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Abstract. This paper proposes an improve nanohybrid composites of Zeolitic Imidazolate Framework-8 (ZIF-8)/Rice husk Derived Graphene (GRHA). The main goal of this work is to prepare the nanohybrid composites with high surface area and enhanced porosity. The composite is prepared via aqueous room temperature method which is simple and fast. Based on Fourier transform Infrared (FTIR) and X-ray Diffraction (XRD) analysis, it shows that the produced ZIF-8 is in sodalite (SOD) structure while GRHA is in amorphous due to the presence of multilayer graphene. Raman analysis shows that the prepared GRHA has a high degree of graphitization. The BET specific surface area (BET_{SSA}) is found to increase up to three times higher (1632.10 m²/g) as compared to pristine GRHA (518.11 m²/g) and ZIF-8 (687.32 m²/g) respectively. Therefore, it is envisaged that this composite can be very useful for hydrogen storage.

1. Introduction

The demand of fossil fuel will keep increasing which can cause severe problem namely global warming. Hence, it is vital to overcome this issue by developing an alternative energies to the fossil fuels [1]. Hydrogen has been one of the best future energy as it can be used in electricity generation and alternative fuel for vehicles. Additionally, hydrogen is a clean energy as it does not emit CO₂ which reduces the environmental pollution [2]. Nevertheless, hydrogen storage has become the major issue that need to be resolved [3]. Currently, there are several ways that can be used to store hydrogen such as liquefaction [4], compression [5] and metal-hydride systems [6]. However, of all the aforementioned methods, adsorption via porous materials such as activated carbon nanofibers [7], porous carbons, zeolites and metal organic frameworks (MOFs) is favorable because of it fast kinetics, good cyclability and excellent adsorption capacity [1]. Currently, adsorption via metal organic frameworks (MOFs) shows a great promise for hydrogen storage [8]. In general, MOFs is composed of inorganic metal connectors and

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organic linkers [9] which have high surface area and porosity [10], ordered crystalline structures as well as enhanced mechanical stability [9]. Zeolitic imidazolate frameworks-8 (ZIF-8) which is made up of zinc (Zn) metal and 2-methylimidazole (MIM) [11] has been studied extensively. Interestingly, ZIF-8 can be produced in deionized water at room temperature [12] which is fast and convenient [13].

It was said that, the properties of ZIF-8 can be further enhanced with the addition of graphene as the composites will have a better performance (hydrogen storage) [14] and higher surface area [15]. Unfortunately, the discovery of methods of producing was still under rapid progress. Even though chemical vapour deposition (CVD) has been an established method to produce graphene [16], but this method was complicated [17]. Currently, conversion of biomass material into graphene has received tremendous attention [18]. Rice husk, waste corn shell, egg shell and gelatin were some of the biomass precursors that can be used as a starting material to produce graphene [18,19]. Typically, potassium hydroxide (KOH) was used to activate biomass precursors as it will helps in the formation of pores [20] and producing a high purity graphene [21]. In general, rice husk contains 80 % of organic constituents such as lignin, cellulose and hemicellulose while the other 20 % is the inorganic silicon dioxide (SiO₂). Therefore, rice husk is one of the best precursors that can be used to synthesis rice husk derived graphene (GRHA) [22]. In this study, rice husk was activated with KOH to produce GRHA. Later, GRHA/ZIF-8 hybrid nanocomposites were synthesized through simple aqueous room temperature method to improve the physicochemical properties and the adsorption capacity of the material to store hydrogen gas.

2. Materials and Methodology

2.1. Materials

Rice husks were collected from Johor (Kilang Beras Jelapang Selatan Sdn. Bhd.). Zinc nitrate hexahydrate ($Zn(NO_3).6H_2O$, 99%), triethylamine (TEA) and 2-methylimidazole (2-MIM, 99%) were purchased from Acros Organics while potassium hydroxide (KOH) was purchased from Quality Reagent Chemical (QReC).

2.2. Synthesis of GRHA

Rice husk was carbonized at 350 °C for 2 h to form rice husk ash (RHA). RHA (3 g) was mixed with KOH powder (15 g) and the mixture was annealed in a muffle furnace at 900 °C for 2 h. The GRHA was sonicated for 30 mins and washed with distilled water. Later, it was centrifuged at 3200 rpm for 10 mins for several times until neutral pH was obtained. Lastly, GRHA was dried in an oven at 80 °C overnight [23].

2.3. Preparation of ZIF-8 and GRHA/ZIF-8

ZIF-8 was prepared using the procedure as discussed by [12]. For the composites, metal solution was prepared by adding 2.95 g of $Zn(NO_3)_2.6H_2O$ in 74 mL of deionized water and in a separate beaker, 0.2 g of GRHA was dispersed in 26 mL of deionized water. Both of the solution was mixed and sonicated for 30 min. Meanwhile, the ligand solution was prepared by adding, 6.5 g of 2-MIM and 20 mL of TEA in 100 mL of deionized water. The ligand solution was then added into the sonicated solution for 1 h under vigorous stirring. The solution mixture was allowed to be centrifuged and washed with deionized water. The product was then dried in an oven at 60 °C for 24 h and grounded into fine particles [24].

2.4. Characterization

In order to determine the functional groups attached on the samples, Fourier Transform Infrared (FTIR) was used. The IR spectra was collected after 32 scans in the $4000 - 500 \text{ cm}^{-1}$ regions. to determine the crystallinity of the samples, X-Ray Diffraction (XRD, Smart Lab, Rigaku) was used. Data was collected using Cu-K α radiation (1.54 nm) in the range of 20 from 5° to 50°. Brunauer-Emmet-Teller (BET) was used to analyze the surface area of GRHA, ZIF-8 and GRHA/ZIF-8. To analyze the degree of graphitization of GRHA, Raman spectroscopy (Horiba, Raman Xplora Plus) was used.

3. Results and discussions

3.1. FTIR analysis

GRHA shows a weak IR spectrum as it is lack of functional groups (Refer figure 1 (a)). Therefore, GRHA only exhibits an obvious C-O vibration around 1080 cm⁻¹ [25]. In contrast, ZIF-8 (Refer figure 1 (b)) confirms the presence of Zn-N and Zn-O bonds as it shows a vibrational peak at 657 cm⁻¹ and 758 cm⁻¹ respectively [26]. On the other hands, peak at 1175 cm⁻¹ and 1580 cm⁻¹ indicates the stretching vibrations of C-N and C=N [27]. Interestingly, when GRHA is introduced into the nanocomposites, new stretching vibrations of C-H bond is formed (2930 cm⁻¹) (Refer figure 1 (c)) [28].



Figure 1. FTIR spectrum of (a) GRHA/ZIF-8 (b) ZIF-8 (c) GRHA.

Figure 2. Raman shift of (a) GRHA/ZIF-8 (b) ZIF-8 (c) GRHA.

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3.2. Raman analysis

 I_D/I_G ratio of GRHA is calculated using Raman spectroscopy. This calculated value will indicate the degree of graphitization of GRHA [29]. In this study, it is found that the I_D/I_G ratio of GRHA (Refer figure 2 (c)) is 0.45 which suggest that the degree of graphitization of the sample is relatively high [30,31]. However, the intensity of D (1343 cm⁻¹), G (1577 cm⁻¹), and 2D (2675 cm⁻¹) peaks are quite low due to the multilayer structural defects [32]. For ZIF-8 (Refer figure 2 (b)), peaks at 390 cm⁻¹ shows the Zn–O–Zn vibrational bands and this peak can still be observed even after the addition of GRHA [33] Therefore, it can be said that the addition of GRHA did not alter the Raman peak of GRHA/ZIF-8 (Refer figure 2 (a)).

3.3. Crystallinity study

Figure 3 (a) depicts that GRHA corresponds to few layers of graphene as it shows broad peak at $2\theta = 26.62^{\circ}$ (002) [34]. In addition, the broad peaks indicates that the GRHA is in amorphous state because of the presence of multilayer GRHA [27]. However, this result is similar as reported by [18]. On the other hand, XRD diffactogram of ZIF-8 (Refer figure 2 (c)) confirms the sodalite (SOD) structure of the samples because all peaks at (011), (002), (012), (022), (013), and (222) are observed [35]. Hence, a pure ZIF-8 has been successfully synthesized via aqueous room temperature method [36]. It can be seen from figure 3 (b) that the GRHA/ZIF-8 XRD diffactogram shows a higher peak intensity as compared to pristine ZIF-8 because the introduction of GRHA able to remove guest molecules in the composites [12]. Nevertheless, the XRD diffactogram of both ZIF-8 and GRHA/ZIF-8 are almost similar. Therefore, it can be said that the addition of GRHA into the composites did not change the formation of ZIF-8 crystal structure [37].



Figure 3. XRD spectrum of (a) GRHA (b) GRHA/ZIF-8 (c) ZIF-8.

3.4. Surface area analysis

In this study, the GRHA shows a BET specific surface area (BET_{SSA}) of 518.11 m²/g while ZIF-8 is 687.32 m²/g (Refer table 1). Interestingly, the nanocompoites of GRHA/ZIF-8 shows a significant change in the surface area where it shows BET_{SSA} up to 1632.10 m²/g. The improvement of BET_{SSA} of GRHA/ZIF-8 is due to the combination of microporosity of ZIF-8 with mesoporosity of GRHA [24]. Moreover, the synergistic effect between GRHA and ZIF-8 also contributed to the increment of BET_{SSA} of the nanocomposites [38]. Besides that, the removal of guest molecules and formation of new pores after the addition GRHA also contributed to this phenomenon [39]. Previously, the highest BET_{SSA} obtained from GO/ZIF-8 is only 202 m²/g [24] and 917 m²/g [38]. This study produced GRHA/ZIF-8 nancomposites with BET_{SSA} of 1632.1 m²/g which proves that our nanocomposites have an enhanced surface area.

Sample	$BET_{SSA} (m^2/g)$	Total pore volume	Micropore
		(cm^3/g)	volume (cm ³ /g)
GRHA/ZIF-8	1632.10	1.1694	0.8318
ZIF-8	687.32	0.3795	0.3036
GRHA	518.11	0.3346	0.2927

Table 1. Specific surface area and porosity data.

4. Conclusions

GRHA was synthesized through chemical activation using KOH whereas ZIF-8 and GRHA/ZIF-8 were formed through aqueous room temperature method. It is found that new C-H bond is formed which confirms the interaction between GRHA and ZIF-8. The prepared ZIF-8 shows a good crystallinity with sodalite (SOD) while the addition of GRHA into ZIF-8 did not alter the crystallinity of ZIF-8 which has been confirmed using XRD. GRHA exhibits amorphous structure because of the presence of multilayer graphene. However, Raman shift shows that GRHA has a relatively high degree of graphitization (I_D/I_G = 0.45). The introduction of GRHA into the composite greatly enhanced the BET_{SSA} which is 1632.10 m²/g. Therefore, these findings suggest that GRHA/ZIF-8 can be a potential material for hydrogen storage since it has an enhanced physicochemical properties.

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References

- [1] Xia Y, Yang Z and Zhu Y 2013 J. Mater. Chem. A 1 9365
- [2] Liew P Y, Kamaruddin M J, Ho W S Asli U A, Bong C P C, Hassim M H,
- Chemmangattuvalappil N G and Mah A X Y 2019 Int. J. Hydrogen Energy 44 5661
- [3] Collins D J and Zhou H C 2007 J. Mater. Chem. 17 3154
- [4] Cardella U, Decker L, Sundberg J and Klein, H 2017 Int. J. Hydrogen Energy 42, 12339
- [5] Mazloomi K, and Gomes C 2012 Renew. Sustain. Energy Rev. 16 3024.
- [6] Bellosta von C J, Ares J R, Barale J, and Baricco M 2019 Int. J. Hydrogen Energy 44 7780
- [7] Che O F E, Yusof N, Hasbullah H and Jaafar J 2017 J. Ind. Eng. Chem. 51 281
- [8] Yan Y, Da Silva I, Blake A J and Dailly A 2018 Inorg. Chem. 57 12050
- [9] Zheng Y Y, Li C X, Ding X T and Yang Q 2017 Chinese Chem. Lett. 28, 1473
- [10] Szczęśniak B, Choma J and Jaroniec M 2018 Appl. Surf. Sci. 459 760
- [11] Lee Y R, Jang M S, Cho H Y and Kwon H J 2015 Chem. Eng. J. 271 276
- [12] Nordin N A H M, Ismail A F, Mustafa A and Goh P S 2014 RSC Adv. 4 33292
- [13] Khan I U, Othman M H D, Jilani A and Ismail A F 2018 Arab. J. Chem. 80
- [14] Pokhrel J, Bhoria N, Anastasiou S and Tsoufis T 2018 Microporous Mesoporous Mater. 267 53
- [15] Szczęśniak B, Choma J and Jaroniec M 2018 J. Colloid Interface Sci. 514 801
- [16] Lee H C, Liu W W, Chai S P and Mohamed A R 2016 Procedia Chem. 19 916
- [17] Seah C M, Chai S P and Mohamed A R 2014 *Carbon N. Y.* 2014 **70** 1
- [18] Purkait T, Singh G, Singh M, Kumar D and Dey R S, Sci. Rep. 7 1
- [19] Chen F, Yang J, Bai T, Long B and Zhou X 2016 J. Electroanal. Chem. 768 18
- [20] Ojha K, Kumar B and Ganguli A K 2017 J. Chem. Sci. 129 397
- [21] Priyanka M and Saravanakumar M P 2017 IOP Conf. Ser. Mater. Sci. Eng. 263 246
- [22] Sankar S, Lee H, Jung H and Kim A 2017 New J. Chem. 41 13792
- [23] Muramatsu H, Kim, Y A, Yang K S and Cruz-Silva, R 2014 Small 10 2766
- [24] Kim D, Kim D W, Hong W G and Coskun A 2016 J. Mater. Chem. A 4 7710
- [25] Çiplak Z, Yildiz N and Çalimli, A 2015 Fullerenes, Nanotub. Carbon Nanostructures 23 361
- [26] Wang J, Li Y, Lv Z and Xie Y 2019 J. Colloid Interface Sci. 542 410
- [27] Jamil N, Othman N H, Alias N H and Shahruddin M Z 2019 J. Solid State Chem. 270 419
- [28] Chen J, Liu K, Jiang M and Han J 2019 Colloids Surfaces A Physicochem. Eng. Asp. 568 461
- [29] Zhipeng W, Hironori O, Shingo M and Josue O M 2015 Carbon N. Y. 94 479
- [30] Cao Y, Wang K, Wang X and Gu Z 2016 *Electrochim. Acta* **212** 839
- [31] Jung S H, Myung Y, Kim B N and Kim I G 2018 Sci. Rep. 8 1
- [32] Shams S S, Zhang L S, Hu R, Zhang R and Zhu J 2015 Mater. Lett. 161 476
- [33] Kumar R, Jayaramulu K, Maji T K and Rao C N R 2013 Chem. Commun. 49 4947
- [34] Gupta B, Kumar N, Panda K and Kanan V 2017 Sci. Rep. 7 45030
- [35] Pokhrel J, Bhoria N, Anastasiou S and Tsoufis T 2018 Microporous Mesoporous Mater. 267 53
- [36] Chen B, Wan C, Kang X and Chen M 2019 Sep. Purif. Technol. 223 113
- [37] Tsoufis T, Tampaxis C, Spanopoulos I and Steriotis, T 2018 Microporous Mesoporous Mater. 262 68
- [38] Thomas M, Illathvalappil R, Kurungot S and Nair, B.N 2016 ACS Appl. Mater. Interfaces 8 29373
- [39] Md Nordin N A H, Racha S M, Matsuura T and Misdan N 2015 RSC Adv. 5 43110