THE GROWTH MECHANISM IN SELF-ASSEMBLY NANOSTRUCTURES OF SILICON/SILICON DIOXIDE INTERFACE

FATIMA ALDAW IDREES

A thesis submitted in fulfilment of the requirements for the award of the degree of Doctor of Philosophy (Physics)

Faculty of Science Universiti Teknologi Malaysia

FEBRUARY 2013

Thanks for "ALLAH" guide me to succeed my work, and thanks for my parents, siblings, nephews and my friends, thanks for encourage me towards academic with pleasant experience.

ACKNOWLEDGEMENTS

I am very pleased for those who support me to write this thesis. First of all, I would like to express my sincerest appreciation to my supervisor, Prof. Dr. Samsudi Sakrani, and my co-supervisor, Associate Prof. Dr. Zulkafli Othaman for their advice and encouragement.

My further appreciation to my research "QuaSR" group members who had supported me to success this project. Their assist and encouragement are inestimably important. Next, I like to thanks all the lab assistants and science officers from Physics Department of Science Faculty and Ibnu Sina Institute for Fundamental Science Studies, UTM for their assist during my research Mr. Imam Sumpono, Mr. Mohamad Nazri and Md. Wani from Ibnu Sina Institute, UTM, and Md. Norhayah, Md. Junidah.

Also, I wish to thank the Ministry of Higher Education for providing part of the financial support through FRSG funding 78416.

Last but not least, I would like to thank my dear parents, family members, my sister Hala, brother Mohamed, nephews Ahmed and Awab for supporting me to pass the troubles during my work.

ABSTRACT

Silicon nanodots is a common zero-dimensional nanomaterial investigated for single-electron device applications in integrated circuits. The current study attempts to look into the ever-popular silicon self-assembly nanodot grown on different substrates, with emphasis on its growth theory and characterizations. Discrepancy in its growth theories has led to misunderstanding and therefore innovative approaches are presented in this study to clarify and resolve the existing problems. A radiofrequency magnetron sputtering method was used for Silicon nanodots deposition, with the following conditions: argon gas flow rate 5-10 sccm, substrate temperature between 300-600 °C, deposition time 7-20 minutes, and radio-frequency power between 100-150 W. This research covers both experimental and simulation works including the classical theory of nucleation. Generally, important parameters were first calculated then simulated using computer programming, and finally matched in order to estimate the values of critical energy ΔG^* , critical radius r^* , surface energy γ , and free energy change per unit area ΔG_{γ} . The associated Volmer-Weber growth mode was then predicted. Observably, optimum growth parameters for the inception of silicon nanodots were found to be at 600 °C/10 minutes/100W formed on corning Structural and optical properties have been characterized using atomic force microscope AFM, energy-dispersive X-ray spectroscopy EDX, X-Ray diffraction XRD, photoluminescence PL and scanning electron microscopy SEM. In addition, the AFM characterization results show the existence of nanodots with the estimated average size of 34.4 nm. The results from PL spectrum reveal the presence of a peak which corresponds to a bandgap energy of 1.80 eV and this was attributed to the quantum confinement of electron-hole pairs in quantum wells. A further confirmation using EDX measurement was made which showed the existence of 0.48 at.% of silicon on the substrate. XRD analysis reveals the crystalline structure for high temperature conditions due to orderly silicon nanodots formed on the substrate. The results proved that the properties of silicon nanodots on quartz SiO₂, corning glass (7059) and silicon substrates were strongly dependent on the experimental conditions.

ABSTRAK

Bintik-nanosilikon merupakan satu bahan nano berdimensi sifar yang dikaji bagi kegunaan peranti elektron tunggal dalam litar bersepadu. Kajian ini cuba mendalami bahan yang semakin diminati di kenali sebagai bintik-nanosilikon yang terbentuk-sendiri pada substrat berbeza, dengan tumpuan kepada teori pertumbuhan dan penciriannya. Percanggahan mengenai teori pertumbuhan telah menyebabkan kecelaruan dan oleh itu pendekatan yang lebih inovasi dipersembahkan dalam kajian ini bagi meleraikan masalah sedia ada. Satu kaedah percikan frekuensi magnetron telah digunakan bagi menghasilkan bintik-nanosilikon pada keadaan berikut: kadar aliran gas argon 5-10 sccm, suhu substrat diantara 300-600 °C, masa pemendapan 7-20 minit, dan kuasa frekuensi-radio 100-150 W. Penyelidikan ini merangkumi kedua-dua eksperimen dan kerja simulasi termasuk teori klasik mengenai penukleusan. Pada dasarnya, parameter yang penting dikira terlebih dahulu ke mudian di simulasi kan menggunakan program komputer dan akhirnya disepadankan kedua-duanya bagi memperolehi nilai-nilai tenaga genting ΔG^* , jejari genting r^* , tenaga permukan γ, dan perubahan tenaga bebas per unit luas ΔGv. Mod pertumbuhan Volmer-Weber kemudiannya boleh diramalkan daripada hasil tersebut. Secara pemerhatian di dapati bahawa parameter pertumbuhan yang optimum bagi bintik-nano silicon adalah 500 °C/ 10 minit/ 100 W pada substrat kaca corning. Pencirian sifat-sifat struktur dan optik telah di lakukan dengan menggunakan mikroskop daya atom AFM, sinar-X sebaran elektron EDX, pembelauan sinar-X XRD, luminesenfoto PL dan mikroskop electron imbasan SEM. Sebagai tambahan, keputusan pencirian AFM telah menunjukkan tentang kewujudan bintik-nano dengan anggaran saiz sekitar 34.4 nm. Keputusan spectrum PL pula memaparkan adanya satu puncak dengan tenaga jalur jurang 1.80 eV yang boleh dikaitkan dengan pemerangkapan kuantum bagi pasangan electron-lohong dalam perigi kuantum. Seterusnya keputusan yang diperolehi telah di sahkan dengan menggunakan pengukuran EDX dimana sebanyak 0.48 atom.% kandungan silikon wujud pada permukaan substrat. Pemerhatian analisis XRD pula memperlihatkan struktur hablur bagi suhu tinggi yang disebabkan oleh pembentukan bintik-nanosilikon yang lebih teratur pada permukaan substrat, Semua keputusan di atas membuktikan bahawa sifat-sifat bintik-nanosilikon pada substrat SiO₂, kaca corning (7059) dan silikon adalah sangat bergantung kepada keadaan eksperimen.

TABLE OF CONTENTS

CHAPTER	TITLE	PAGE
	DECLARATION	ii
	DEDICATION	iii
	ACKNOLEDGEMENTS	iv
	ABSTRACT	v
	ABSTRAK	vi
	TABLE OF CONTENTS	vii
	LIST OF TABLES	xi
	LIST OF FIGURES	xii
	LIST OF ABBREVIATIONS	xvii
	LIST OF SYMBOLS	xix
	LIST OF APPENDICES	xxiii
1	INTRODUCTION	1
	1.1 Research Background	1
	1.2 Statement of Problem	5
	1.3 Research Objectives	9
	1.4 Scope of The Study	10
	1.5 Significance of The Study	11
2	LITERATURE REVIEW	12
	2.1 Introduction	12
	2.2 Bulk Material and Quantum Dots	12
	2.3 Structure Properties of Silicon Nanocrystals	14
	2.4 Optical Properties of Silicon Nanocrystals	15
	2.5 The Quantum Well and Multilayers	20

	2.6	Classi	cal Theory of Nucleation	26
		2.6.1	Homogeneous Nucleation	28
		2.6.2	Heterogeneous Nucleation:	33
	2.7	The G	rowth Modes	35
	2.8	Chemi	ical Vapour Deposition (CVD)	39
	2.9	Physic	cal Vapour Deposition (PVD)	39
	2.10	Radio	Frequency Magnetron Sputtering	43
	2.11	Sputte	ring Yield	47
	2.12	Chapte	er summary	48
3	RES	EARCI	H METHODOLOGY	49
	3.1	Introd	uction	49
	3.2	Simula	ation Program	49
	3.3	Nanoc	lots Growth Procedure	54
		3.3.1	Sample Preparation	54
		3.3.2	Quartz Substrate	55
		3.3.3	Corning Glass (7059) Properties (Barium	56
			Borosilicate)	
		3.3.4	Silicon Substrate	56
	3.4	Radio	Frequency Magnetron Sputtering System	57
	3.5	High \	Vacuum Coater Setting Parameters	62
		3.5.1	Si/SiO ₂ Thin Film Growth Parameters	62
		3.5.2	Nanodots Growth Parameters	63
		3.5.3	Rapid Thermal Annealing Parameters	67
	3.6	QDs N	Measurements	68
		3.6.1	Atomic Force Microscopy (AFM)	68
		3.6.2	Photoluminescence (PL)	69
		3.6.3	Energy Dispersive X-Ray Spectroscopy	72
			(EDX)	
		3.6.4	X-ray Diffraction (XRD)	73
		3.6.5	Scanning Electron Microscope (SEM)	77
	3.7	Chapte	er Summary	78

4	RES	ULTS .	AND DISCUSSIONS	79
	4.1	Introd	uction	79
	4.2	Simul	ation Result	80
	4.3	Surfac	ee Morphology Analysis	84
	4.4	Atomi	c Force Microscopy (AFM) Measurements	85
		4.4.1	Quartz Substrate	85
		4.4.2	Corning Glass Substrate	94
		4.4.3	Silicon Substrate	102
		4.4.4	Silicon Thin Film Annealing	108
	4.5	The G	rowth of Si/SiO ₂ Thin Film	111
	4.6	Scann	ing Electron Microscopy (SEM)	112
		Measu	rement	
	4.7	Energ	y Dispersive X-ray Spectroscopy (EDX)	114
		Measu	rement	
		4.7.1	Quartz Substrate	114
		4.7.2	Corning Glass (7059) Substrate	116
		4.7.3	Silicon Substrate	118
	4.8	High l	Resolution X-ray Diffraction (HR-XRD)	120
	4.9	Photo	luminescence (PL)	124
		4.9.1	Corning Glass (7059) Substrate	124
		4.9.2	Silicon Substrate	128
		4.9.3	Si/SiO ₂ Thin Film	132
	4.10	Chapt	er Summary	138
5	CONCLUSION			140
	5.1	Summ	nary	140
	5.2	Sugge	estions and Future Work	142
REFERENCES				143
Appendices A - B				153

LIST OF TABLES

TABLE NO	TITLE	PAGE
2.1	The four main dimensionality arrangements	18
3.1	The SiNDs growth parameters for quartz substrate at fixed RF power = 100 W.	62
3.2	The SiNDs growth parameters for corning glass (7059) substrate.	64
3.3	The SiNDs growth parameters for each SiO ₂ substrate	65
3.4	SiNDs and Si/SiO $_2$ thin films annealing parameters under N $_2$ flow rate at fixed gas pressure 200 Pa.	66
3.5	Si thin films annealing parameters - N_2 gas pressure 200 Pa.	67
4.1	Experimental parameters and simulation result	80
4.2	Average dots size and surface roughness with deposition times.	88
4.3	Average dots size and surface roughness with substrate temperature.	90
4.4	Average dots size and surface roughness with substrate temperature.	100
4.5	Average dots size and surface roughness with deposition times.	101
4.6	Average dots size and surface roughness with substrate	106

temperature.

4.7	Average dots size and surface roughness with deposition times.	107
4.8	Annealing temperature with islands density.	110
4.9	Atomic and weight composition of silicon thin film deposition on quartz substrate at different deposition times.	115
4.10	Atomic and weight composition of silicon grown on corning glass (7059) substrate at different times of deposition.	117
4.11	Atomic and weight composition of silicon grown on silicon substrate at different deposition times.	119
4.12	Size of Si (100) nanocrystals as estimated from (400) Bragg diffraction for samples deposited at various deposition times.	123
4.13	Substrate annealing temperature and average grain size with the surface roughness.	137

LIST OF FIGURES

FIGURE NO	TITLE	PAGE
1.1	The crystalline and amorphous silicon thin film nanostructure.	6
	Silicon thin film deposition and the growth parameters of RF	7
	magnetron sputtering system.	
1.3	The literature survey of radio frequency magnetron sputtering.	8
2.1	Meeting Bottom-up technique and top-down technique.	13
2.2	Electron-beam lithography processing for single-electron device	14
	fabrication.	
2.3	Silicon four valance electrons.	15
2.4	Electron's orbit around the single nucleus.	16
2.5	Energy transition among VB and CB levels for: (a) bulk Si, (b)	17
	Quantum dots.	
2.6	(a) Si indirect and (b) GaAs direct energy gap.	19
2.7	The diagram of optical transition of SiO ₂ /Si/SiO ₂ multilayers as a	20
	quantum confinement in PL1 influenced by Si sublayer thickness	
	and (b) Si/SiO ₂ interface PL2.	
2.8	PL spectrums obtained for MLs while only the Si sublayer	22
	thickness varies.	
2.9	PL spectrums for Si/SiO ₂ multilayers with varied Si layer	23
	thickness: (a) - (c) 4.5 nm, (d) 3 nm, (e) 1.5 nm, also (a) as-	
	deposited, annealed at (b) 800 °C, and (c),(d),(e)1100 °C.	
2.10	The XRD patterns of the Si/SiO ₂ multilayers annealed at as-	23
	deposited 800–1100 °C.	
2.11	The AFM images of the Ge/Si thin films growth at varied	24
	sputtering powers: (a) 2 W/cm ² ; (b) 6 W/cm ² .	
2.12	Normalized PL peaks for samples with SiNDs diameters from 1.8	25

	to 7 nm.	
2.13	PL spectrums for amorphous Si films grown at substrate	25
	temperature 200 °C and filament temperatures: $1-1735$ °C, $2-$	
	1785 °C and 3 − 1881 °C.	
2.14	The atomic nucleation and deposition process.	26
2.15	(a) Homogeneous nucleation, (b) heterogeneous nucleation.	27
2.16	Configuration of microcrystallites by liquid.	28
2.17	Nucleation of nanocrystallites. (a) Variation of Δg_a vs.	31
	temperature, $\Delta g_a = 0$ at T_d . (b) Variation of ΔG_f , free enthalpy of	
	nucleus formation, vs radius of nucleus.	
2.18	Quantum dot formation and three surface energies.	33
2.19	Three growth modes (F-M, V-W and S-K mode).	36
2.20	PVD and CVD vapor deposition.	39
2.21	RF magnetron sputtering system.	41
3.1	The simulation procedure flow chart.	49
3.2	Software program via heterogeneous nucleation.	50
3.3	Heterogeneous nucleation software toolbar (plot graphs display).	51
3.4	Heterogeneous nucleation software toolbar (Goto display).	51
3.5	Software program via homogeneous nucleation.	53
3.6	(a) The three substrates of Silicon, corning glass (7059) and	55
	quartz, (b) substrate holder with screw, (c) Purity of 99.99999%	
	SiO ₂ target, (d) Purity of 99.99999% Si target.	
3.7	High Vacuum Chamber.	58
3.8	The substrate heater in HVC system.	59
3.9	Radio-frequency magnetron sputtering system.	60
3.10	Si/SiO ₂ thin films deposition flow chart.	61
3.11	SiNDs and SiO ₂ experimental procedure.	62
3.12	Schematic diagram AFM instrument.	8
3.13	Quantum well and quantum dot luminescence.	71
3.14	PL instrument structure.	72
3.15	(a) Emission of x-ray by an atom, (b) Elementary processes	73
	visualized in the term of scheme.	
3.16	Crystal diffraction of X-ray.	74

3.17	X-ray diffraction of crystal solid and liquid or amorphous solid.	75
3.18	Diffraction of X-ray beam by crystal.	76
	SEM instrument structure.	77
4.1	(a) Simulation and (b) Experimental nanodots configuration result.	81
4.2	Critical energy versus temperature.	82
4.3	Critical radius versus temperature.	82
4.4	AFM topography image of blank Si wafer with RMS of 0.0532	80
	nm.	
4.5	AFM image for SiNDs grown on quartz substrate RF power 100	84
	W, gas flow rate 5 Sccm and deposition times (a) 12, (b) 15 min	
	temperature of 200 °C (scan area 1000 nm).	
4.6	AFM image for SiNDs deposited on quartz substrate RF power	85
	100 W, gas flow rate 5 Sccm, deposition time 15 mins, substrate	
	temperature 200 °C with surface roughness 0.47 nm.	
4.7	AFM image for SiNDs grown on quartz substrate RF power 100	86
	W, gas flow rate 5 Sccm, deposition time 5 min temperature of	
	600 °C.	
4.8	Average dots size and surface roughness versus substrate	87
	temperature.	
4.9	AFM image for SiNDs grown on quartz substrate at different	88
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	200 °C, RF power 100 W and gas flow rate 5 Sccm.	
4.10	Average dots size and surface roughness versus deposition time.	89
4.11	AFM image for SiNDs grown on quartz substrate at different	90
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	200 °C, RF power 100 W and gas flow rate 10 Sccm.	
4.12	AFM image for SiNDs grown on quartz substrate at different	92
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	300 °C, RF power 100 W and gas flow rate 10 Sccm.	
4.13	AFM image for SiNDs grown on quartz substrate at different	93
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	300 °C, RF power 100 W and gas flow rate 5 Sccm.	
4.14	AFM dot size measurement for SiNDs grown on corning glass	94

	substrate at deposition time of 10 min, temperature of 600 °C, RF	
	power 100 W and gas flow rate 5 Sccm.	
4.15	AFM image for SiNDs grown on corning glass substrate at	95
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min,	
	temperature of 600 °C, RF power 150 W and gas flow rate 5 Sccm	
	at scan area 1000 nm.	
4.16	AFM image for SiNDs grown on corning glass (7059) substrate at	96
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min,	
	temperature of 500 °C, RF power 100 W and gas flow rate 5	
	Scem.	
4.17	AFM image for SiNDs grown on corning glass (7059) substrate at	97
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min,	
	temperature of 400 °C, RF power 100 W and gas flow rate 5	
	Scem.	
4.18	AFM image for SiNDs grown on corning glass (7059) substrate at	98
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min,	
	temperature of 400 °C, RF power 150 W and gas flow rate 5	
	Scem.	
4.19	AFM image for SiNDs grown on corning glass substrate at	99
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min,	
	temperature of 400 °C, RF power 150 W and gas flow rate 10	
	Seem.	
4.20	Average dots size and surface roughness versus the Substrate	100
	temperature.	
4.21	Average dots size and surface roughness versus the deposition	101
	times.	
4.22	AFM (a) 3D Images (b) cross section for SiNDs grown at	102
	temperature 600 °C, RF power 100 W and deposition time 10 min,	
	at scan area 500 nm.	
4.23	AFM images of SiNDs grown on silicon substrate at different	103
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	400 °C, RF power 100 W and gas flow rate 5 Sccm.	
4.24	AFM image for SiNDs grown on silicon substrate at different	103

	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	500 °C, RF power 150 W and gas flow rate 5 Sccm.	
4.25	AFM images of SiNDs grown on silicon substrate at different	105
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min, temperature of	
	500 °C, RF power 100 W and gas flow rate 5 Sccm.	
4.26	AFM images for SiNDs deposited on silicon substrate at 15 mins	105
	deposition time, substrate temperature 500 °C, RF power 150 W	
	and gas flow rate 5 Sccm.	
4.27	Average dots size and the surface roughness with substrate	106
	temperature.	
4.28	Average dots size and the surface roughness with substrate	107
	deposition times.	
4.29	AFM images of silicon NDs grown on silicon substrate: (a) before	109
	annealing (5000 nm scan area), (b) before annealing (1000 nm	
	scan area), (c) after annealing for 5 min and 600 °C (5000 nm scan	
	area), (d) after annealing for 5 min and 600 °C (1000 nm scan	
	area) (e) after annealing for 10 min and 800 $^{\circ}\text{C}$ (5000 nm scan	
	area), (f) after annealing for 10 min and 800 °C (1000 nm scan	
	area).	
4.30	AFM images of Si/SiO ₂ grown on silicon substrate at different	110
	deposition times (a) as-SiO ₂ deposited, (b) 5, (c) 10, (d) 15 min	
	for Si, temperature of 400 °C, RF power 100 W and gas flow rate	
	5 Scem.	
4.31	Plan-view SEM image for Si/SiO ₂ multilayers grown on silicon	112
	substrate: (a) Si nanoparticles view and (b) Si nanoparticles size	
	measurement at time of deposition 10 min temperature of 400 °C,	
	RF power 100 W and gas flow rate 10 Sccm.	
4.32	Cross-section view SEM for $\mathrm{Si}/_{\mathrm{SiO2}}$ multilayers grown on silicon	113
	substrate at different deposition time 10 min (a) Si and (b)Si/Si O_2	
	thin films thickness, temperature of 400 °C, RF power 100 W and	
	gas flow rate 10 Sccm.	
4.33	EDX patterns of quartz substrate before silicon thin film	114
	deposition.	

4.34	EDX patterns of silicon nanodots growth on quartz substrate with	115
	deposition time 10 min, substrate temperature 400 °C, and RF	
	power 100 W.	
4.35	EDX patterns of corning glass substrate before silicon nanodots	116
	deposition.	
4.36	EDX patterns of Silicon nanodots growth on corning glass (7059)	117
	substrate with deposition time 10 min, substrate temperature 400	
	°C and RF power 100 W.	
4.37	EDX pattern of silicon substrate before silicon nanodots	118
	deposition.	
4.38	EDX patterns of silicon nanodots growth on silicon substrate with	119
	deposition time 10 min, substrate temperature 400 °C, and RF	
	power 100 W.	
4.39	XRD patterns of bulk crystalline Si.	120
4.40	XRD patterns of silicon substrate.	121
4.41	XRD patterns of SiNDs growth on silicon substrate at different	122
	deposition times: (a) before deposition, (b) 8, (c) 10, (d) 15 min	
	and substrate temperature 450 °C, RF power 100 W, and gas flow	
	rate 5 sccm.	
4.42	Photoluminescence characterization for SiNDs grown on corning	125
	glass at time of deposition 5 min, RF power 100 W and different	
	temperature (a) 300, (b) 400, (c) 500, (d) 600 °C.	
4.43	Photoluminescence characterization for SiNDs grown on corning	126
	glass at time of deposition 15 min, RF power 100 W and different	
	temperature (a) 300, (b) 400, (c) 500, (d) 600 °C.	
4.44	Photoluminescence characterization for SiNDs grown on corning	127
	glass at growth temperature 500 °C, RF power 100 W and	
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min	
4.45	Photoluminescence characterization for SiNDs grown on silicon at	129
	growth temperature 500 °C, RF power 100 W and different	
	deposition times (a) 5, (b) 7, (c) 10, (d) 15 min.	
4.46	Photoluminescence characterization of SiNDs grown on silicon at	129
	growth temperature 400 °C, RF power 100 W and different	

	deposition times (a) 7, (b) 10, (c) 15, (d) 20 min.	
4.47	Photoluminescence characterization for SiNDs grown on silicon at	130
	time of deposition 15 min, RF power 100 W and different	
	temperature (a) 300, (b) 400, (c) 450, (d) 500 °C.	
4.48	Photoluminescence characterization for SiNDs grown at silicon	131
	substrate and annealing at different temperature (a) as-deposit, (b)	
	500, (c) 600, (d) 700, (e) 800 °C.	
4.49	Photoluminescence characterization for Si/SiO ₂ thin film grown	133
	on silicon at growth temperature 500 °C, RF power 100 W and	
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min.	
4.50	Photoluminescence characterization for Si/SiO ₂ thin film grown	134
	on silicon at growth temperature 600 °C, RF power 100 W and	
	different deposition times (a) 5, (b) 7, (c) 10, (d) 15 min.	
4.51	Photoluminescence characterization for Si/SiO ₂ thin film grown	135
	on silicon at time of deposition 15 min, RF power 100 W and	
	different temperature (a) 300, (b) 400, (c) 500, (d) 600 °C.	
4.52	Photoluminescence characterization for Si/SiO ₂ thin film	136
	annealing at different temperature (a) as-deposit, (b) 500, (c) 600,	
	(d) 700, (e) 800 °C.	
4.53	Annealing temperature versus the energy gap for both Si and	137
	Si/SiO ₂ samples.	

LIST OF ABBREVIATIONS

AFM - Atomic force microscope

AO - Atomic orbit

CB - Conduction band

CVD - Chemical vapor deposition

DC - Direct current

EBL - Electron beam lithography

EDX - Energy dispersive X-ray

FESEM - Field emission scanning electron microscope

F-M - Frank Vander Merwe

FWHM - Full width at half maximum

HRTEM - High resolution transmission electron microscope

HVC - High vacuum coater

ITRS - International technology roadmap of semiconductor

LC - Loading circuit

LCD - Liquid crystal display

LPCVD - Low pressure chemical vapor deposition

MBE - Molecular beam epitaxy

ML - Multilayer

Mo - Molecular orbit

MosFETs - Metal oxide semiconductor field effect transistor

 A_n - Nanoparticles

Nb₃Ge - Niobium germanium

n - Number of atoms

0D - One dimension

PL - Photoluminescence

PECVD - Plasma-enhanced chemical vapour deposition

PL - Photoluminescence

Ps - Porous silicon

PVD - Physical vapour deposition

QC - Quantum confinement

QW - Quantum well

RF - Radio frequency

RIE - Reactive ion etching

RMS - Root mean square

RTP - Rapid thermal annealing processing

S,P,D,F - Spherical orbit

SEM - Scanning electron microscope

SiNDs - Silicon nanodot

SiQDs - Silicon quantum dots

S-K - Stranski Krastanov

SiNWs - Silicon nanowire

TRE - Thermally reactive evaporation

3D - Three dimensions

TEM - Transmission electron microscope

2D - Two dimensions

UHV - Ultra high vacuum

VB - Valence band

VLSI - Very large scale integrated circuits

V-W - Volmer Weber

GIXRD - Grazing incidence X-ray diffraction

0D - Zero dimension

- Extreme programming

LIST OF SYMBOLS

a - New phase volume

Ar - Argon

 Ar^+ - Argon ion

a-Si - Amorphous silicon

at. % - Atomic percentage

c - Radius of nuclei

 c_{hetro}^* - Critical radius for heterogeneous

 c^*_{homo} - Critical radius for homogenous

*c*_i - Number of nuclei atoms

 c_0 - Number of liquid atoms

c-Si - Crystalline silicon

 e^{-} - Electron

 E_c - Conduction energy

 E_g - Energy bandgap

Ek - The kinetic energy

eV - Electron Volt

Ga - Gallium

As - Arsenide

Ge - Germanium

 h, \hbar - Planck's constant

K - Boltzmann constant

k - Wavevector or momentum vector

 L_h - Latent heat of the material fusion per unit of volume

m - Effective mass

N - Dimensionality

 N_c - Directions of confinement

 N_f - Freedom grades

 N_2 - Nitrogen

O - Oxygen

RMS - Root mean square roughness

Si - Silicon

SiO - Silicon monoxide

SiO₂ - Silicon dioxide

Si-O - Silicon oxide bond

sccm - Standard cubic centimeters per minute

T - Temperature

 T_d - Transition temperature or solidification temperature

 2θ - Diffraction angle (two-theta)

< - Less than

γ - Surface energy change per unit area

 γ_S - Surface energy between the liquid-substrate

 γ_N - Surface energy between the liquid-solid nucleus

 γ_{NS} - Surface energy between the solid nucleus-substrate

 ΔC_{ν} - Variation in specific heat

 ΔC_{ν}^{h} - Variation in the specific heats at the melting point

 ΔE - Energy gap

 ΔG - The total variation in free energy

 ΔG_a - Variation in free energy change per unit volume

 ΔG_c - Variation in free energy change per unit surface

 Δg_a - Free enthalpy per unit volume

 ΔG_f - Net change of energy for homogenous nucleation

 ΔG_f^* - Critical energy for homogenous nucleation

 Δh_a - Change in enthalpy

 Δs_a - Entropy change

 ΔT - Undercooling temperature

 θ - Contact Angle

 λ - Wavelength

 π - Pi = 3.1415926

 v_o - Molecular volume

v - Velocity

LIST OF APPENDICES

APPENDIX	TITLE	PAGE
A	Photographs of Experimental Equipments	153
В	Published Papers	156

CHAPTER 1

INTRODUCTION

1.1 Research Background

From the beginning of the last century, nanotechnology has become one of the most common techniques promising for new artificial atoms having diameter less than 100 nm to fabricate nanodevices into single integrated chip [1,2]. Now the International Technology Roadmap of Semiconductors (ITRS) predicts that the physical gate length of high performance metal oxide semiconductor field effect transistor (MOSFETs) will reach sub 10 nm in 2016 [3]. Therefore, the fast improvement in the fabrication techniques of silicon electronic devices and single electron very large scale integrated circuits (VLSI) has been achieved. This has been gradually developed by miniaturizing the device dimensions by exponential growth of microelectronics capabilities. Currently, single electron transistor device was reported using silicon self-assembled nanodots (NDs) and room temperature characteristics have been measured. Consequently, there is a chance for using SiNDs as a memory cell, due to the nanodots parallel to the multitunnel junction structure that has strong changed of random background charges. On the other hand, maintaining the conventional top-down methods improvement becomes very hard due to the basic phenomena, scientific confines and cost effectiveness [4].

Thin film semiconductors devices around over decades have been fabricated by a thin film deposited on a semiconductor substrate. However the integrated

electronic circuits system depends on electrical charges confined in the interfaces between different materials with different electronic properties. This system created by composite different materials of the thin film such as thin films deposition that has the desired properties on substrate. The material structure applications at high temperature depend on thin film methods to provide the suitable coating, increase the efficiency and increase the material lifetime. The surface and near surface properties are important in varying the functionality of the material (catalysis, deformation, electrical and optical properties). Hence, rapidly growing of economics and application require high performance, low cost, compact and reliable adaptable devices that are driving the technology of thin film.

The fabrication of thin film microstructure must carry out the influences of contaminations in the vapour phase, kinetic energy of the incident ions, base pressure, deposition rate, nature of substrate material, substrate temperature; in addition, its surface cleanliness, nanostructure. The deposited microstructure can also be affected by energetic particles such as electron photon and ion bombardment. Operating the growth parameters permits to avoid the defects in the depositing polycrystalline film, as well as control the point of thin film thickness and grain size in thin films, which is only possible in some cases with certain deposition process. In integrated circuits a low resistivity film and high threshold current density is required for interconnections by semiconducting films. All of these properties are particularly influenced by the fault structure of the film, and in fact it depends on the techniques used to fabricate the thin film [5].

The technique of thin film deposition has four stages. First, the essential configuration of the vapor phase from the condensed phase and its properties. There are some methods of vapor formation such as evaporation, sputtering, electron beam evaporation and ion deposition. The second stage is the transport of ionized particles from the target to the substrate. Also some techniques consider the reactivation of evaporated atoms during the transport to the substrate such as reactive evaporation, electron cyclotron resonance plasma assisted growth and other else. The most important field of thin film fabrication is the study of the capability to form an epitaxial thin film. This capability has exclusive control the formation, construction

and faults of the thin film. Thirdly, concerning those procedures that lead to reorganize the coalescence on the substrate surface, or rearrange the structure of the thin film to give the desired properties for the device. Some characteristics are desired to fabricate the final microstructure of the thin film appearance indicates the controlling of nanostructure and surface morphology of the thin films by set the defects position as preferred or give them the desired crystal structure [5,6].

Self-assembled Si nanostructure deposition by using radio frequency sputtering technique has much recommended for large-scale integrated circuits fabrication. The production of silicon nanostructures has opportunity to use for electrical devices. The magnetron sputtering method becomes one of the most common methods for fabricating the silicon nanostructures due to high deposition rate and safety currently [2]. In fact to make sufficient devices, it is important to improve the methods which permit the commercial production of semiconductor thin films. The usual ions transportation from target to substrate will establish the regularity of that thin film. The real properties of the film are affected by the substrate selected and its surface cleanliness. Moreover the nanostructure of the films through deposition depends on different parameters, such as the substrate temperature, ion energy and angle of incidence of the depositing particles [7].

Thin film techniques are important to produce nanoscale materials, such as quantum wires, quantum dots and superlattices. All these structures make thin film field not only more attractive material properties and process reliant, but also give interesting novels to explain the material phenomena. Thin film properties are clearly related to the existence of surface at two materials in close connection with each other. Where the properties vary from bulk material, divides the amount of atoms at any time on the surface on average 10¹⁰ atoms/m [5].

Nanostructure of materials and its properties (the electron and phonon states) change radically as their size decrease in one or more dimensions to nanometer scale, that have follow down almost the atomic energy levels [8]. Si nanodots are one of typical materials used in nanotechnologies, because of their unique and useful

functions originating from quantized electron energy state [3]. The observed visible luminescence at room temperature from Si gives attention to improve the electronic and oscillation properties of silicon nanocrystals. The recognized electronic devices motivates many investigations of different Si nanostructures using varied preparation methods. Currently, research in nanocrystalline silicon nc-Si confirmed considerable changes in the luminescence properties and energy levels, due to quantum confinement caused by the miniaturized size of the particles and a collapse of *k*-vector protection [9]. In addition, zero- and one dimensional nature of electronic states in the individual SiNDs realizes new electronic and photonic properties, which are not achieved with bulk silicon [3]. Nevertheless, the observations show some disagreement with QC create from other phenomenon that may come from the PL. However, the PL properties are clearly following the measured structure (composite, ML, amorphous, crystalline, etc.) and also depend on the fabrication method [2].

In the practical applications, it is important to form high-density and similarly-sized SiNDs. Although various formation techniques have been developed so far, it is generally difficult to achieve high-density and nanometer-size with low dispersion of distribution simultaneously. Silicon nanostructures can be formed on non-Si substrates, such as glass and plastic, the Si-based bottom-up approach may lead to high performance and large-area electronics [8].

The research area of nanostructure silicon fabrication methods have been widely improved last time as the finding of room-temperature photoluminescence PL function of porous silicon. The researchers were faithful to nanodots structures 0D and to layered formations 2D for the construction of optoelectronic devices suitable for the silicon structure, due to the strong mechanical performance and stable porous silicon 1D. For explaining the combined materials consist of SiNDs embedded in apparent and insulating matrix SiO₂, they are usually prepared using plasma enhanced chemical vapour deposition PECVD or by magnetron sputtering of Si⁺ ion implantation in thermally grown silica. The ML structure clears more fitting than the complex material, while it presents at least a confident of the nanoparticles size in the formation method [2, 10]. Moreover, hard and costly equipment of molecular beam epitaxy system which limits the number of multilayers. On the other hand, the

magnetron sputtering technique shows more improving of visible photoluminescence Si/SiO₂ [11].

The accurately controlled nanostructures can be fabricated easily using self-assembly of atoms by operating the organic molecules as a structure block for nanoscale devices which have more consideration. This bottom-up approach changed the hard processes of the existing silicon top-down approach. Since the conductivity of the organic molecular configurations is less than silicon conductivity, the electron moving among the single molecule is mainly controlled by passing the conduction. Silicon nanodots SiNDs and nanowires SiNWs can give a resolution to these problems by assembling the involvements of both bottom-up approach and optimum electron transport. Since silicon nanostructures can be grown on non-Si substrates, such as glass and plastic, the silicon supported bottom-up techniques may guide to fine properties and wide electronics applications. However, nanostructure considers new electrical and photonic behaviors, which are not realized with bulk silicon. Also interconnecting the bottom-up approach with the classic top-down Si methods permit to investigate silicon nanoelectronics [5, 12].

1.2 Statement of Problem

Silicon nanostructure has two types: crystalline nanostructure which has regular periodic arrangement of atoms in the lattice structure such as Poly crystalline silicon, and amorphous that has dangling bonds and random arrangement of atoms such as hydrogenated amorphous silicon as shows in Fig 1.1. This nanostructure depends on the substrate temperature during thin film deposition. Therefore, high substrate temperature causes the kinetic energy of atoms exceeding the surface barrier and relocated during deposition; as a result, increase the surface roughness of the thin film. Whereas, low temperature causes the sputtered atoms lose their kinetic energy before catch the substrate and creates amorphous thin film nanostructure [13,14].

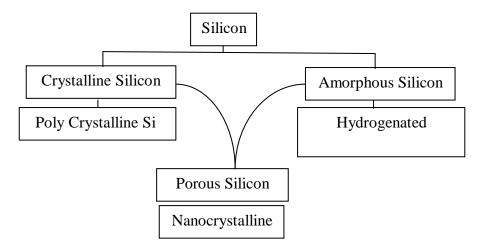


Figure 1.1 The crystalline and amorphous silicon thin film nanostructure.

Silicon nanofabrication is essential to motivate nanotechnology [15]. The real challenges in nanomaterials fabrication are the accuracy of the diameter and fine identified shapes. To fabricate a nanodevice, considerations must be given to nanoscale diameters, shapes, and properties of materials or components [16]. The silicon thin film nanostructure grown by using RF-magnetron sputtering system is affected by the different growth parameters such as substrate temperature as shows in Fig 1.2. However by changing the substrate temperature different nanodots diameter, islands density, islands shape, surface roughness, crystalline structure and optical properties could be obtained. It is essential to precise the size of the nanoparticles, their density, and surface morphology, as the emission in visible scope is clearly showed by a quantum confinement. Widely, the fabrication system applied these values and the reduction of nanodots size follow by energy gap increasing [17,18].

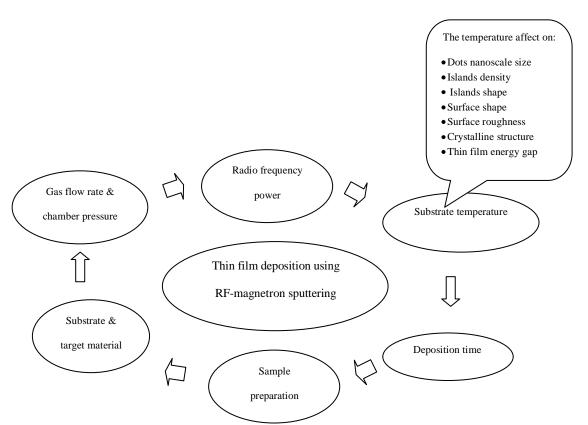


Figure 1.2 Silicon thin film deposition and the growth parameters of RF magnetron sputtering system.

There were few improvement of works using the conventional deposition techniques which discovered the formation of SiNDs with less identify structures. RF-magnetron sputtering system – which located at Ibnu Sina institute for fundamental studies, UTM - had been used [20,21] to deposit silicon thin films at substrate temperature less than 400 °C, as shows in Figure 1.3. Whereas low substrate temperature less than 300 °C cause defects (dangling bonds) in the thin film nanostructure, that create the random arrangement of the atoms in the lattice structure (amorphous structure).

In fact the silicon/silicon dioxide nanocrystal is desired for high electron mobility in the microelectronic devices, which is low in amorphous silicon. Therefore, by replacing each atom of silicon with oxygen [19] to reduce the typical defect like silicon dangling bonds, Si/SiO₂ multilayers could keep a lattice match for low defects [11, 17, 20-24].

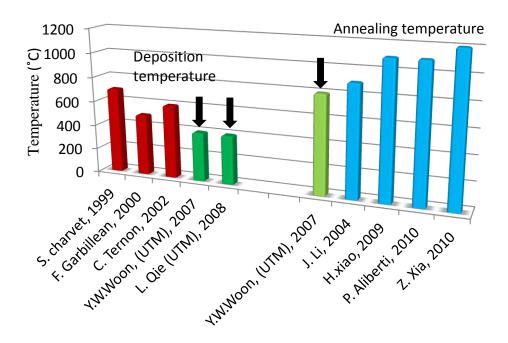


Figure 1.3 The literature survey of radio frequency magnetron sputtering.

This study is involved which will result in quality of the SiNDs fabrication on different substrates. The effect of temperature, chamber pressure, gas flow rate, and RF-power on structural, optical and electrical properties of SiNDs growth has been given attention so far. SiNDs deposited in plasma excitation radio frequency also depend on these parameters. So by varying these parameters, the corresponding structural and optical properties are expected to be significantly improved [7]. Therefore, to obtain silicon thin film with a high crystalline structure and high surface roughness, it is proposed to increase the substrate temperature until 500 °C or more.

Nucleation is a random process; in which the number of nuclei formed in a fixed period of times is a random quantity and is subject to statistical laws. However, the average values can be calculated and are subject to the kinetic theory of nucleation. This research has been done to compare the simulation results with the experimental works and confirm the classical theory of nucleation and growth mode of *Volmer-Weber*. The formation of SiNDs can be predicted using a simple model of the islands growth, where the atoms or vapours diffuse to each other

followed by agglomeration that form the nuclei of new phase. Thus, this research describes the nucleation and growth of silicon nanodots for the vapour/liquid transition with detailed analysis of parameters deduced from the theory and experimental [25].

1.3 Research Objectives

The objectives of this research as to:

- 1. Reestablish the simulation works in order to confirm the classical theory of nucleation and growth mode of *Volmer-Weber*.
- 2. Determine the optimum growth parameters of SiNDs and Si/SiO₂ interface for HVC system (deposition time, substrate temperature, gas flow rate, gas pressure, and radio frequency power), as well as modify the HVC system.
- Characterize silicon thin film structural properties of surface roughness and nanndots size using AFM measurements follow by SEM measurement, and determine the silicon atomic percentage that has deposit on the substrate by used EDX measurements.
- 4. Characterize the energy gap (optical properties) of silicon thin film using PL measurement to confirm the reduction of nanodots size follow by energy gap increasing.
- 5. Characterize the grain (nanocrystal) size and silicon thin film crystalline structure using XRD measurements.

1.4 Scope of the Study

This research explains the theoretical and simulation procedure of heterogeneous nucleation. The simulation work represents the nucleation and growth theory, including the transition from liquid-solid phase, also to prove the appearance changing for net energy and nuclei size. The simulation program is conducted using Microsoft Visual Basic 6.0 professional. As well as to determine the calculated parameters using a computer programe are designed for a dome-like nucleus assumed in Volmer-Weber growth theory.

Recently self-assembly technique to fabricate SiNDs is experimentally carrying out by using radio frequency magnetron sputtering system for future nanodevice applications. In this research, the quartz SiO₂, corning glass (7059), and silicon wafer have been chosen as substrates. Therefore, the experimental conditions are identified at high substrate temperature of 600 °C in order to create silicon thin film with high crystalline structure. Consequently, this deposition process is performed using different growth parameters (deposition time, substrate temperature, gas flow rate, gas pressure, and radio frequency power). SiNDs structural properties are measured using AFM, EDX, SEM and XRD equipments following by optical properties using PL measurement to calculate the energy gap of silicon thin film.

1.5 Significance of the Study

This study is essential to enhance the understanding of SiNDs growth mechanism and its properties. The possibility to form a semiconductor thin film on the surface of the glass substrates open a new aspect in understanding the structure, electrical and optical properties of thin films. However these properties are attracting researchers to fabricate various structural properties which are absolutely unlike the naturally accessible properties [5].

Thin film deposition technique is adding new commercial devices and allowing further characteristic in fabrication. Therefore it is a promising to improve the devices utility and also keep the possessions materials wastes by the conventional fabrication methods. In addition the quantum confinement of charge carriers improves the field of nanostructure as influences the properties of materials. The key to convergence the computing, communications and consumer electronics is the optical, magnetic and electronic properties of thin films. It is apparent that thin film has an important function to influence humanity in the future [5].

Recently, research related to Si nanostructure has been given consideration and continuing trend towards integration of microelectronics and optoelectronics devices. However, bulk Si has an indirect band-gap structure that limited its application in optoelectronics for a very long time, due to the lower luminescence efficiency [27]. Silicon becomes the preferred material for single-electron and quantum electronic devices due to its unique structure, electrical and optical properties. Recently SiNDs applications in electronics device is recommended and agreed by the design of single memory and other devices requires the coulomb blockade effect [16].

REFRENCES

- 1. Bryant, G.W. and G. Solomon, *Optics of quantum dots and wires*. 2005: Artech House.
- 2. Ternon, F. Gourbilleau, R. Rizk, and C. Dufour, Si/SiO₂ multilayers: synthesis by reactive magnetron sputtering and photoluminescence emission. Physica E: Low-dimensional Systems and Nanostructures, 2003. **16**(3-4): p. 517-522.
- 3. Mizuta, H. and S. Oda, *Bottom-up approach to silicon nanoelectronics*. Microelectronics Journal, 2008. **39**(2): p. 171-176.
- 4. Zahi, H. Vergnes, B. Caussat, A. Estève, M. Djafari Rouhani, P. Mur, P. Blaise, and E. Scheid, *Towards multiscale modeling of Si nanocrystals LPCVD deposition on SiO₂: From ab initio calculations to reactor scale simulations*. Surface and Coatings Technology, 2007. **201**(22-23): p. 8854-8858.
- 5. SreeHarsha, *Principles of physical vapor deposition of thin films*. 2006: Elsevier.
- 6. Cha, G. Kim, H.J. Doerr, and R.F. Bunshah, *Effects of activated reactive evaporation process parameters on the microhardness of polycrystalline silicon carbide thin films*. Thin Solid Films, 1994. **253**(1-2): p. 212-217.
- 7. Yoo, C.S., Semiconductor manufacturing technology. 2008: World Scientific.
- 8. Henini, A. Patanè, A. Polimeni, A. Levin, L. Eaves, P.C. Main, and G. Hill, *Electrical and optical properties of self-assembled quantum dots*. Microelectronics Journal, 2002. **33**(4): p. 313-318.
- 9. Sirenko, R. Fox, I.A. Akimov, X. Xi, S. Ruvimov, and Z. Liliental-Weber, *In situ Raman scattering studies of the amorphous and crystalline Si nanoparticles*. Solid State Communications, 2000. **113**(10): p. 553-558.

- 10. Baron, T., F. Martin, P. Mur, C. Wyon, and M. Dupuy, *Silicon quantum dot nucleation on Si*₃N₄, *SiO*₂ and *SiO*_xN_y substrates for nanoelectronic devices.

 Journal of Crystal Growth, 2000. **209**(4): p. 1004-1008.
- 11. Gourbilleau, P. Voivenel, X. Portier, and R. Rizk, *A novel method for the deposition of Si-SiO*₂ *superlattices*. Microelectronics Reliability, 2000. **40**(4-5): p. 889-892.
- 12. Sun, and H.C. Kang, Synthesis and characterization of self-assembled ZnO nano-dots grown on SiN_x/Si(001) substrates by radio frequency magnetron sputtering. Thin Solid Films, 2010. **518**(22): p. 6522-6525.
- 13. S. Sakrani, F.A. Idrees, Z. Othaman, and A.K. Ismail, *The Growth Mechanism of Silicon Nanodots Synthesized by Sputtering Method.* AIP Conference Proceedings, 2011. **1341**(1): p. 109-113.
- 14. Lopez Villanueva, J.A. Jimenez Tejada, A. Palma, S. Rodriguez Bolivar, and J.E. Carceller. *A simple model to analyze electron confinement and trapping in silicon nanodots*. in *Electron Devices*, 2005 Spanish Conference on. 2005.
- 15. Teo, X.H. Sun, *Silicon-Based Low-Dimensional Nanomaterials and Nanodevices*. Chemical Reviews, 2007. **107**(5): p. 1454-1532.
- 16. Xiao, S. Huang, J. Zheng, G. Xie, and Y. Xie, *Optical characteristics of Si/SiO*₂ multilayers prepared by magnetron sputtering. Microelectronic Engineering, 2009. **86**(11): p. 2342-2346.
- 17. Huang, Z. Xia, H. Xiao, J. Zheng, Y. Xie, and G. Xie, *Structure and property of Ge/Si nanomultilayers prepared by magnetron sputtering*. Surface and Coatings Technology, 2009. **204**(5): p. 558-562.
- 18. Jung, Y.M. Jung, L.R. Shaginyan, and J.G. Han, *Polycrystalline Si thin film growth on glass using pulsed d.c. magnetron sputtering*. Thin Solid Films, 2002. **420-421**: p. 429-432.
- 19. Lewis, B. and J.C. Anderson, *Nucleation and growth of thin films*. 1978: Academic Press.
- L.Qie Jie, 2008, Silicon self-assembled nanodots prepared using radio frequency magnetron sputtering, Master Thesis, Universiti Teknologi Malasyia, Skudai.
- 21. Yeong Wai Woon, 2007, *The formation structural and optical properties of silicon nanocrystals embedded in silicon dioxide*, Master Thesis, Universiti Teknologi Malaysia, Skudai.

- 22. Charvet, R. Madelon, F. Gourbilleau, and R. Rizk, *Ellipsometric* spectroscopy study of photoluminescent Si/SiO₂ systems obtained by magnetron co-sputtering. Journal of Luminescence, 1998. **80**(1-4): p. 257-261.
- 23. Ternon, F. Gourbilleau, X. Portier, P. Voivenel, and C. Dufour, *An original approach for the fabrication of Si/SiO*₂ *multilayers using reactive magnetron sputtering*. Thin Solid Films, 2002. **419**(1-2): p. 5-10.
- 24. Li, X.L. Wu, D.S. Hu, Y.M. Yang, T. Qiu, and J.C. Shen, *Splitting of X-ray diffraction peak in (Ge:SiO₂)/SiO₂ multilayers*. Solid State Communications, 2004. **131**(1): p. 21-25.
- 25. Aliberti, S.K. Shrestha, R. Teuscher, B. Zhang, M.A. Green, and G.J. Conibeer, *Study of silicon quantum dots in a SiO*₂ *matrix for energy selective contacts applications*. Solar Energy Materials and Solar Cells, 2010. **94**(11): p. 1936-1941.
- 26. Xia, Z. and S. Huang, Structural and photoluminescence properties of silicon nanocrystals embedded in SiC matrix prepared by magnetron sputtering. Solid State Communications, 2010. **150**(19-20): p. 914-918.
- 27. Kondo, T. Ueyama, E. Ikenaga, K. Kobayashi, A. Sakai, M. Ogawa, and S. Zaima, *Formation of high-density Si nanodots by agglomeration of ultra-thin amorphous Si films*. Thin Solid Films, 2008. **517**(1): p. 297-299.
- 28. Ruzmetov, Y. Seo, L.J. Belenky, D.M. Kim, X. Ke, H. Sun, V. Chandrasekhar, C.B. Eom, M.S. Rzchowski, and X. Pan, *Epitaxial magnetic perovskite nanostructures*. Advanced Materials, 2005. **17**(23): p. 2869-2872.
- 29. Ignac Capek, Slovak Academy of Sciences, Dúbravská cesta 9, Bratislava, Slovakia, Chapter 1 Nanotechnology and nanomaterials, in Studies in Interface Science, I. Capek, Editor. 2006, Elsevier. p. 1-69.
- 30. Diener, D. Kovalev, G. Polisski, and F. Koch, *Luminescence properties of two-photon excited silicon nanocrystals*. Optical Materials, 2001. **17**(1-2): p. 117-120.
- 31. Zhang, S. Shrestha, S.J. Huang, P. Aliberti, M.A. Green, and G. Conibeer, Structural studies of multilayered Ge nanocrystals embedded in matrix fabricated using magnetron sputtering. Energy Procedia, 2010. **2**(1): p. 243-250.

- 32. Teo, X.H. Sun, T.F. Hung, X.M. Meng, N.B. Wong, and S.T. Lee, *Precision-Cut Crystalline Silicon Nanodots and Nanorods from Nanowires and Direct Visualization of Cross Sections and Growth Orientations of Silicon Nanowires*. Nano Letters, 2003. **3**(12): p. 1735-1737.
- 33. A. Biswas, S. Bayer, A. Biris, T. Wang, E. Dervishi, F. Faupel, Advances in top–down and bottom–up surface nanofabrication: Techniques, applications and future prospects, Vol 170, Issues 1–2, 15 January 2012, Pages 2–27
- 34. Li, S.H. Chen, and H.L. Chen, *Thermal-flow techniques for sub-35nm contact-hole fabrication using Taguchi method in electron-beam lithography*. Microelectron. Eng., 2009. **86**(11): p. 2170-2175.
- 35. Guo, E. Leobandung, L. Zhuang, and S.Y. Chou, *Fabrication and characterization of room temperature silicon single electron memory*. Journal of Vacuum Science & Technology B: Microelectronics and Nanometer Structures, 1997. **15**(6): p. 2840-2843.
- 36. Moliton, Solid-State Physics for Electronics. 2009: John Wiley & Sons.
- 37. Harrison, Quantum wells, wires, and dots: theoretical and computational physics. 2000: John Wiley & Sons.
- 38. Schmid, G., Nanoparticles: from theory to application. 2004: Wiley-VCH.
- 39. Fox, *Optical properties of solids*. 2010: Oxford University Press.
- 40. Germanenko, S. Li, S.J. Silvers, and M.S. El-Shall, *Characterization of silicon nanocrystals and photoluminescence quenching in solution*. Nanostructured Materials, 1999. **12**(5-8): p. 731-736.
- 41. Yu, X. Wang, W. Lu, S. Wang, Y. Bian, and G. Fu, *Effects of substrate temperature on microstructural and photoluminescent properties of nanocrystalline silicon carbide films*. Physica B: Condensed Matter, 2010. **405**(6): p. 1624-1627.
- 42. Nayfeh, L. Mitas, Silicon Nanoparticles: New Photonic and Electronic Material at the Transition Between Solid and Molecule, in Nanosilicon, K. Vijay, Editor. 2008, Elsevier: Amsterdam. p. 1-78.
- 43. Eberl, O.G. Schmidt, O. Kienzle, and F. Ernst, *Preparation and optical properties of Ge and C-induced Ge quantum dots on Si.* Thin Solid Films, 2000. **373**(1-2): p. 164-169.
- 44. Chiquito, Y.A. Pusep, S. Mergulhão, J.C. Galzerani, and N.T. Moshegov, Investigation of the InAs/GaAs self-assembled quantum dots using the

- relationship between the capacitance and the density of states. Physica E: Low-dimensional Systems and Nanostructures, 2001. **9**(2): p. 321-325.
- 45. Dolino, D. Bellet, Variations in the lattice parameter of porous silicon produced by wetting and vapour adsorption. Thin Solid Films, 1995. **255**(1-2): p. 132-134.
- 46. Charvet, R. Madelon, R. Rizk, B. Garrido, O. González-Varona, M. López, A. Pérez-Rodríguez, and J.R. Morante, *Substrate temperature dependence of the photoluminescence efficiency of co-sputtered Si/SiO*₂ *layers*. Journal of Luminescence, 1998. **80**(1-4): p. 241-245.
- 47. Rebohle, J. von Borany, W. Skorupa, I.E. Tyschenko, and H. Fröb, *Photoluminescence and electroluminescence investigations at Ge-rich SiO2 layers*. Journal of Luminescence, 1998. **80**(1-4): p. 275-279.
- 48. Ma, L. Wang, K. Chen, W. Li, L. Zhang, Y. Bao, X. Wang, J. Xu, X. Huang, and D. Feng, *Blue light emission in nc-Si/_{SiO2} multilayers fabricated using layer by layer plasma oxidation*. Journal of Non-Crystalline Solids, 2002. **299-302**(Part 1): p. 648-652.
- 49. Crupi, D. Corso, S. Lombardo, C. Gerardi, G. Ammendola, G. Nicotra, C. Spinella, E. Rimini, and M. Melanotte, *Memory effects in MOS devices based on Si quantum dots*. Materials Science and Engineering: C, 2003. **23**(1-2): p. 33-36.
- Xiaoli, S. Yi, G. Shulin, Z. Jianmin, Z. Youdou, S. Kenichi, I. Hiroki, and H. Toshiro, Effects of interface traps in silicon-quantum-dots-based memory structures. Physica E: Low-dimensional Systems and Nanostructures, 2000.
 8(2): p. 189-193.
- Puglisi, S. Lombardo, G. Ammendola, G. Nicotra, and C. Gerardi, *Imaging of Si quantum dots as charge storage nodes*. Materials Science and Engineering: C, 2003. 23(6-8): p. 1047-1051.
- 52. Lombardo, S. Coffa, C. Bongiorno, C. Spinella, E. Castagna, A. Sciuto, C. Gerardi, F. Ferrari, B. Fazio, and S. Privitera, *Correlation of dot size distribution with luminescence and electrical transport of Si quantum dots embedded in SiO*₂. Materials Science and Engineering B, 2000. **69-70**: p. 295-298.

- 53. Chen, Y. Ren, R.l. Xiong, Y.y. Zhao, and M. Lu, *Modulation of the photoluminescence of Si quantum dots by means of CO₂ laser pre-annealing*. Applied Surface Science, 2010. **256**(16): p. 5116-5119.
- 54. Reinig, B. Selle, F. Fenske, W. Fuhs, V. Alex, and M. Birkholz, *Highly oriented growth of polycrystalline silicon films on glass by pulsed magnetron sputtering*. Vol. 20. 2002: AVS. 2004-2006.
- 55. Torchynska, *Photoluminescence of Si nanocrystallites in different types of matrices*. Journal of Non-Crystalline Solids, 2006. **352**(23-25): p. 2484-2487.
- 56. Torchynska, T., J. Aguilar-Hernandez, M. Morales Rodriguez, C. Mejia-Garcia, G. Contreras-Puente, F.G. Becerril Espinoza, B.M. Bulakh, L.V. Scherbina, Y. Goldstein, A. Many, and J. Jedrzejewski, Comparative investigation of photoluminescence of silicon wire structures and silicon oxide films. Journal of Physics and Chemistry of Solids, 2002. 63(4): p. 561-568.
- 57. Papon, J. Leblond, and P.H.E. Meijer, *The physics of phase transitions:* concepts and applications. 2006: Springer-Verlag.
- 58. Myerson, R. Ginde, *Crystals, crystal growth, and nucleation*, in *Handbook of Industrial Crystallization (Second Edition)*, S.M. Allan, Editor. 2002, Butterworth-Heinemann: Woburn. p. 33-65.
- 59. Spaepen, Homogeneous Nucleation and the Temperature Dependence of the Crystal-Melt Interfacial Tension, in Solid State Physics, E. Henry and T. David, Editors. 1994, Academic Press. p. 1-32.
- 60. Salk, D.E. Hagen, and C.K. Lutrus, *Propensity rule in homogeneous nucleation processes*. Chemical Physics Letters, 1988. **146**(6): p. 605-608.
- 61. Rao, B.J. Berne, *Nucleation in finite systems: Theory and computer simulation*. Astrophysics and Space Science, 1979. **65**(1): p. 39-46.
- 62. Mullin, *Nucleation*, in *Crystallization (Fourth Edition)*. 2001, Butterworth-Heinemann: Oxford. p. 181-215.
- 63. Cao, Nanostructures & nanomaterials: synthesis, properties & applications. 2004: Imperial College Press.
- 64. Idrees, F.A., S. Sakrani, and Z. Othaman, *Formation and Characterization of Silicon Self-assembled Nanodots*. AIP Conference Proceedings, 2011. **1341**(1): p. 324-327.

- 65. Priester, C. and M. Lannoo, *Growth aspects of quantum dots*. Current Opinion in Solid State and Materials Science, 1997. **2**(6): p. 716-721.
- 66. Le Thanh, V. and V. Yam, Superlattices of self-assembled Ge/Si(0 0 1) quantum dots. Applied Surface Science, 2003. 212-213: p. 296-304.
- 67. Larsson, A. Elfving, P.O. Holtz, G.V. Hansson, and W.X. Ni, *Luminescence* study of Si/Ge quantum dots. Physica E: Low-dimensional Systems and Nanostructures, 2003. **16**(3-4): p. 476-480.
- 68. Talalaev, G.E. Cirlin, A.A. Tonkikh, N.D. Zakharov, P. Werner, U. Gösele, J.W. Tomm, and T. Elsaesser, *Miniband-related 1.4-1.8 [mu]m Luminescence of Ge/Si Quantum Dot Superlattices*, in *Handbook of Self Assembled Semiconductor Nanostructures for Novel Devices in Photonics and Electronics*, H. Mohamed, Editor. 2008, Elsevier: Amsterdam. p. 324-345.
- 69. Müller, O. Kirfel, A. Rastelli, H. von Känel, and D. Grützmacher, Successful shape-preservation of Ge-clusters during Si-coverage at low temperature. Materials Science and Engineering B, 2003. 101(1-3): p. 142-145.
- 70. Zhou, R.A. Johnson, and H.N.G. Wadley, *Vacancy formation during vapor deposition*. Acta Materialia, 1997. **45**(11): p. 4441-4452.
- 71. Wood, Thin-film optical filters: 3rd edition: H.A. Macleod; Thin Film Centre Inc., Tuscon, Arizona & University of Arizona, USA, Institute of Physics Publishing, 2001, 672pp, ISBN 0-7503-0688-2. Optics and Lasers in Engineering, 2002. 37(6): p. 673-674.
- 72. Palmero, H. Rudolph, and F.H.P.M. Habraken, *One-dimensional analysis of the rate of plasma-assisted sputter deposition*. Vol. 101. 2007: AIP. 083307.
- 73. Margaritondo, R. Joynt, M. Onellion, and A.V. Society, *High Tc* superconducting thin films, devices, and applications, *Atlanta*, *GA*, 1988. 1989: American Institute of Physics.
- 74. Bach, D. Krause, *Thin films on glass*. 1997: Springer.
- 75. Shi, Functional thin films and functional materials: new concepts and technologies. 2003: Springer.
- 76. Mattox, *Physical Sputtering and Sputter Deposition (Sputtering)*, in *Handbook of Physical Vapor Deposition (PVD) Processing (Second Edition)*. 2010, William Andrew Publishing: Boston. p. 237-286.

- 77. Mattox, Film Characterization and Some Basic Film Properties, in Handbook of Physical Vapor Deposition (PVD) Processing. 1998, William Andrew Publishing: Westwood, NJ. p. 569-615.
- 78. Mattox, Vacuum Evaporation and Vacuum Deposition, in Handbook of Physical Vapor Deposition (PVD) Processing. 1998, William Andrew Publishing: Westwood, NJ. p. 288-342.
- 79. Behrisch, W. Eckstein, Sputtering by particle bombardment: experiments and computer calculations from threshold to MeV energies. 2007: Springer.
- 80. Behrisch, K. Wittmaack, Sputtering by particle bombardment III: characteristics of sputtered particles, technical applications. 1991: Springer-Verlag.
- 81. Hattum, A.P., H.R. W. M. Arnoldbik, and F.H.P.M. Habraken, *Distinct processes in radio-frequency reactive magnetron plasma sputter deposition of silicon suboxide films*. Journal of applied physics, 2007(124505): p. 102.
- 82. Licari, L.R. Enlow, *Thin Film Processes*, in *Hybrid Microcircuit Technology Handbook (Second Edition)*, J.L. James and R.E. Leonard, Editors. 1998, William Andrew Publishing: Westwood, NJ. p. 63-103.
- 83. Kodigala, K. Subba Ramaiah, *Structural Properties of I-III-VI2 Absorbers*, in *Thin Films and Nanostructures*. 2010, Academic Press. p. 115-194.
- 84. Benchiheb, M.S. Aida, and N. Attaf, *Plasma optical emission spectroscopy diagnostic during amorphous silicon thin films deposition by Rf sputtering technique*. Materials Science and Engineering: B, 2010. **172**(2): p. 191-195.
- 85. Allen, D.J. Tildesley, *Computer Simulation of Liquids*. 1989: Oxford University Press, USA.
- 86. S. Sakrani, L.Qie.Jie, Yussof Wahab, *The formation of nanoscale clusters nanofilms / quantum dots predicted using a capillary model of nucleation.*Journal of Fundamental Sciences, 2005: p. 23-34.
- 87. Madras, B.J. McCoy, Temperature effects on the transition from nucleation and growth to Ostwald ripening. Chemical Engineering Science, 2004. **59**(13): p. 2753-2765.
- 88. Qin, *Bubble formation in lattice Boltzmann immiscible shear flow*. The Journal of Chemical Physics, 2007. **126**(11): p. 114506-5.

- 89. M. Ali, Growth and study of magnetostrictive FeSiBC thin films for device applications, in Department of Physics & Astronomy. 1999, The University of Sheffield.
- 90. Birkholz, P.F. Fewster, and C. Genzel, *Thin film analysis by X-ray scattering*. 2006: Wiley-VCH.
- 91. Birkholz, *Principles of X-ray Diffraction*, in *Thin Film Analysis by X-Ray Scattering*. 2006, Wiley-VCH Verlag GmbH & Co. KGaA. p. 1-40.
- 92. Cullity, S.R. Stock, *Elements of x-ray diffraction*. 2001: Prentice Hall.
- 93. Samavati, A., F. Aldaw, S.K. Ghoshal, Z. Othaman, and S. Sakrani, *Light emitting germanium and silicon nanoislands grown by RF magnetron sputtering*. Journal of Ovonic Research. **8**(3): p. 65-72.
- 94. Chen, Z. Jiao, M.H. Wu, C.H. Shek, C.M.L. Wu, and J.K.L. Lai, *Microstructural evolution of oxides and semiconductor thin films*. Progress in Materials Science, 2011. **56**(7): p. 901-1029.
- 95. Hasegawa, M. Sakata, T. Inokuma, and Y. Kurata, *Structural change of polycrystalline silicon films with different deposition temperature*. Journal of applied physics, 1999. **85**(7): p. 3844-3849.
- 96. Setyawan, M. Shimada, Y. Imajo, Y. Hayashi, and K. Okuyama, Characterization of particle contamination in process steps during plasma-enhanced chemical vapor deposition operation. Journal of Aerosol Science, 2003. **34**(7): p. 923-936.
- 97. Mattox, 5 Vacuum Evaporation and Vacuum Deposition, in Handbook of Physical Vapor Deposition (PVD) Processing. 1998, William Andrew Publishing: Westwood, NJ. p. 288-342.
- 98. De Los Santos, D. Lee, J. Seo, F.L. Leon, D.A. Bustamante, S. Suzuki, Y. Majima, T. Mitrelias, A. Ionescu, and C.H.W. Barnes, *Crystallization and surface morphology of Au/SiO2 thin films following furnace and flame annealing*. Surface Science, 2009. **603**(19): p. 2978-2985.
- 99. Chen, L. Lai, C.H. Shek, and H.D. Chen, *Production of amorphous tin oxide* thin films and microstructural transformation induced by heat treatment. Applied Physics A: Materials Science & Processing, 2005. **81**(5): p. 1073-1076.

- 100. Sun, H.C. Kang, Synthesis and characterization of self-assembled ZnO nanodots grown on SiNx/Si(001) substrates by radio frequency magnetron sputtering. Thin Solid Films. **518**(22): p. 6522-6525.
- 101. Germanenko, Effect of atmospheric oxidation on the electronic and photoluminescence properties of silicon nanocrystals. Pure Appl. Chem, 2000, Vol. 72, Nos. 1–2, pp. 245–255.
- 102. Chen, H. Chang, A.-T. Chou, T.-M. Hsu, P.-S. Chen, Z. Pei, and L.-S. Lai, Optical properties of stacked Ge/Si quantum dots with different spacer thickness grown by chemical vapor deposition. Applied Surface Science, 2004. **224**(1-4): p. 148-151.
- 103. Gfroerer, *Photoluminescence in Analysis of Surfaces and Interfaces*, in *Encyclopedia of Analytical Chemistry*. 2006, John Wiley & Sons, Ltd.
- 104. Kenyon, P. Trwoga, C.W. Pitt, and G. Rehm, *The origin of photoluminescence from thin films of silicon-rich silica*. Journal of applied physics, 1996. **79**(12): p. 9291-9300.
- 105. Ma, L. Wang, K. Chen, W. Li, L. Zhang, Y. Bao, X. Wang, J. Xu, X. Huang, and D. Feng, *Blue light emission in nc-Si/SiO*₂ *multilayers fabricated using layer by layer plasma oxidation*. Journal of Non-Crystalline Solids, 2002. **299-302**, **Part 1**(0): p. 648-652.
- 106. Chen, Z. Ma, X. Huang, J. Xu, W. Li, Y. Sui, J. Mei, and D. Zhu, Comparison between light emission from Si/SiNX and Si/SiO₂ multilayers: role of interface states. Journal of Non-Crystalline Solids, 2004. **338-340**(0): p. 448-451.
- 107. Zheng, Y.H. Zuo, W. Wang, Y.L. Tao, C.L. Xue, B.W. Cheng, and Q.M. Wang, *Efficient emission and energy transfer in Si/Er-Si-O multilayer structure*. Materials Research Bulletin, 2011. **46**(2): p. 262-265.
- 108. Fafard, Ventra, Massimiliano, Evoy, Stephane Heflin, James, *Quantum-Confined Optoelectronic Systems Introduction to Nanoscale Science and Technology*. 2004, Springer US. p. 443-483.