# STRUCTURAL STUDY OF CO-EXTRUDED DUAL-LAYER HOLLOW FIBRE FOR MICRO-TUBULAR SOLID OXIDE FUEL CELL

SITI HALIMAH BINTI AHMAD

A thesis submitted in fulfilment of the requirements for the award of the degree of Master of Engineering (Gas)

Faculty of Chemical and Energy Engineering Universiti Teknologi Malaysia

FEBRUARY 2016

To my beloved family,

Mr. Azam Omar, Mr. Ahmad Salleh, Madam Aminah Rahmat, Mr. Md. Shamsuddin Salleh, Madam Rohaizat Baharuddin, Muhamad Muzakkir, Muhamad Muzamil, Siti Nur Ain

With love and gratitude.

#### ACKNOWLEDGEMENTS

In the Name of Allah, the Most Gracious, the Most Merciful. All praise and deepest gratitude to Allah SWT, for His Mercy has given me patience, health and strength to accomplish this research study and dissertation. Besides, my deepest appreciation is dedicated to my father, mother and beloved family, which has been encouraging me to do my best and success in life. Thank you again for their treasured support and love pouring into my life.

Special thanks go to my supervisor, Dr. Mohd Hafiz Dzarfan bin Othman and Dr Mukhlis A. Rahman as my co-supervisor for their brilliant ideas, valuable time, vast knowledge, financial support, technical support and for tolerating with all my mischievous behaviour. Their kindness made this thesis a remarkable success.

In addition, I wish to express my sincere appreciation to my husband, Nur Azam Bin Omar, thank you very much for your continuous support, advices and motivation. To my best friends, Zaleha, Maryam, Syuada, Syarad, Naza, Hamizah, and especially, Syafikah Huda (my classmates from the secondary school until now), thank you for the value of friendship that make them always been at my side through thick and thin.

Exceptional dedication to my best teammates and friends, Shuhaida, Azureen, Fazliana, Fatin, Khalisah, Munira, Ermala, Adha, Hilmi, Muhazri, Taufik and Azuwa for their cooperation, knowledge, assistance and friendship during my study. Do not forget, millions thanks go to all lecturers and colleagues in Advanced Membrane Technology Research Centre (AMTEC). Last but not least, I am grateful to Universiti Teknologi Malaysia (UTM), Ministry of Higher Education Malaysia (MOHE) and Ministry of Science, Technology and Innovation Malaysia (MOSTI) for financial support.

### ABSTRACT

This study provides information on the structure of solid oxide fuel cell (SOFC). Porous anode provides active site reaction while dense electrolyte layer prevents direct flow of gases through one electrode to another. Therefore, anode and electrolyte structural modifications were thoroughly investigated using different types of pore formers and various ceramic particle sizes, respectively. In the preliminary study, 0 to 10 wt.% corn starch and polyetherethylketone (PEEK) functioned as pore former were added into nickel oxide-yttria-stabilized zirconia (NiO-YSZ) anode suspensions. The results showed that high loading of pore former increased the porosity in anode but reduced the mechanical strength. As compared to using corn starch, addition of 2 wt.% PEEK produced anode with better porous structure by generating more connected open pores and the hollow fibre (HF) was 67 % stronger. The electrolyte of dual-layer HF was subsequently modified by varying the loading of YSZ particle sizes (i.e. micron, submicron and nano-sized) during suspension preparation. The most promising electrolyte layer with thin, dense, gastight and defect-free was comprised of 70 % submicron-YSZ and 30 % nano-YSZ. The HF yielded the highest mechanical strength of 85 MPa, good gas-tightness behaviour of 3.16x10<sup>-6</sup> molm<sup>-2</sup>s<sup>-1</sup>Pa<sup>-1</sup> and successfully reduced the co-sintering temperature from 1450 to 1400 °C. Lastly, the anode suspension consist of 2 wt.% PEEK and electrolyte suspension composed of a mixture of particle sizes of micron, submicron and nano-sized YSZ in a ratio of 3:4:3 were co-extruded and co-sintered to produce the structural modified electrolyte/anode. Results revealed that the connected open pores at the entrance of anode inner surface resulted from the addition of pore former was significantly contributed to gas permeability of anode. However, the electrolyte was not fully densified due to less efficient electrolyte packing particles which resulted to the reduction in mechanical strength and integrity of HF.

### ABSTRAK

Kajian ini menyediakan maklumat mengenai struktur sel bahan api pepejal teroksida (SOFC). Anod berliang menyediakan ruang tindak balas aktif manakala lapisan elektrolit padat menghalang aliran langsung gas dari satu elektrod kepada yang lain. Oleh itu, ubahsuai struktur anod dan elektrolit masing-masing telah disiasat menggunakan pelbagai jenis pembentuk liang dan saiz zarah seramik. Dalam kajian awal, 0 hingga 10 % jisim kanji jagung dan polieteretilketon (PEEK) yang berfungsi sebagai pembentuk liang telah ditambah ke dalam ampaian anod nikel oksida-ytria-zirkonia-terstabil (NiO-YSZ). Hasil kajian menunjukkan muatan pembentuk liang yang lebih tinggi meningkatkan lagi keliangan dalam anod tetapi mengurangkan kekuatan mekanikal. Berbanding dengan menggunakan kanji jagung, penambahan 2 % jisim PEEK menghasilkan anod dengan struktur berliang yang lebih baik dengan menjana lebih banyak liang terbuka yang bersambung dan gentian geronggangnya (HF) adalah 67 % lebih kuat. Elektrolit dwi-lapisan HF kemudiannya diubahsuai dengan mengubah muatan saiz zarah YSZ (iaitu mikron, submikron dan nano) semasa penyediaan ampaian. Lapisan elektrolit yang paling berpotensi dengan ciri-ciri nipis, tumpat, kedap gas dan tanpa kecacatan itu terdiri daripada 70 % submikron-YSZ dan 30 % nano-YSZ. HF tersebut menghasilkan kekuatan mekanikal tertinggi iaitu 85 MPa, sifat gas-ketat yang baik iaitu 3.16x10<sup>-6</sup> molm<sup>-2</sup>s<sup>-1</sup>Pa<sup>-1</sup> dan berjaya mengurangkan suhu pensinteran bersama dari 1450 ke 1400 °C. Akhir sekali, ampaian anod yang mengandungi 2 % jisim. PEEK dan ampaian elektrolit dengan campuran saiz zarah mikron, submikron dan nano YSZ dalam nisbah 3:4:3 telah disemperit dan disinter bersama bagi menghasilkan elektrolit/anod dengan struktur terubahsuai. Keputusan menunjukkan bahawa liang terbuka yang berhubung di jalan masuk permukaan dalaman anod hasil daripada penambahan pembentuk liang adalah penyumbang ketara kepada kebolehtelapan gas anod. Walau bagaimanapun, elektrolit tidak tumpat sepenuhnya kerana penyendatan zarah elektrolit yang kurang cekap yang menyebabkan kepada pengurangan kekuatan mekanikal dan integriti HF.

## TABLE OF CONTENTS

CHAPTEI	R TITLE	PAGE	
	TITLE PAGE	i	
	DECLARATION	ii	
	DEDICATION		
	ACKNOWLEDGEMENTS	iv	
	ABSTRACT	v	
	ABSTRAK	vi	
	TABLE OF CONTENTS	vii	
	LIST OF TABLES	xi	
	xiii		
	xvi		
	LIST OF SYMBOLS	xix	
	LIST OF APPENDICES	xxi	
1	INTRODUCTION 1.1 Research Background	1 1	
	1.2 Problem Statements	6	
	1.3 Objectives of Study	7	
	1.4 Scopes of Study	7	
2	LITERATURE REVIEW	9	
	2.1 Introduction 2.2 Operation Principles of Solid Oxida Fuel Cell	9	
	2.2 Operation randiples of solid Oxide Fuel Cell	11	

2.3	High Temperature Solid Oxide Fuel Cell		
	Comp	ponents	14
	2.3.1	Anode as Support 2.3.1.1 Anode Function	14 14
		2.3.1.2 Anode Materials	16
		2.3.1.3 Porous Anode Structure	16
	2.3.2	Thin and Dense Electrolyte Layer 2.3.2.1 Electrolyte Function	18 18
		2.3.2.2 Electrolyte Materials	19
		2.3.2.3 Electrolyte Structure	20
	2.3.3	Porous Cathode 2.3.3.1 Cathode Function	21 21
		2.3.3.2 Cathode Materials	21
2.4	Fabric	cation Techniques of Ceramic Dual-Layer	
	Subst	rate for MT-SOFC	22
	2.4.1	Fabrication Technique of Anode- Supported MT-SOFC 2.4.1.1 Pores in Anode Prepared by Ram	22
		Extrusion and Dry-Jet Wet	
		Extrusion	25
	2.4.2	Deposition of Thin Electrolyte Layer onto	27
	2.4.3	Anode-Supported Hollow Fibre Single-Step Fabrication Method of Dual-	27
2.5	D	Layer Hollow Fibre	28
2.5 Preparation of Ceramic Dual-Layer Hollow Fibre			
	via Ph	hase Inversion-Based Co-Extrusion and Co-	20
	Sinter		30
	2.5.1 2.5.2	Preparation of Spinning Suspensions Phase Inversion Based Co-Extrusion	30
	252	Technique	34
2.6	2.5.3 Sumn	nary of Literature Review	36 39
RES	EARC	H METHODOLOGY	40
3.1	Resea	rch Framework	40
3.2	Select	tion of Materials	40
	3.2.1	Nickel Oxide (NiO)	41
3.3	<ul><li>3.2.2 Yttria Stabilized Zirconia (YSZ)</li><li>3.3 Sintering Shrinkage Analysis</li></ul>		

	3.4	4 Preparation of Anode and Electrolyte Spinning		
		Suspe	nsions	43
		3.4.1	Anode Suspension with Different Pore	
			Former Loading	44
		3.4.2	Anode and Electrolyte Suspension with	
			Different Particle Size	45
	3.5	Fabric	ation of Flat Sheet Anode and	
		Electr	olyte/anode Dual-Layer Hollow Fibre	47
		3.5.1	Flat-Sheet Anode with Different Pore Former Loading Dual-Layer Hollow Fibre with Different	47
	3.6	Chara	Electrolyte Particle Size cterizations of Flat Sheet Anode and Dual-	48
		Layer	Hollow Fibre	49
		3.6.1 3.6.2 3.6.3 3.6.4 3.6.5 3.6.6 3.6.7	X-Ray Diffraction Analysis Viscosity Test Thermo-Gravimetric Analysis Scanning Electron Microscopy Analysis Three-Point Bending Analysis Apparent Porosity Test Gas Permeation Test	50 51 51 51 52 53 54
4	<b>RES</b> 4.1	ULTS A Sinter	AND DISCUSSION ing Shrinkage Behaviour of Raw Materials	56 56
	4.2	Effect	of Pore Former Addition on the Anode	
		Prope	rties	59
		4.2.1 4.2.2 4.2.3 4.2.4	Dope Suspension Viscosity Pore Former Removal During Sintering Microstructure of Flat Sheet Anode Apparent Porosity and Mechanical	59 61 63
	4.3	Effect	Strength of Different Particle Sizes on the	66
		Electr	olyte of Dual Layer Hollow Fibre	68
		4.3.1 4.3.2	Dope Suspension Viscosity Microstructure of Dual-Layer Hollow	68
		4.3.3	Fibre Mechanical Strength and Gas-Tightness of Dual-Layer HF	69 76
	4.4	Comp	arative Study between the Unmodified and	
		Modif	ied Dual Layer Hollow Fibre	79

ix

5 CONCLUSIONS AND RECOMMENDATIONS 5.1 Conclusions			85 85
	5.2	Recommendations	87
REFEREN	NCES		88
Appendix	A-B		94
List ofPub	licatio	ons	99

## LIST OF TABLES

TABLE NO.	TITLE	PAGE
1.1	The advantages and disadvantages to different MT-SOFC performances.	
2.1	Various fuel cell technologies, characteristics and intended applications (Bernay <i>et al.</i> , 2002)	10
2.2	Category of pores in ceramic membranes (Li, 2007).	17
2.3	Major comparisons between plastic mass ram extrusion and dry-jet wet extrusion.	23
2.3	Fabrication of dual-layer hollow fibre via single-step co- extrusion and co-sintering method.	29
2.4	Features and influences of spinning suspension step to the fabricated HF properties.	33
2.5	Features and influences of phase-inversion based co- extrusion step to the fabricated hollow fibre properties.	36
2.6	Features and influences of co-sintering step to the fabricated hollow fibre properties.	38
3.1	The characteristics of various YSZ powders.	42
3.2	Different NiO-YSZ loading in anode composition for sintering shrinkage analysis.	43
3.3	The basic composition of inner (anode) and outer (electrolyte) suspension.	44
3.4	Three batches of formulations during anode suspension preparation with the addition of pore former.	45
3.5	Different loading of YSZ particle sizes in electrolyte suspensions.	46
3.6	Summary of characterizations depending on its purpose and type of samples.	50
4.1	Composition of pore former and ceramic content in anode suspensions.	59

4.2	The compositions of electrolyte suspensions and viscosity at shear rate of 50 s <sup>-1</sup> .	69
4.3	The radius and resultant thickness of layers for HF 8 (precursor) as a function of bore fluid rate.	73
4.4	The radius and resultant thickness of layers for sintered HFs at the same co-extrusion parameters. The fibres were sintered at 1300 °C for 10 hr.	73

## LIST OF FIGURES

FIGURE	TITLE		
NO.			
1.1	Schematic diagram of anode-supported MT-SOFC comprised of a thin electrolyte layer and two conducting electrodes, where anode acts as support.		
2.1	The schematic diagram of internal reforming reaction fueled by hydrogen with air as an oxidant in micro-tubular SOFC.	12	
2.2	Schematic of internal reforming in an SOFC stack showing both direct and indirect reforming approaches.	13	
2.3	Diagram of active site reaction or active three phase boundaries in porous anode SOFC.	15	
2.4	Images of (a) spinneret, (b) internal structure of spinneret, and (c) examples of anode-supported fibre (Yang <i>et al</i> , 2008).	24	
2.5	Schematic diagrams of different structures of MT-SOFCs; (a) Symmetric structure from ram extrusion and, (b) Asymmetric structure from dry-jet wet extrusion.	25	
2.6	SEM images of sintered anode tubes before reduction (a) without pore former (b) with pore former, fabricated using plastic mass ram extrusion technique (Suzuki <i>et al.</i> , 2009).	26	
2.7	Fabrication method of anode-supported micro-tubular SOFCs using combined co-extrusion and co-sintering method.	27	
2.8	Schematic of random dense packing of; (a) Excessive small ceramic particles, (b) Excessive large ceramic particles, and (c) Optimal packing where the small spheres fill all voids in large ceramic particles packing (Li, 2007).	31	
2.9	Schematic diagram of phase inversion-based co- extrusion process.	34	

2.10	Schematic diagram of the final stages of sintering where ceramic microstructure is developed.			
3.1	The research framework in fabricating a high-quality SOFC.			
3.2	Schematic illustration of degassing system.	47		
3.3	Schematic diagram of phase inversion-based casting process.	48		
3.4	Schematic diagram of phase inversion-based co- extrusion process.	49		
3.5	Test setup for 3-P bending test.	53		
3.6	Schematic diagram of test setup for gas-tightness test.	55		
4.1	(a) Sintering curves and (b) sintering rate curves of the electrolyte (100 wt% YSZ) and anode materials using a heating rate of 5 °Cmin <sup>-1</sup> . The anode comprised of various NiO loading of mixture NiO-YSZ.			
4.2	Viscosity of anode suspensions using different batches of the formulation at a shear rate of 50 per second.			
4.3	Sintering profile where thermolysis process takes place.			
4.4	Thermo gravimetric analysis of sintered anode with corn starch and PEEK as pore former.	62		
4.5	XRD patterns of sintered anode with corn starch and PEEK as pore former and the patterns were compared with raw powder of NiO and YSZ. The anode was sintered at 1450 °C.			
4.6	SEM images of the cross-sectional plain anode structure without pore former. The anode was sintered at 1450 °C for 6 hours.			
4.7	SEM micrographs of cross-sectional anode (sintered at 1450 °C for 6 hours) showing the microstructures of anode substrate fabricated with various pore former (a) corn starch and (b) PEEK, at different loadings: (1) 2wt%, (2) 4wt%, (3) 6wt%, (4) 8wt% and (5) 10wt% of total content.			
4.8	Micrographs of (a) corn starch and (b) PEEK particle as pore formers.	66		
4.9	The apparent porosity and mechanical strength of sintered anode as a function of corn starch loading from 0 wt% to 10 wt% of total composition. The anode was sintered at 1450 °C for 6 hours.			

4.10	The apparent porosity and mechanical strength of sintered anode as a function of PEEK loading from 0 wt% to 10 wt% of total composition. The anode was sintered at 1450 °C for 6 hours.	67
4.11	Photographic images of fabricated dual-layer HFs via phase inversion-based co-extrusion method before sinter; a) imperfect lumen and b) well-rounded of lumen.	71
4.12	SEM images of co-sintered dual-layer HFs consisting of different electrolyte microstructure and thicknesses. The HFs were co-sintered at 1300 °C for 10 hours.	75
4.13	Schematic illustrations of the ceramic particle packing principles comprised of; a) sub-micron, b) mixed sub- micron and nano, and c) mixed micron, sub-micron and nano-sized particles.	77
4.14	Mechanical strength of co-sintered dual-layer HFs for different type of electrolyte layer.	77
4.15	Gas tightness of co-sintered dual-layer HFs for different type of electrolyte layer.	77
4.16	SEM images of the cross-sectional of the dual-layer HFs consisting of different anode and electrolyte microstructures. The HFs were co-sintered at 1300 °C for 10 hours.	80
4.17	SEM images of the inner surface of co-sintered anode single layer HFs; a) without pore former, and b) with the addition of 2 wt% PEEK as pore former.	81
4.18	Gas permeability of the unmodified anode (without pore former) single-layer HFs as a function of inlet pressure. The HFs sintered at different target sintering temperature for 10 hours.	82
4.19	Gas permeability of the modified anode (with 2 wt% PEEK as the pore former) single-layer HFs as a function of inlet pressure. The HFs were sintered at different target sintering temperature for 10 hours.	82
4.20	Mechanical strength of co-sintered dual-layer HFs before (HF 2) and after (HF 9 and HF 10) the electrolyte/anode structural modification.	
4.21	Gas-tightness of co-sintered dual-layer HFs before (HF 2) and after (HF 9 and HF 10) the electrolyte/anode structural modification.	
4.22	SEM images of the (a) overall view, (b) cross-sectional view, (c) electrolyte region view, (d) anode inner surface,	

XV

and (e) electrolyte outer surface of co-sintered HF 10	
after the electrolyte/anode structural modification.	84

## LIST OF ABBREVIATIONS

AFC	-	Alkaline fuel cell
ASTM	-	American Society for Testing and Materials
CaO	-	Calcium oxide
CO <sub>2</sub>	-	Carbon dioxide
CGO	-	Cerium gadolinium oxide
DIR	-	Direct internal reforming
DMFC	-	Direct methanol fuel cell
DMSO	-	Dimethyl sulfoxide
Fe <sub>3</sub> O <sub>4</sub>	-	Ferum oxide
$H_2$	-	Hydrogen
H <sub>2</sub> O	-	Water
HF	-	Hollow fibre
HT	-	High temperature
IIR	-	Intermediate internal reforming
IT	-	Indirect temperature
K <sub>2</sub> CO <sub>3</sub>	-	Potassium carbonate
КОН	-	Potassium hydroxide
Li <sub>2</sub> CO <sub>3</sub>	-	Lithium carbonate

LaMnO <sub>3</sub>	-	Lanthanum manganite
LSCF	-	Lanthanum strontium cobalt ferrite
LSM	-	Lanthanum strontium manganite
MCFC	-	Molten carbonate fuel cell
MgO	-	Magnesia
MIEC	-	Mixed ionic electron conductor
МТ	-	Micro-tubular
Na <sub>2</sub> CO <sub>3</sub>	-	Sodium carbonate
Ni	-	Nickel
NiO	-	Nickel oxide
NMP	-	N-methyl-2-pyrrolidinone
O <sub>2</sub>	-	Oxygen
OCV	-	Open circuit voltage
PAFC	-	Phosphoric fuel cell
PEEK	-	Polyetheretherketone
PEMFC	-	Proton exchange membrane fuel cell
PES	-	Polyethersulfone
PF	-	Pore former
PMMA	-	Poly methyl methacrylate beads
RedOx	-	Reduced and Oxidation Atmosphere
Sc <sub>2</sub> O <sub>3</sub>	-	Scandia
SEM	-	Scanning electron microscopy

SOFC	-	Solid oxide fuel cell
SSR	-	Sintering shrinkage rate
TEC	-	Thermal expansion coefficient
TGA	-	Thermo-gravimetric analysis
TPB	-	Triple-phase boundaries
USM	-	Universiti Sains Malaysia
XRD	-	X-ray diffraction
$Y_2O_3$	-	Yttria
YSZ	-	Yttria-stabilized zirconia
ZrO <sub>2</sub>	-	Zirconia

## LIST OF SYMBOLS

Α	-	Area of hollow fibre
b	-	Width
$\mathbf{B}_{\mathrm{F}}$	-	Bending strength
cm	-	Centimetre
cP	-	Centipoise
D	-	Dry weight
d	-	Thickness
$D_i$	-	Inner diameter
$D_O$	-	Outer diameter
g	-	Gram
J	-	Joule
К	-	Kelvin
L	-	Length
L	-	Length of hollow fibre
m	-	Metre
min	-	Minute
mol	-	Mole
Ν	-	Load

nm	-	Nanometre	
М	-	Saturated weight	
Р	-	Gas permeability	
$p_a$	-	Atmospheric pressure	
Pa	-	Pascal	
$p_o$	-	Initial pressure	
$p_t$	-	Final pressures	
R	-	Gas constant	
S	-	Second	
S	-	Suspended weight	
Т	-	Temperature	
t	-	Time for measurements	
V	-	Voltage	
$V_c$	-	Volume of the test cylinder	
W	-	Watt	
wt	-	Weight	
°C	-	Degree Celsius	
%	-	Percent	
μm	-	Micrometre	

### LIST OF APPENDICES

APPENDIX	TITLE	PAGE 94
A	Standard Test Method for Water Absorption, Bulk Density, Apparent Porosity, and Apparent Specific Gravity of Fired Whiteware Products (ASTM C373- 88(2006)).	
В	Schematic diagram of dual-layer hollow fibre.	97

### **CHAPTER 1**

### INTRODUCTION

### 1.1 Research Background

The fuel cell is extensively discovered as clean and sustainable technology for producing electricity. This is due to the major threats that need to be tackled such as the rising of oil price, serious global warming and the soaring of human demand on energy (Traversa, 2009). Fuel cell shows promising characteristics to solve these problems as it consumes fuel at high efficiency with less impact on the environment. The technology does not generate significant amounts of pollutants such as carbon monoxide and nitrogen oxides, and yields more electricity from the same amounts of fuel especially when compared with internal combustion engines (Atkinson *et al.*, 2004).

Generally, the fuel cell is defined as an electrochemical device which converts chemical energy of fuel directly into electrical energy. It produces electricity through a chemical reaction, without undergoing combustion process. The energy conversion system is very efficient in generating electricity either for stationary or transportation applications (Barbir, 2008). In addition to that, the fuel cell is mechanically ideal as it does not require any moving parts during operation, and therefore, making them quiet and reliable sources of power.

The fuel cell is made up of an electrolyte and two conducting electrodes, called as anode (positive electrode) and cathode (negative electrode). The electrolyte allows the ion transfer (depend on the type of fuel cells; anion-conducting and

cation-conducting electrolyte) from one electrode to the other. In electrodes, the operation of fuel cells involves a combined oxidation-reduction reaction. Oxidant is reduced on the cathode side and fuel is oxidized in the anode side. Electrons that released by anode are transferred to the cathode via external load, by which the production of electricity is realized.

There are many kinds of fuel cell have been introduced. However, this study is focusing on solid oxide fuel cell (SOFC) based on solid oxide electrolyte. SOFC becomes a promising technology in the power generation field (Zuo *et al.*, 2012) because of its great flexibility of fuel from biomass to pure hydrogen (Lo Faro *et al.*, 2012). SOFC operates at high operating temperature ranging from 500 °C to 1000°C. Operation in such high operating temperature allows an internal reforming reaction (conversion of hydrogen from hydrocarbon) to occur. Therefore, fuel pre-treatment is not required for SOFC system since the reforming reactions could directly promote within anode cells due to the high operating temperature (Assabumrungrat *et al.*, 2004).

Two structural designs of the SOFC have been commercialized; planar and tubular SOFCs. The tubular design has been introduced by extrusion method in order to prevent the thermal shock problem facing by conventional planar design. Since power density is inversely proportional to the tubular cell diameter, Singhal and Kendall (2003) introduced a smaller cell diameter known as micro-tubular SOFCs (MT-SOFCs) to boost the performance. In fact, the development of this advanced cell design, i.e. MT-SOFCs promote an excellent thermal stability during rapid heat cycling, quick start-up capability, high power output density, low capital cost and portable characteristics as compared to the conventional planar and tubular SOFCs (Meng *et al.*, 2013).

The first generation of MT-SOFC was designed in an electrolyte-supported SOFC system with yttria-stabilized zirconia (YSZ) electrolyte tubes up to 5 mm in diameter. Thick electrolyte layer was first produced to serve as "cell backbone", that is responsible to provide mechanical strength to the entire cell for deposition of the remaining cell layers (Wei *et al.*, 2008). Other promising designs are electrode-

supported SOFC which uses thick anode or cathode as the supporting layer. Table 1 shows the difference of the three supported systems (i.e. Electrolyte, anode and cathode supporting layer). Based on this comparison, it can be seen that anode-supported MT-SOFC is more favourable because it allows the application of a thin electrolyte layer, which results to the reduction in ohmic lose and consequently, enhance cell power density (Buchkremer *et al.*, 1997; Mizutani, 2008).

Current method for fabricating the anode-supported MT-SOFC is very challenging. The SOFC comprises of cathode/electrolyte/anode multi-layer hollow fibre (HF) as shown in Figure 1.1, need to be fabricated. This fabrication of multi-layer HF involves many steps of extrusion and layer depositions (Wei *et al.*, 2008). Each step needs to undergo a sintering process and these repetition steps lead to high fabrication cost. Fortunately, the cost can be reduced by using an economical fabrication method, i.e. single-step co-extrusion and co-sintering method.



**Figure 1.1:** Schematic diagram of anode-supported MT-SOFC comprised of a thin electrolyte layer and two electrodes, where anode acts as support (Othman, 2011).

The single-step fabrication reduces the cost, energy, chemical and time production because it fabricates electrolyte/anode dual-layer hollow fibres (HFs) simultaneously with smaller diameter and better adhesion (Droushiotis *et al.*, 2009). The phase inversion technique was also applied during co-extrusion in order to form asymmetric structure on the prepared HF (Li, 2007). Therefore, in this study, the single-step phase inversion-based co-extrusion and co-sintering method can be employed to fabricate the electrolyte/anode dual-layer hollow fibre with desired anode and electrolyte microstructure.



### Table 1.1: The advantages and disadvantages to different MT-SOFC performances.

Disadvantages	• High ohmic losses resulting	• Low mechanical reliability due to porous	• Lack of study and research based on
	from thick electrolyte layer	structures and low RedOx stability	cathode-supported
	(Mizutani, 2008)	(Sammes et al., 2005)	• Induce chemical reaction between
			cathode and electrolyte at high
			sintering temperature (Sarikaya et
			al., 2012)
			• High polarisation resistance
			(Sarikaya <i>et al.</i> , 2012)

### **1.2 Problem Statements**

The electrolyte/anode dual-layer HF can be fabricated via single-step phase inversion based-co-extrusion method and, followed by co-sintering process. A number of researches have been reported on the feasibility of using this fabrication method to produce well-adhered electrolyte/anode dual-layer HF in economic ways (Droushiotis *et al.*, 2009; Droushiotis *et al.*, 2010; Othman *et al.*, 2010b; Othman *et al.*, 2011b). Nevertheless, the most challenging issues that require to be tackled is during co-sintering. Co-sintering for HF comprises of two different materials at high temperatures (1450  $^{\circ}$ C) to obtain full densification of the electrolyte layer might reduce the porosity in anode layer and it utilizes high energy consumption. Dense electrolyte layer has been compulsory since electrolyte acts as a barrier between electrodes, preventing the direct flow of fuel and oxidant whereas, anode should be porous to facilitate fuel and products.

Therefore, the conventional techniques of adding pore formers and different ceramic particle sizes may be good alternatives to create pores in anode and enhance the particle packing of the electrolyte layer respectively. It was shown that the addition of degradable pore-forming agent such as graphite (Lee, 2003), starch (Haslam *et al.*, 2005), poly methyl methacrylate beads (PMMA) (Suzuki *et al.*, 2009) into the anode suspension had successfully induced the macro-size pore in anode substrate after sintering. While, the addition of different particle sizes (especially nano-sized) is believed to improve the particle packing (Liu *et al.*, 2006) thus increase the densification of electrolyte layer during sintering.

Nevertheless, aforementioned techniques of using pore forming agent and different ceramic particle sizes have never been reported for fabricating electrolyte/anode dual-layer HF using the phase inversion-based co-extrusion and co-sintering method. Hence, this study significantly reports on the effect of pore formers in generating anode with highly porous structure and the influence of particle sizes in producing electrolyte with dense and gas-tight property. As the results of excellent electrolyte densification for dual-layer HF at low co-sintering temperature, it may also preserve the porosity in anode layer and minimizes the energy consumption.

### **1.3** Objectives of Study

The primary purpose of this work is to fabricate yttria-stabilized zirconia (YSZ) - based dual-layer hollow fibre consists of porous anode and dense electrolyte layer for high temperature MT-SOFC. The specific targets are as follows:

- i. To investigate the fabrication of porous anode using corn starch and polyetheretherketone (PEEK) as pore former via phase inversion-based casting method.
- To examine the densification of thin electrolyte layer using different loading of ceramic particle size via phase inversion-based co-extrusion and cosintering method.
- iii. To analyze the properties of prepared anode and electrolyte in terms of their microstructure, crystal structure, mechanical strength, gas-tight and porosity.

### 1.4 Scopes of Study

In order to achieve the objectives of this research, the following scopes are outlined:

- Analyzing the sintering shrinkage behaviour of 30-70 wt% nickel oxide (NiO) loading in anode composition and also the YSZ electrolyte so that well-matched between layers are achieved during co-sintering process.
- ii. Investigating the addition of polyetheretherketone (PEEK) and corn starch as pore former with different loadings from 0-10 wt% into anode suspension as an effort to produce anode with adequate porosity and mechanical strength.

Phase inversion based-casting method and sintering are applied to produce the flat sheets thick anodes.

- iii. Characterizing the anode flat sheet before and after sintering in terms of microstructure, crystal structure, porosity, mechanical durability and thermal stability.
- iv. Examining the effect of different loading of YSZ particle sizes (i.e. Micron, submicron and nano-sized) in electrolyte suspension. HFs are formed through the phase inversion-based co-extrusion method using a triple orifice spinneret and followed by co-sintering at different temperature (1300, 1350, 1400 and 1450 °C).
- v. Examining the properties of the precursor and sintered dual-layer HF using scanning electron microscopy (SEM), gas tightness test and 3-point bending test.
- iv. Comparing the properties of dual-layer HF before and after the anode, and electrolyte modifications. A modified dual-layer HF with a combination of the best pore former loading in anode and different loading of particle sizes in electrolyte was made via phase inversion-based co-extrusion and co-sintering method.

### REFERENCES

- Assabumrungrat, S., Pavarajarn, V., Charojrochkul, S. and Laosiripojana, N. (2004). Thermodynamic Analysis for a Solid Oxide Fuel Cell with Direct Internal Reforming Fueled by Ethanol. *Chemical Engineering Science*. 59 (24), 6015– 6020.
- Atkinson, A., Barnett, S., Gorte, R. J., Irvine, J. T. S., Mcevoy, A. J., Mogenson, M., Singha, S. C. and Vohs, J. (2004). Advanced Anodes for High-Temperature Fuel Cells. *Nature Materials*. 3, 17–27.
- Baker, R. W. (2004). Membrane Technology and Applications. (2<sup>nd</sup> ed.) Chichester, U.K.: Wiley.
- Barbir, F. (2008). Fuel Cell Basic Chemistry, Electrochemistry and Thermodynamics. In Kakaç, S., Pramuanjaroenkij, A. and Vasiliev, L. (Eds.) *Mini-Micro Fuel Cells*. (pp. 13–26). The Netherlands: Springer.
- Bernay, C., Marchand, M. and Cassir, M. (2002). Prospects of Different Fuel Cell Technologies for Vehicle Applications. *Journal of Power Sources*. 108, 139– 152.
- Buchkremer, H. P., Diekmann, U., De Haart, L. G. J., Kabs, H., Stimming, U. and Stöver, D. (1997). Advances in The Anode Supported Planar SOFC Technology. *Proceedings of 5th International Symposium on SOFC*. Aachen, Germany, 160–170.
- Cheng, Z., Wang, J. H. and Liu, M. (2009). Anodes. In Fergus, J. W., Hui R., Li, X., Wilkinson, D. P. and Zhang, J. (Eds.) Solid Oxide Fuel Cells: Materials Properties and Performance. (pp. 73-130). Boca Raton, F. L.:CRC Press.
- Clarke, S. H., Dicks, A. L., Pointon, K., Smith, T. A. and Swann, A. (1997). Catalytic Aspects of The Steam Reforming of Hydrocarbons in Internal Reforming Fuel Cells. *Catalysis Today*. 38, 411–423.

- Dogan, F. (2006). Solid-Oxide Fuel Cells Operating with Direct Alcohol and Hydrocarbon Fuels. In Minteer, S.(Ed.) *Alcoholic Fuels*. (pp. 203-214). Boca Raton, F. L.: CRC Press.
- Droushiotis, N., Doraswami, U., Kanawka, K., Kelsall, G. H. and Li, K. (2009). Characterization of NiO-Yttria Stabilised Zirconia (YSZ) Hollow Fibres For Use as SOFC Anodes. *Solid State Ionics*. 180 (17-19), 1091–1099.
- Droushiotis, N., Othman, M. H. D., Doraswami, U., Wu, Z., Kelsall, G. and Li, K. (2009). Novel Co-Extruded Electrolyte–Anode Hollow Fibres For Solid Oxide Fuel Cells. *Electrochemistry Communications*. 11 (9), 1799–1802.
- Droushiotis, N., Doraswami, U., Ivey, D., Othman, M. H. D., Li, K. and Kelsall, G. (2010). Fabrication by Co-Extrusion and Electrochemical Characterization of Micro-Tubular Hollow Fibre Solid Oxide Fuel Cells. *Electrochemistry Communications*. 12 (6), 792–795.
- Lo Faro, M., Antonucci, V., Antonucci, P. L. and Aricò, A. S. (2012). Fuel Flexibility: A Key Challenge For SOFC Technology. *Fuel*. 102, 554–559.
- Gain, A. K., Song, H. Y. and Lee, B.T. (2006). Microstructure and Mechanical Properties of Porous Yttria Stabilized Zirconia Ceramic Using Poly Methyl Methacrylate Powder. *Scripta Materialia*. 54, 2081–2085.
- Hara, E., Yokozeki, T., Hatta, H., Iwahori, Y., Ogasawara, T. and Ishikawa, T. (2012). Comparison of Out-Of-Plane Tensile Strengths of Aligned CFRP Obtained by 3-Point Bending and Direct Loading Tests. *Composites. Part A: Applied Science and Manufacturing*. 43 (11), 1828–1836.
- Haslam, J. J., Pham, A. -Q., Chung, B. W., DiCarlo, J. F. and Glass, R. S. (2005).
  Effects of the Use of Pore Formers on Performance of an Anode Supported Solid Oxide Fuel Cell. *Journal of the American Ceramic Society*. 88 (3), 513– 518.
- Jamil, S. M., Othman, M. H. D., Rahman, M. A., Jaafar, J., Ismail, A. F. and Li, K. (2015). Recent Fabrication Techniques for Micro-Tubular Solid Oxide Fuel Cell Support: A Review. *Journal of the European Ceramic Society*. 35, 1–22.
- Jang, J. H., Ryu, J. H. and Oh, S. M. (2000). Microstructure of Ni / YSZ Cermets According to Particle Size of Precursor Powders and Their Anodic Performances in SOFC. *Ionics*. 6, 86–91.

- Jiang, S. P. and Li, J. (2009). Cathodes. In Fergus, J. W., Hui R., Li, X., Wilkinson, D. P. and Zhang, J. (Eds.) Solid Oxide Fuel Cells: Materials Properties and Performance. (pp. 131-178). Boca Raton, F. L.: CRC Press.
- Jin, C., Liu, J., Li, L. and Bai, Y. (2009). Electrochemical Properties Analysis of Tubular NiO–YSZ Anode-Supported SOFCs Fabricated by The Phase-Inversion Method. *Journal of Membrane Science*. 341 (1-2), 233–237.
- Kingsbury, B. F. K. and Li, K. (2009). A Morphological Study of Ceramic Hollow Fibre Membranes. *Journal of Membrane Science*. 328 (1-2), 134–140.
- Laosiripojana, N. and Assabumrungrat, S. (2007). Catalytic Steam Reforming of Methane, Methanol, and Ethanol Over Ni/YSZ: The Possible Use of These Fuels in Internal Reforming SOFC. *Journal of Power Sources*. 163, 943–951.
- Lee, J. (2003). The Impact of Anode Microstructure on The Power Generating Characteristics of SOFC. *Solid State Ionics*. 158 (3-4), 225–232.
- Li, K. (2007). *Ceramic Membranes for Separation and Reaction*. Imperial College London, U. K.: Wiley.
- Liu, J. and Barnett, S. A. (2003. Operation of Anode-Supported Solid Oxide Fuel Cells on Methane and Natural Gas. *Solid State Ionics*.158 (1-2), 11–16.
- Liu, L., Tan, X. and Liu, S. (2006). Yttria Stabilized Zirconia Hollow Fiber Membranes. *Journal of the American Ceramic Society*. 89 (3), 1156–1159.
- Lu, K. and Shen, F. (2014). Long Term Behaviors of La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3</sub> and La<sub>0.6</sub>Sr<sub>0.4</sub>Co<sub>0.2</sub>Fe<sub>0.8</sub>O<sub>3</sub> as Cathodes for Solid Oxide Fuel Cells. *International Journal of Hydrogen Energy*. 39 (15), 7963–7971.
- Meng, X., Gong, X., Yang, N., Tan, X., Yin, Y. and Ma, Z. -F. (2013). Fabrication of Y<sub>2</sub>O<sub>3</sub>-Stabilized-ZrO<sub>2</sub>(YSZ)/ La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3-α</sub>–YSZ Dual-Layer Hollow Fibers for The Cathode-Supported Micro-Tubular Solid Oxide Fuel Cells by a Co-Spinning/Co-Sintering Technique. *Journal of Power Sources*. 237, 277–284.
- Mizutani, Y. (2008). Planar Solid Oxide Fuel Cells: From Materials to Systems. In Kakaç, S., Pramuanjaroenkij, A. and Vasiliev, L. (Eds.) *Mini-Micro Fuel Cells*. (pp. 349–358). The Netherlands: Springer.
- O'Hayre, R., Cha, S. -W., Colella, W. and Fritz, B. P. (2009). *Fuel Cell Fundamentals*. (2<sup>nd</sup> ed.). Hoboken, N. J.: Wiley.
- Othman, M. H. D. (2011). High Performance Micro-Tubular Solid Oxide Fuel Cell. Imperial College London.

- Othman, M. H. D., Droushiotis, N., Wu, Z., Kanawka, K., Kelsall, G. and Li, K. (2010a). Electrolyte Thickness Control and Its Effect on Electrolyte/Anode Dual-Layer Hollow Fibres For Micro-Tubular Solid Oxide Fuel Cells. *Journal* of Membrane Science. 365 (1-2), 382–388.
- Othman, M. H. D., Wu, Z., Droushiotis, N., Doraswami, U., Kelsall, G. and Li, K. (2010b). Single-Step Fabrication and Characterisations of Electrolyte/Anode Dual-Layer Hollow Fibres For Micro-Tubular Solid Oxide Fuel Cells. *Journal* of Membrane Science. 351 (1-2), 196–204.
- Othman, M. H. D., Wu, Z., Droushiotis, N., Kelsall, G. and Li, K. (2010c). Morphological Studies of Macrostructure of Ni–CGO Anode Hollow Fibres For Intermediate Temperature Solid Oxide Fuel Cells. *Journal of Membrane Science*. 360 (1-2), 410–417.
- Othman, M. H. D., Droushiotis, N., Wu, Z., Kelsall, G. and Li, K. (2011a). High-Performance, Anode-Supported, Microtubular SOFC Prepared From Single-Step-Fabricated, Dual-Layer Hollow Fibers. *Advanced materials*. 23 (21), 2480–2483.
- Othman, M. H. D., Droushiotis, N., Wu, Z., Kelsall, G. and Li, K. (2011b). Novel Fabrication Technique of Hollow Fibre Support For Micro-Tubular Solid Oxide Fuel Cells. *Journal of Power Sources*. 196 (11), 5035–5044.
- Othman, M. H. D., Droushiotis, N., Wu, Z., Kelsall, G. and Li, K. (2012). Dual-Layer Hollow Fibres With Different Anode Structures For Micro-Tubular Solid Oxide Fuel Cells. *Journal of Power Sources*. 205, 272–280.
- Poon, M. and Kesler, O. (2012). The Influence of Pore Formers on The Microstructure of Plasma-Sprayed NiO – YSZ Anodes. *Journal of Power Sources*. 210, 204–217.
- Primdahl, S., Sørensen, B. F. and Mogensen, M. (2000). Effect of Nickel Oxide / Yttria-Stabilized Zirconia Anode Precursor Sintering Temperature on the Properties of Solid Oxide Fuel Cells. *Journal of the American Ceramic Society*. 83 (3), 489–494.
- Sammes, N. M., Du, Y. and Bove, R. (2005). Design and Fabrication of A 100 W Anode Supported Micro-Tubular SOFC Stack. *Journal of Power Sources*.145, 428–434.

- Sanson, A., Pinasco, P. and Roncari, E. (2008). Influence of Pore Formers on Slurry Compositionand Microstructure of Tape Cast Supporting Anodes For SOFCs. *Journal of the European Ceramic Society*. 281. 221–1226.
- Sarikaya, A., Petrovsky, V. and Dogan, F. (2012). Effect of The Anode Microstructure on The Enhanced Performance of Solid Oxide Fuel Cells. *International Journal of Hydrogen Energy*. 37 (15), 11370–11377.
- Schoenbein, C. F. (1839). On The Voltaic Polarization of Certain Solid and Fluid Substances. *Phil. Mag. (III)*. 14, 43–45.
- Singhal, S. C. and Kendall, K. (2003). High temperature solid oxide fuel cells: Fundamentals, design, and applications. (1<sup>st</sup> ed.). Oxford, U. K.: Elsevier Ltd.
- Sun, C. & Stimming, U. (2007). Recent anode advances in solid oxide fuel cells. Journal of Power Sources. 171, 247–260.
- Sun, C. W., Hui, R. and Roller, J. (2010). Cathode Materials For Solid Oxide Fuel Cells: A Review. *Journal of Solid State Electrochemistry*. 14, 1125–1144.
- Suzuki, T., Yamaguchi, T., Fujishiro, Y. and Awano, M. (2006). Fabrication and Characterization of Micro Tubular SOFCs For Operation in The Intermediate Temperature. *Journal of Power Sources*. 160, 73–77.
- Suzuki, T., Funahashi, Y., Yamaguchi, T., Fujishiro, Y. and Awano, M. (2009). Effect of Anode Microstructure on The Performance of Micro Tubular SOFCs. *Solid State Ionics*. 180 (6-8), 546–549.
- Tan, X., Liu, S. and Li, K. (2001). Preparation and Characterization of Inorganic Hollow Fiber Membranes. *Journal of Membrane Science*. 188, 87–95.
- Tarancón, A. (2009). Strategies for Lowering Solid Oxide Fuel Cells Operating Temperature. *Energies*. 2 (4), 1130–1150.
- Tompsett, G. A., Finnerty, C., Kendall, K., Alston, T. and Sammes, N. M. (2000). Novel Applications For Micro-SOFCs. *Journal of Power Sources*. 86 (1-2), 376–382.
- Traversa, E. (2009). Toward the Miniaturization of Solid Oxide Fuel Cells. *The Electrochemical Society Interface*. 18, 49–52.
- Wei, C. C., Chen, O. Y., Liu, Y. and Li, K. (2008). Ceramic Asymmetric Hollow Fibre Membranes—One Step Fabrication Process. *Journal of Membrane Science*. 320 (1-2), 191–197.

- Wu, Z., Wang, B. and Li, K. (2010). A Novel Dual-Layer Ceramic Hollow Fibre Membrane Reactor For Methane Conversion. *Journal of Membrane Science*. 352 (1-2), 63–70.
- Xia, C. (2009). Electrolytes. In Fergus, J. W., Hui R., Li, X., Wilkinson, D. P. and Zhang, J. (Eds.) Solid Oxide Fuel Cells: Materials Properties and Performance. (pp. 1-72). Boca Raton, F. L.: CRC Press.
- Yang, N., Tan, X. and Ma, Z. (2008). A Phase Inversion/Sintering Process to Fabricate Nickel/Yttria-Stabilized Zirconia Hollow Fibers as The Anode Support for Micro-Tubular Solid Oxide Fuel Cells. *Journal of Power Sources*. 183 (1), 14–19.
- Zuo, C., Liu, M. and Liu, M. (2012). Solid Oxide Fuel Cells. In Aparicio, M., Jitianu,
  A. & Klein, L. C. (Eds.) Sol-Gel Processing for Conventional and Alternative Energy. (pp. 7–36). New York, U. S.: Springer.