

**SYNTHESIS, CRYSTALLIZATION KINETIC AND OPTICAL PROPERTIES
OF EUROPIUM DOPED TELLURITE NANO - GLASS**

EZZA SYUHADA BINTI SAZALI

UNIVERSITI TEKNOLOGI MALAYSIA

SYNTHESIS, CRYSTALLIZATION KINETIC AND OPTICAL PROPERTIES OF
EUROPIUM DOPED TELLURITE NANO - GLASS

EZZA SYUHADA BINTI SAZALI

A thesis submitted in fulfilment of the
requirements for the award of the degree of
Master of Science (Physics)

Faculty of Science
Universiti Teknologi Malaysia

AUGUST 2012

This thesis is specially dedicated to:

*My beloved parents,
Sazali Idris, Zaini Zainuddin*

*My supportive siblings,
Addam, Lilli, Dianna, Djohan*

My dedicated lecturers,

My endless spirits

and all my friends.

.....thanks.....

ACKNOWLEDGEMENT

In the name of Allah, the Most Gracious and Most Merciful. All praise to Allah SWT, the Almighty, for His love given me the strength, perseverance, hard work and satisfaction in completing this project.

Special thanks especially to my supervisor, Prof Dr. Md. Rahim Sahar for his guide and criticism throughout the course of this project which taught me the meaning of many efficiency and independence. Thousand appreciations for Assoc. Prof Dr. Sib Krishna Ghoasal, Dr. Ramli, and Dr. Khaidzir for their criticism.

I also would like to express my gratitude to Ministry of Science Technology and Innovation (MOSTI) for supporting scholarship for four semesters Master MyBrain.

Finally, much appreciation and heartfelt for the love and inspiration from my parents, Sazali Idris and Zaini Zainuddin and my siblings, Addam, Lilli, Dianna and Djohan. This thesis proved my efforts this whole time and May it is the best gift for all.

ABSTRACT

Tellurite glass of composition $(80-x) \text{TeO}_2 - 5 \text{Na}_2\text{O} - 15 \text{MgO} - (x) \text{Eu}_2\text{O}_3$, with the concentration region of $0 \leq x \leq 2.5$ mol% was prepared using conventional melt-quenching technique. The transition temperature, T_g , crystallization temperature, T_c and melting temperature, T_m were determined using Differential Thermal Analysis (DTA). The glass with nano-crystalline particle was prepared by heating the as-cast glass at temperature 15 - 20 °C above the T_c . The X-Ray Diffraction (XRD) technique with the Scherrer equation was used to determine the size of nano-crystalline particle of the samples while the Scanning Electron Microscopy (SEM) technique was used to identify the formation of nano-crystalline phase. The crystallization kinetics was investigated by using DTA at various heating rate while the glass density and hardness was determined by Precisa Densitometer and Vickers microhardness, respectively. Meanwhile, the emission characteristic was evaluated using the Photoluminescence Spectroscopy (PL). The thermal analysis showed that the T_g and T_c increased with the increasing of Eu^{3+} content while T_m showed a break in linearity as the Eu^{3+} increased. The result also showed that the glass stability up to 109.29. XRD spectra confirmed the presence of nano particles with an average diameter around 68.7 nm. Meanwhile the SEM studies revealed the existence nano-crystalline morphology which was associated with the existence of crystallized phase. The influence of Eu^{3+} content on crystallization kinetics showed a shift of crystallization temperature peak towards a higher temperature with the increasing of heating rate as described by Kissinger and Ozawa method. The activation energy (E_a) was found to decrease from 306.9 eV to 48.9 eV with an increasing of dopant concentration. The density was found to be in the range of 5.234 to 5.334 g cm⁻³ while the Vickers microhardness was found to vary from 2.59 to 2.84 GPa depending on the dopant concentration. A detailed study on the luminescence spectra showed that all emission peaks for $^5\text{D}_0 \rightarrow ^7\text{F}_0$, $^5\text{D}_0 \rightarrow ^7\text{F}_1$, $^5\text{D}_0 \rightarrow ^7\text{F}_2$, $^5\text{D}_0 \rightarrow ^7\text{F}_3$ and $^5\text{D}_0 \rightarrow ^7\text{F}_4$ transitions were found to be around 568 nm, 600 nm, 628 nm, 664 nm and 712 nm, respectively. It was also found that the heat-treated glass and the inverse quality factor decreased with increasing Eu^{3+} dopant concentration.

ABSTRAK

Kaca tellurite dengan komposisi $(80-x) \text{ TeO}_2 - 5 \text{ Na}_2\text{O} - 15 \text{ MgO} - (x) \text{ Eu}_2\text{O}_3$, mempunyai kepekatan $0 \leq x \leq 2.5$ mol% telah disediakan menggunakan teknik pelindapan leburan konvensional. Suhu peralihan, T_g , suhu penghabluran, T_c dan suhu lebur, T_m ditentukan menggunakan analisis kebezaan terma (DTA). Kaca dengan hablur nano diperolehi dengan memanaskan kaca pada suhu $15 - 20$ °C di atas suhu penghabluran, T_c . Teknik pembelauan sinar-X (XRD) dan persamaan Scherrer digunakan bagi menentukan saiz zarah nano hablur manakala teknik pengimbas mikroskopi elektron (SEM) digunakan bagi mengenal pasti pembentukan fasa nano hablur. Kinetik penghabluran dikaji dengan menggunakan DTA pada kadar pemanasan yang berbeza manakala ketumpatan dan kekerasan kaca masing-masing ditentukan dengan menggunakan Precisa Densitometer dan kekerasan mikro Vickers. Sementara itu, pancaran dinilai dengan menggunakan spektroskopi kefotopendarcayaan (PL). Untuk analisis terma, T_g dan T_c meningkat dengan peningkatan Eu^{3+} sementara T_m tidak menunjukkan peningkatan linear dengan penambahan kepekatan Eu^{3+} . Keputusan juga menunjukkan bahawa julat kestabilan kaca sehingga 109.29 °C boleh dicapai. Spektrum XRD membuktikan kewujudan zarah hablur nano dengan saiz purata diameter 68.7 nm. Sementara itu, kajian SEM membuktikan morfologi hablur nano yang boleh dikaitkan dengan kewujudan fasa hablur. Pengaruh kandungan Eu^{3+} pada kinetik penghabluran menunjukkan peralihan suhu puncak penghabluran ke arah yang lebih tinggi dengan peningkatan kadar pemanasan seperti diterangkan menggunakan kaedah Kissinger dan Ozawa. Tenaga pengaktifan (E_a) didapati menurun daripada 306.9 eV hingga 48.9 eV dengan peningkatan kepekatan pendopan. Ketumpatan kaca didapati berada dalam julat 5.234 hingga 5.334 g cm⁻³ manakala kekerasan mikro Vickers dari 2.59 hingga 2.84 GPa bertambah mengikut kepekatan pendop. Kajian yang teliti ke atas spektrum pendarcayaan menunjukkan bahawa semua pancaran bagi transisi $^5\text{D}_0 \rightarrow ^7\text{F}_0$, $^5\text{D}_0 \rightarrow ^7\text{F}_1$, $^5\text{D}_0 \rightarrow ^7\text{F}_2$, $^5\text{D}_0 \rightarrow ^7\text{F}_3$ dan $^5\text{D}_0 \rightarrow ^7\text{F}_4$ masing-masing didapati berada di sekitar 568 nm, 600 nm, 628 nm, 664 nm dan 712 nm. Didapati juga bagi kaca yang dirawat haba, faktor kualiti songsang menurun dengan pertambahan kepekatan Eu^{3+} .

TABLE OF CONTENTS

CHAPTER	TITLE	PAGE
	DECLARATION	ii
	DEDICATION	iii
	ACKNOWLEDGEMENT	iv
	ABSTRACT	v
	ABSTRAK	vi
	TABLE OF CONTENT	vii
	LIST OF TABLES	x
	LIST OF FIGURES	xii
	LIST OF ABBREVIATIONS	xvi
	LIST OF SYMBOLS	xvii
1	INTRODUCTION	
	1.1 Background of the Study	1
	1.2 Statement of the Problem	4
	1.3 Objective of the Study	5
	1.4 Scope of the Study	5
	1.5 Significance of the Study	6
	1.6 Thesis Plan	7

2 LITERATURE REVIEW

2.1	Introduction	8
2.2	Glass Formation	8
2.2.1	Non-Crystalline Structure	9
2.2.2	The Transformation Range	11
2.3	Glass Structure	12
2.4	Research Survey	12
2.5	Synthesis of Nanocrystallized Glass	18
2.5.1	Heat Treatment Process	19
2.5.2	Nucleation	20
2.5.3	Crystal Growth	20
2.6	X-Ray Diffraction	22
2.7	Thermal Analysis	25
2.8	Calculation of Activation Energy	26
2.9	Scanning Electron Microscopy	28
2.10	Density and Molar Volume	29
2.11	Photoluminescence	30

3 RESEARCH METHODOLOGY

3.1	Introduction	31
3.2	Sample Preparation	31
3.3	Thermal Analysis	35
3.4	Nano Glass Preparation	35
3.4.1	Heat Treatment	35
3.5	Identification of Nano Glass	36
3.5.1	X-Ray Diffraction	36
3.5.2	Scanning Electron Microscopy	37
3.6	Physical Analysis	38
3.6.1	Density measurement	38
3.6.2	Vickers Microhardness Testing	38
3.7	Photoluminescence Spectroscopy	39

4	RESULTS AND DISCUSSION	
4.1	Introduction	40
4.2	Glass Preparation	40
4.3	X-Ray Diffraction	42
4.4	Thermal Parameters	43
4.5	Crystallization Properties	48
	4.5.1 X-Ray Diffraction	49
	4.5.2 Scanning Electron Microscopy	50
4.6	Crystallization Kinetics	51
	4.6.1 Kissinger and Ozawa Method	54
	4.6.2 Activation Energy	57
	4.6.3 Avrami Parameter	59
4.7	Physical Properties	62
	4.7.1 Density Measurement	62
	4.7.2 Vickers Microhardness Testing	66
4.8	Luminescence Spectroscopy	68
5	CONCLUSION	76
	REFERENCES	79
	Appendices	91

LIST OF TABLES

TABLE NO.	TITLE	PAGE
2.1	Distance between components in structure of α -TeO ₂	15
3.1	The nominal composition of the (80 - x) TeO ₂ - 5 Na ₂ O - 15 MgO - (x) Eu ₂ O ₃ glass system	32
4.1	Nominal composition of the as-prepared glass	43
4.2	DTA characteristics for Eu ³⁺ doped tellurite glass as function of dopant concentration	47
4.3	Crystallization peak temperature for Eu ³⁺ doped TeO ₂ - Na ₂ O - MgO glass system with heat treatment process	48
4.4	Crystallization peak temperature for Eu ³⁺ doped TeO ₂ - Na ₂ O - MgO glass system for heat treatment process	52
4.5	Activation energy for TeO ₂ - Na ₂ O - MgO doped Eu ³⁺ glass System	57
4.6	Avrami exponent for TeO ₂ - Na ₂ O - MgO doped Eu ³⁺ glass system	60

4.7	Avrami exponent interpretation through the mechanism of Transformation	61
4.8	Density and molar volume of Eu^{3+} doped $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ glass system	63
4.9	Density of Eu^{3+} doped $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ heat-treated glass System	64
4.10	Molar volume of Eu^{3+} doped $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ heat-treated glass system	65
4.11	The results of microhardness for Eu^{3+} doped $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ glass system	67
4.12	The peak wavelength, λ_p with corresponding transitions for 1mol% Eu_2O_3	70
4.13	The color region interpretation through the wavelength of emission	72

LIST OF FIGURES

FIGURE NO.	TITLE	PAGE
2.1	Crystal structure of TeO_2	10
2.2	Schematic diagram for $\alpha\text{-TeO}_2$ structure of TeO_2 unit	12
2.3	Europium atomic number from the periodic table	13
2.4	Schematic diagram of a hypothetical two-dimensional A_2O_3 oxide with AO_3 triangles as structural units structure in both crystalline and glassy forms	17
2.5	Model of the relationship between specific volume and Temperature	18
2.6	Illustration of the tellurium atom coordination by oxygen	20
2.7	Dependence of growth rate on temperature	23
2.8	Bragg diffraction from a cubic crystal lattice	24
2.9 (a)	Spectrum for crystalline phase	25
2.9 (b)	Diffuse spectrum for amorphous phase	25

2.10	Full width half maximum for diffraction peak of [111] plane	26
2.11	A typical DTA curve for glass	27
2.12	Schematic illustration of SEM operation	30
3.1	Schematic diagram for glass preparation process	35
3.2	Glass preparation flow chart	36
3.3	X-Ray Diffractometer	38
3.4	Nanosecond Luminescence Spectroscopy System, Ekspla Model NT340/1 tunabled Nd: YAG	44
4.1	Glass sample for 79 TeO ₂ - 5Na ₂ O - 15MgO – 1 Eu ₂ O ₃ glass composition	43
4.2	X-ray diffraction patterns for the (80-x) TeO ₂ - 5Na ₂ O - 15MgO - (x) Eu ₂ O ₃ glass system	44
4.3	DTA patterns for TeO ₂ - Na ₂ O - MgO doped Eu ³⁺ glass system	46
4.4	The dependencies of T _g , T _c and T _m with the Eu ₂ O ₃ dopant concentration	48
4.5	The thermal stability with the Eu ₂ O ₃ dopant	49
4.6	The glass forming tendency towards different Eu ₂ O ₃ dopant concentration	49
4.7	XRD spectra at room temperature for 78.5 TeO ₂ – 5 Na ₂ O – 15 MgO – 1.5 Eu ₂ O ₃ glass after heat-treatment at 442.7°C. The inset shows the XRD patterns for the same glass in the	49

as-cast condition

- | | | |
|----------|--|----|
| 4.8 | SEM pictures obtained for heat-treated Eu_2O_3 (x) doped glass system above T_c with (a) $x = 0$, (b) $x = 0.5$, (c) $x = 1.0$ and (d) $x = 2.0$ | 50 |
| 4.9 | DTA curves with different Eu_2O_3 dopant concentration, x as function of different heating rates | 52 |
| 4.10 | Crystallization temperature against the dopant concentration as function of different heating rate | 53 |
| 4.11 | The Kissinger plot (a) and Ozawa plot (b) for Sample 1 | 55 |
| 4.12 (a) | The Kissinger plot pertaining to the crystallization temperature, T_p as function of dopant concentration | 57 |
| 4.12 (b) | The Ozawa plot pertaining to the crystallization temperature, T_p as function of dopant concentration | 58 |
| 4.13 | The activation energy of both Kissinger and Ozawa plot | 61 |
| 4.14 | Avrami exponent with different Eu_2O_3 dopant concentration | 60 |
| 4.15 | The dependence of density with Eu_2O_3 concentration for $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ glass system | 63 |
| 4.16 | The dependence of molar volume against Eu_2O_3 concentration for $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ glass system | 64 |
| 4.17 | A plot of hardness against Eu_2O_3 content for $\text{TeO}_2 - \text{Na}_2\text{O} - \text{MgO}$ for both unheat-treated and heat-treated glass system | 67 |

4.18	A luminescence spectra of Eu^{3+} doped TeO_2 - Na_2O - MgO for unheat-treated glass system excited at 400 nm.	68
4.19	A luminescence spectra of Eu^{3+} doped TeO_2 - Na_2O - MgO for heat-treated glass system excited at 400 nm.	69
4.20	An emission spectra of glass after doped with 1 mol% of Eu_2O_3 which is corresponding to different between untreated and heat-treated glass.	70
4.21	Peak wavelength as function of concentration Eu_2O_3	71
4.22	Peak wavelength as function as function of transition	72
4.23	Peak intensity as function as function of different transitions	73
4.24	FWHM as function as function of transition	74
4.25	Q^{-1} versus transition	75

LIST OF ABBREVIATIONS

bp	-	Bipyramid
BO	-	Bridging oxygen
DTA	-	Differential Thermal Analysis
FWHM	-	Full width half maximum
g	-	gram
IR	-	Infrared
NBO	-	Non-bridging oxygen
NIR	-	Near infrared
nm	-	nanometer
PL	-	Photoluminescence
SEM	-	Scanning electron microscopy
tbp	-	Trigonal bipyramid
XRD	-	X-ray diffraction

LIST OF SYMBOLS

α	-	Crystallization fraction
b	-	Heating rate
d	-	Atomic spacing
β	-	Full width half maximum
d	-	Sample thickness
E_a	-	Activation energy
ΔE	-	Free energy difference per unit volume
ΔJ	-	Electric dipole
k	-	Planck's constant
K_g	-	Glass forming tendency
m	-	Mass
M	-	Molecular weight
n	-	Avrami parameter
ρ	-	Density
ρ_a	-	Air density
ρ_l	-	Liquid density
Q	-	Quality factor
Q^{-1}	-	Inverse quality factor
R	-	Universal gas constant
S	-	Thermal stability
ΔS	-	Spin forbidden emission band
t	-	Time
T	-	Temperature
T_c	-	Crystallization temperature
T_d	-	Developed temperature

T_g	-	Glass transition temperature
T_m	-	Melting temperature
T_n	-	Nucleated temperature
U	-	Growth rate
V_m	-	Molar volume
W_a	-	Weight of sample in air
W_l	-	Weight of sample in immersion fluid
ν_o	-	Frequency
θ	-	Angle
λ	-	Wavelength
λ_{exci}	-	Excitation wavelength

CHAPTER 1

INTRODUCTION

1.1 Background of the Study

Every definition of glass defines it as an amorphous solid. Meanwhile, as a non-crystalline solid, glass can be defined as inorganic material that produced by melting process then annealed it to solid state or to a rigid condition without crystallization process (Rawson, 1980). Glasses can also be described as supercooled liquid or as a solid and therefore have a finite viscosity of ambient temperatures (Zachariasen, 1932). In addition, as a rigid material, glass does not flow when it subjected to moderate forces (Doremus, 1973). The difference between glass and crystal is that it possesses lack of long-range order therefore random in the arrangement of its molecular structure while crystal is otherwise. According to Bengisu (2001), glass are built up of random clustered of ordered atoms which contributes to some aspects of a crystalline solid but its random atomic structure and short range orders is actually a characteristic of a liquid.

The viscosity of glass is in the range of 10^{14} Poise (P) and always presented the short-range order when imposing the X-Ray beam (Sahar, 1998; Santos *et.al*, 2001). It flows at an incredibly slow rate. It has been said that glass could only flow if some considerations are ignored as it forms without any continuous where a

viscosity of the material is increased (Clare *et.al*, 1991; Holloway, 1973). Some of them are of commercially useful compositions and in fact is a rigid solid at ordinary temperatures (Uhlman *et.al*, 1980). Since then, glasses are technologically important as they arise much interest. If compared with other materials, glass exhibit various behaviour as they have their own distinguishable properties. They have no reaction or difficult to be influenced with atmospheric agent and environment. Besides, according to Sahar (1998), glass is not easily corroded. Some workers emphasized glass stability base on the crystallization rate during cooling or reheating (Gabrielson, 1957).

As glass are widely been used in many high technology devices, the development of glass properties is an important research area. The uses of tellurite glasses as a host material have incredibly increases as their applicable potentials in laser and fiber applications (Tanabe *et.al*, 1990). They have low melting temperature, low glass transition temperature and high thermal stability (Tanabe *et.al*, 1990). Furthermore, tellurite shows good transmission in the visible and the near – IR up to 4.5 μm which gave high contribution of optical devices (Burger *et.al*, 1992; Kowada, 1996; Sabadel *et.al*, 1997; Ravi *et.al*, 2001). Besides, the understanding of their microscopic mechanism gave much technical applications and fundamental interests for both academic and the industry (El-Mallawany, 2002). Thus, they have become increasingly important as a new optical functional material because of the low phonon energy and fast response time (Hao Liu *et.al*, 2006). The absence of hygroscopic properties makes tellurite glass is more applicable than phosphate and borate glasses (Sidek *et.al*, 2006). Hence, their corrosion resistance in various environments is of considerable technological importance (Singh *et.al*, 2007; Linda Lee *et.al*, 2007). Modifier such as Na_2O or Li_2O is necessary to add to TeO_2 because the melted material quickly recrystallized. Previous researcher reported that the modifying elements such as Mg, Zn, and Ba were chosen for their ability to aid in the formation of stable tellurite glasses (Nishida *et. al*, 1990). The concentration of about 10 mol% of sodium is necessary to increase the glass forming stability of the mixtures was proven by McLaughlin in 2000.

As tellurite glass are resistant to atmospheric moisture, this allows the incorporations of large concentration of rare-earth ions into the matrix (Nageno *et.al*, 1993). The rare earth doping into tellurium based glass has great implications on the property of the glass. Interest in tellurite glass containing rare earth element are expected for nonlinear optical devices as for their large third-order nonlinear optical susceptibility (Wang, 1994). Moreover, the assimilation of rare earth ions can stabilize the metastable crystalline phase which leads to a development of optical devices (Matthew, *et.al*, 2002; K. Hirano *et.al*, 2002).). Among all glasses, the Eu^{3+} doped tellurite glass has arrised great attention as they can perform persistent spectral hole burning in the ${}^7\text{F}_0 \rightarrow {}^5\text{D}_0$ transition (P. Giridhar *et.al*, 2000). The Eu^{3+} ion is also used as a probe for finding the local structure around the rare earth ion in a crystal or a glass due to relative simplicity of its energy level structure with non-degenerate ground ${}^7\text{F}_0$ and emitting ${}^5\text{D}_0$ states (P. Babu *et.al*, 2000). Hence, their synthesis and dynamic is of considerable technological importance especially for domestic applications (Singh *et.al*, 2007). The integration of rare-earth ions into some kinds of glasses elucidated the significance of rare earth ion-glass host interaction for engineering waveguide devices (Animesh *et.al*, 2007). All of these considerations comprise a powerful control feature for the fabrication of rare-earth doped tellurite glass.

Interestingly, these properties of tellurite glasses are easily influenced by structural effects and kinetics of crystallization. Understanding the crystallization behavior is very important in order to determine the ideal conditions to fabricate glasses for optical applications and for manufacturing glass ceramics (Animesh *et.al*, 2007). This contributes to determination of the stability of glass where the formation of crystal must be avoided (Ray and Delbert, 1990). By using the non-isothermal method, the physical parameters of temperature is studied in detail. In most cases the sample is heated at a fix rate rather than by using isothermal method (Mehta *et.al*, 2004). Moreover, the application of isothermal process is rarely used and usually proceeds under non-isothermal conditions (Zhang *et.al*, 1990). Thus, the determination of thermal characteristics is of current attention to this study. Reheating the glass produces a characteristic DTA curve (Pye *et.al*, 1972). The simple and yet useful method of using DTA for thermal analysis techniques may

contribute to obtain the information about the nucleation and the crystallization mechanisms in such glass system (Elliot, 1983). This systematic study allows this study to understand the mechanism of crystallization kinetics in tellurite glass.

1.2 Statement of Problem

The synthesis of nano-crystalline Eu^{3+} doped TeO_2 based glass has been less extensive been reported and the growth dynamic of the glass system is yet unclear (Ravi *et.al*, 2001). The rare-earth doping into nano-crystalline glass has great implications over the properties of glass which expected to be used in nonlinear optical devices because of their large third-order nonlinear optical susceptibility (Holand *et.al*, 2002). Hence, their synthesis and dynamic understanding is of considerable technological importance (Hirano, *et.al*, 2002). However, According to the previous reported phase diagrams, the tellurite glasses can easily be obtained with a high concentration of europium (Watanabe *et.al*, 2000). Moreover, properties of Eu^{3+} doped TeO_2 glasses which undergo heat treatment process are studied only recently. Therefore, the aims of this study are to prepare the tellurite glass doped with europium (Eu^{3+}) via melt quenching technique. The nano-crystalline glass will be prepared by the heat-treatment process. The effect of dopant throughout all samples towards the physical, thermal, structural and optical properties will be investigated. This study will focus on the behaviors of crystallization kinetics of Eu^{3+} doped TeO_2 - Na_2O - MgO . The activation energy, E_a of crystal growth is deduced through Kissinger and Ozawa analysis under the non-isothermal condition. The kinetic study on crystallization of glass is further complemented by using Avrami exponent method. In addition, the optical properties are investigated using spectroscopy photoluminescence.

1.3 Objectives of the Study

The objectives of this study are:

- i. To prepare the glass and heat-treated nano-glass based $(80-x) \text{ TeO}_2 - 5 \text{ Na}_2\text{O} - 15 \text{ MgO} - (x) \text{ Eu}_2\text{O}_3$.
- ii. To determine the thermal properties of Eu^{3+} doped tellurite glass using Differential Thermal Analysis (DTA).
- iii. To calculate the crystallizations kinetic of Eu^{3+} doped tellurite glass using DTA.
- iv. To identify the crystallization phases and occurrence of nano particles in Eu^{3+} doped tellurite glass using X-Ray Diffraction (XRD) and Scanning Electron Microscopy (SEM).
- v. To study the physical properties of Eu^{3+} doped tellurite glass in terms of density and hardness.
- vi. To investigate the optical properties of Eu^{3+} doped tellurite glass in terms of Photoluminescence (PL).

1.4 Scope of the Study

This study has been divided into several scopes as follows:

- i. Preparation of tellurite glass and heat-treated tellurite glass of composition $(80-x) \text{ TeO}_2 - 5\text{Na}_2\text{O} - 15\text{MgO} - (x) \text{ Eu}_2\text{O}_3$ system which $(0.0 \leq x \leq 2.5 \text{ mol } \%)$.
- ii. Determination of thermal properties and calculation of the crystallization kinetics by implemented Kissinger, Ozawa and Avrami method using DTA.
- iii. Identification of the crystalline nature of the occurrence of nano-particles in glass matrix using XRD and SEM.
- iv. Determination of physical properties from density and hardness.

- v. Investigation of the luminescence spectra of Eu^{3+} doped tellurite glass using Photoluminescence spectroscopy.

1.5 Significance of Study

Among several compositions of glasses which were tested in this research, the glass which had the highest stable thermally phase will be determined. This result will imply that the crystallographic stability is enhanced in rare earth doped samples which is expected to give improvement in poor mechanical properties of tellurite glasses (Sidek *et.al*, 2006; Linda Lee *et.al*, 2007). Thus, this research is done to create more understanding on the nano-glass and to develop nano-structured amorphous in parallel with the development of nano technology. A clearer understanding of growth dynamic of the glass system has be employed. This is very important in order to understand the synthesis and crystallization behavior of the nanocrystalline glass system. Of particular interest is the Eu^{3+} luminescence in the yellow–red region that is frequently exploited for polychromatic displays. This luminescence, may be influenced by presence of nano-crystallized particle in the sample and shows the luminescence enhancement.

1.6 Thesis Plan

The thesis describes the preparation and characterization of europium doped tellurite based nano glasses prepared by melt-quenching technique. This thesis has been divided to five main chapters. Chapter 1 outlines the introduction part which are including general background of study and some reviews from previous work and describes the problem statements which led to this research, objectives, scope and the significance of study. Chapter 2 will describe all the literature reviews, current knowledge and related theories regarding the glass especially for tellurite based glass. The theories describes including the physical properties which are the density and hardness, thermal properties, structural properties, crystallization kinetics and optical properties in terms of absorption spectra and luminescence spectra. Next, Chapter 3 which is research methodology will explain all the experimental procedure including the sample preparation and the experimental techniques. In Chapter 4, the experimental results and discussion throughout this research will be explained in detail. Finally, Chapter 5 summarize all the experimental findings and conclusions.

REFERENCES

- Adorno, A. T., Silva, R. A. G. and Neves, T. B. (2006). Ag Precipitation and Dissolution Reactions in the C-3 Wt% Al-4wt. % Ag Alloy. *Materials Science and Engineering*. **441**, 259-265.
- Aldon, L., Mathurin Leh Deli, Lippens, P. E., Oivier Fourcade, J. and Jomas, J. C. (2010). Thermal Stability of Some Glassy Compositions of the Ge-As-Te Ternary. *Chalcogenide Lett.*, **7**, 187-196.
- Aly, K. A., Othman, A. A. and Abousehly, A. M. (2009). Effect of Te Additions on the Glass Transition and Crystallization Kinetics of $(\text{Sb}_{15}\text{As}_{30}\text{Se}_{55})_{100-x}\text{Te}_x$ Amorphous Solids. *Journal of Alloys and Compounds*, **467**, 417-423.
- Angell, C. A. and Sichina, W. (2006). Thermodynamics of the Glass Transition. *Empirical Aspects*, 53–67.
- Animesh Jha, Shaoxiong Shen, Li Hui Huang, Billy Richards and Joris Lousteau (2007). Rare-Earth Doped Glass Waveguides for Visible, Near-IR and Mid-IR Lasers and Amplifiers. *J. Mater Sci: Mater Electron*. **18**, 315–320
- Artem Malakho, Marc Dussauze and Evelyn Fargin (2005). Crystallization and Second Harmonic Generation in Thermally Poled Niobium Borophosphate Glasses. *J.Solid-State Chem*. **178**, 1888-1857.
- Askeland, D. R. (1990). *The Science and Engineering of Materials*. Chapman and Hall, London.

- Babu, P. and Jayasankar, C. K. (2000). Optical Spectroscopy of Eu^{3+} Ions in Lithium Borate and Lithium Fluoroborate Glasses, *Physica B* **279** 262-281.
- Bagnall, K. W. (1966). *The Chemistry of Selenium, Tellurium and Polonium*. M Elsevier, London pp 59-60.
- Bengisu, M. (2001). *Engineering Ceramics*. Springer-Verlag Berlin Heidelberg, New York pp 9.
- Berzelius, J. T. (1834). *Ann. Phys. Chem.*, **169**, 577.
- Beyer, V. H. (1967). Refinement of the Crystal Structure of Tellurite, the Rhombic TeO_2 . *Journal of Crystallography*. **124**, 228-237.
- Budi Astuti (2005). *Study of Optical Properties of $\text{P}_2\text{O}_5 - \text{Sm}_2\text{O}_3 - \text{MnO}_2$ Glass System*, M.Sc. Thesis, Universiti Teknologi Malaysia.
- Burcu Oz, (2006). *Thermal Microstructural and Optical Characterization of TeO_2 - K_2O Glasses*. M.Sc. Thesis, Istanbul Technical University.
- Burger, H., Kneipp, K., Hobert, H. and Vogel, W. (1992). Glass Formation, Properties and Structure of Glasses in the TeO_2 - ZnO System. *J. Non-Cryst. Solid*, Vol **151**, Issue 1-2, Pages 134-142.
- Callister, W. D. Jr. (1985). *Materials Science and Engineering: An Introduction*. John Wiley and Sons, Inc, New York.
- Charton, P., Thomas, P. and Armand, P. (2003). Raman and Crystallization Behaviors of TeO_2 Sb_2O_4 Glasses. *J.Non-Cryst.Solid*. Vol **321**, 81-88.
- Chen, D., Liu, Y. H., Zhang, Q. Y., Deng, Z. D. and Jiang Z. H., (2005). *Thermal Stability and Spectroscopic Properties of Er^{3+} Doped Niobium Tellurite Glasses for Broadband Amplifiers*. **90**, 78-82.

- Clare, C. Y. and Coppersmith, S. N. (1991). Gravity-Induced Flow of a Structural Glass at Zero Temperature. *J. Non-Cryst.Solids*. **131-133**, 476-478.
- Cooper, A. R. (1982). W.H. Zachariasen - The Melody Lingers On. *J. Non-Cryst. Solids*, **49**, 1-17.
- Deshazer, L. G. and Dieke, G. H. (1963). Spectra and Energy Levels of Eu^{3+} in LaCl_3 . *J. Chem. Phys.*, **38**, 2190
- Doremus R. H. (1973). *Glass Science*. John Wiley and Sons, New York.
- El Mousy, E. K. and El Dem, N. B. (1987). International Atomic Energy Agency and United Nation Educational Scientific and Cultural Organization 6271.
- Elliot, S. R. (1983). *Physics of Amorphous Materials*, London: Longman Group Ltd.
- El-Mallawany, R. A. H., Adly H. El-Sayed and Abd El-Gawad, M. M. H. (1995). ESR and electrical conductivity studies of $(\text{TeO}_2)_{0.95}(\text{CeO}_2)_{0.05}$ semiconducting glasses. *Materials Chemistry and Physics*.**41**, 87 - 9.
- El-Mallawany, R. A. H. (2002). *Telurite Glass Handbook: Physical Properties and Data*. CRC Press, Boca Raton, Florida.
- Gabrielson, O. (1957). Magnussonite, A New Arsenite Mineral from the Langban Mine in Sweden. *Arkiv Mineral, Geol.*, **2**, 133-135.
- Giridhar, P., Sailaja, S., Bhushana Reddy, M., Vemasevana Raju, K., Nageswara Raju, C. and Sudhakar Reddy, B. (2011). Spectroscopic Studies of RE^{3+} (RE =Eu, Tb, Sm & Dy): Lithium Lead Boro Tellurite Glasses, *Ferroelectrics Letters*,**38**:1–10.
- Hao Liu, Qiming Liu and Xiujian Zhao (2006). Crystal Growth and Optical Properties of Cds-Doped Lead Silicate Glass. *Materials characterization*, **58** 96-100.

- Harold N. Ritland (1954). Density Phenomena in the Transformation Range of a Borosilicate Crown Glass. *J. of Am. Ceram. Soc.*, **37**, 370-377.
- Helmut Mehrer (2007). *Diffusion in Solids*.
- Heo, J. (1992). Spectroscopic Analysis of the Structure and Properties of Alkali Tellurite Glasses. *J. Am. Ceram. Soc.*, **75**, 277-81.
- Hilling, W. B. and Turnbull, D. (1956). Theory of Crystal Growth in Undercooled Pure Liquids. *J. Chem. Phys*, **24** 914.
- Hirano, K., Benino, Y., and Komatsu, T. (2001). Rare Earth Doping Into Optical Nonlinear Nanocrystalline Phase in Transparent TeO₂-Based Glass-Ceramics. *J.Phys. Chem.of Solids* **62**,2075-2082.
- Holand, W. and Beall, G. H. (2002). *Control of Nucleation in Glass Ceramic*. Glass Ceramic Technology, Wiley-Blackwell, Berlin.
- Holland, L. (1964). *The Properties of Glass Surface*. F InstP., Brunel College Acton.
- Holloway, D.G. (1973). *The Physical Properties of Glass*, London & Winchester: Wykeham Publ. (London) Ltd.
- Huber, D. L. (1984). Low-Temperature Optical Dephasing of Rare-Earth Ions in Glass. *Phys. Re. Lett.* **52**, 2281 – 2284.
- Jia Liu, James M. O'Reilly, Thomas W. Smith and Paras N. Prasad (2005). Photo-Patterning Hybrid Sol-Gel Glass Materials Prepared from Ethylene Tellurate And Alkoxysilane. *J.Non-Cryst. Solids*, **351**. 2440-2445.
- Jurn, W., Schmelzer, P., Vladimir, M., Fokin, Alexander, S., and Abyzov (2010). How Do Crystal Form and Growi Glass-Forming Liquids: Ostwald's Rule Of Stages And Beyond. *International J. of App. Glass Sci.* 1 [1] 16-26.

- Kassim, A. (2010). *Physical and Optical Properties of Neodymium and Neodymium Co-Doped Tellurite Glass System*. PhD Thesis, Universiti Teknologi Malaysia.
- Kaur, G., Komatsu, T. and Thangara, R. (2000). *J. Mater. Sci.*, **35**, 903.
- Kowada, Y., Moromoto, K., Adachi H., Tatsumisago, M. and Minami, T. (1996). Electronic States Of Binary Tellurite Glasses. *J. Non-Cryst. Solids*, **196**, 204-209.
- Linda Lee and Norbert D. Greene (2007). *Corrosion Characteristics of the Rare-Earth and Yttrium Metals*. Corrosion Research Laboratory Department of Materials Engineering Rensselaer Polytechnic Institute Troy, New York.
- Luciana, R. P., Kassab, Ricardo de Almeida, Davinson M. da Silva, Thiago A. A. De Assumpção, and Cid B. D Araújo (2009). Enhanced Luminescence Of Tb³⁺/Eu³⁺ Doped Tellurium Oxide Glass Containing Silver Nanostructures. *J. Appl. Phys.* **105**, 103505.
- Luciano, A. Bueno, Petr melnikov, Younes Messaddeq and Sidney, J. L., Ribeiro (1999). Er³⁺ and Eu³⁺ Containing Transparent Glass Ceramics in the system PbGeO₃ – PbF₂ – CdF₂. *J. Non-Cryst. Solids*, **247**, 87-91.
- Mahajan, S. V. (2006). *J. of the Chem. Soc.*, Chemical Communications, Issues 25-28.
- Marjanovic, S., Toulouse, J., Jain, H., Sandman, C., Dieroff, V. and Kortan, A. R. (2003). *J.Non-Cryst.Solid.* **322**, 311-318.
- Matthew, J., Dejneka and Alexander Streltsov (2002). Science and Technology Division, Corning Incorporated, Corning, NY 14831.
- McColm, I. J. (1983). *Ceramic Sciences of Materials Technologists*. Leonard Hill Glaslow, Scotland.

- McLaughlin, C. Tagg, S. L. Zwanziger, J. W. Haeffner, D. R. and Shastri, S. D. (2000). The Structure of Tellurite Glass: A Combined NMR, Neutron Diffraction, and X-Ray Diffraction Study. *J. Non-Cryst. Solids*, **274**, 1–8.
- Mehl, R. F. (1939). *Trans Am. Inst. Min. Metal. Eng.* **135**, 416.
- Mehta, N., Zulfeqar, M. and Kumar, A. (2004). Crystallization Kinetics of Some Se - Te - Ag Chalcogenide Glasses. *J. Opto. Advanced Mat.* **6**, No. 2, 441-448.
- Mekki, V, Khattak, G. D. and Wenger, L. E., (2006). Structural and Magnetic Investigations of Fe_2O_3 - TeO_2 Glasses. *J. Non-Cryst. Solids*, **352**, 3326–333.
- Mirgorodsky, A. P., Merle-Me'jean, T., Champarnaud, J. C., Thomas, P. and Frit, B. (2000). Dynamics and Structure of TeO_2 Polymorphs: Model Treatment of Paratellurite and Tellurite; Raman Scattering Evidence for New G- And D- Phases. *J. Phys. Chem. Solids*, **61** 501–509.
- Mochida, N., Takahashi, K., Nakata, K., Shibusawa, S. and Yogyo-Kyokai-Shi (1978). Tellurite-Based Glass and Structural Properties of Two-Component, Including Monovalent Cations, Divalent. *J. Non-Cryst. Solids*, **86**, 316.
- Murugan, G. S. and Ohishi, Y. (2004). TeO_2 -BaO-SrO-Nb₂O₅ Glasses: A New Glass System for Waveguide Devices Applications. *J. Non-Cryst. Solids*, **341**, 86–92.
- Nageno, Y. Takeke, H. and Morinaga (1993). Correlation between Radiative Transition-Probabilities of Nd^{3+} and Composition in Silicate, Borate, and Phosphate Glasses. *J. of Am. Ceram. Soc.*, **76** (12) 3081-3086.
- Narottam P. Bansal and Eleanor A. Gamble (2005). Crystallization Kinetics of a Solid Oxide Fuel Cell Seal Glass by Differential Thermal Analysis. *J. of Power Sources*. **147**, 107-115.

- Nazabal, V., Todoroki, S., Nukui, A., Matsumoto, T., Suehara, S., Hondo, T., Araki, T., Inoue, S., Rivero, C. and Cardinal, T. (2003). Oxyfluoride Tellurite Glasses Doped by Erbium: Thermal Analysis, Structural Organization and Spectral Properties. *J. Non-Cryst. Solid.* **325**, 85-102.
- Nishida, T., Yamada, M., Ide, H. and Takashima, Y. (1990). "Correlation between the Structure and Glass Transition Temperature of Potassium, Magnesium and Barium Tellurite Glasses," *J. Mater. Sci.*, **25**, 3546–3550.
- Nukui, A., Tanaguchi, T. and Miyata, M. (2002). In Situ High-Temperature X-Ray Observation of Structural Changes of Tellurite Glasses with P-Block Oxides; ZnO–TeO₂ Glasses. *J. Non-Cryst. Solid.* **293-295**, 255-260.
- Omar, M. A. (1975). *Elementary Solid State Physics: Principles and Applications*. Lebanon, Indiana, U.S.A.: Addison-Wesley.
- Ovecoglu, M. L., Ozen, G. B. and Demirata, A. (2001). Microstructural Characterization and Crystallization Kinetics Of (1–X) TeO₂–(X)LiCl (X=0.6–0.4 Mol) Glasses. *J. Euro. Ceram. Soc.* **21**, 177-183.
- Ozawa, T. (1971). Kinetics of Non-Isothermal Crystallization. *Polymer.* **12**, 150-158.
- Paul, A. (1982). *Chemistry of Glasses*, Chapman and Hall, London, New York.
- Ping, D.H., Li, D.X. and Ye, H.Q., (1995). Microstructural Characterization of Nanocrystalline Materials. *J. Mater. Sci.Lett.*, **14**, 1536-1540.
- Praveen, K. Jain, Deepika, K., Rathore, S. and Saxena, N. A. (2008). Phase Transformation and Crystallization Kinetics of Se₉₀In₈Sb₂ Chalcogenide Glass. *Chalcogenide Lett.*, **5**, 126-136.

- Pye, L. D. Stevens, H. J. and W. C. Lacourse (1972). *Introduction to Glass Science*. Proceedings of a Tutorial Symposium Held at the State University of New York, College of Ceramics at Alfred University , Alfred, N.Y., 8-19.
- Rahul Vaish and Varma, K. B. R. (2009). The Glass Transition and Crystallization Kinetic Studies on Borosilicate Glasses. *J. Phys. D. Appl. Phys.*, **42**, 015409, 7pp.
- Rao, T.V.R., Reddy, R.R., Nazeer Ahammed Y., Parandamaiah, Sooraj Hussain, N., Buddhudu, S. and Purandar, K., (2000). Luminescence Properties Of Nd³⁺: TeO₂-B₂O₃-P₂O₅-Li₂O Glass. *J. Infrared Phys. and Tech.* **41**, 247-258.
- Ravi Kumar, V., Bhatnagar, V., Anil, K. and Jagannathan, R. (2001). Structural and Optical Studies of Pr³⁺, Nd³⁺ and Eu³⁺ Ions in Tellurite Based Oxyfluoride, TeO₂-Cif Glasses .
- Rawson H. (1980). *Properties and Applications of Glass*. New York: Elsevier Scientific Publishing.
- Ray, S. C. and Delbert, E. D. (1990). Determining the Nucleation Rate Curve for Lithium Disilicate Glasses by Differential Thermal Analysis. *J.Am. Ceram. Soc.*, **73**, 439-442.
- Ricardo de Almeida, Davinson, M. Da Silva, Luciana, R. P., Kassab and Cid B. de Araujo (2008). Eu³⁺ Luminescence in Tellurite Glasses with Gold Nanostructures. *Optics Communications*. **281** 108–112.
- Ritland, H. N. (2006). Density Phenomena in the Transformation Range of a Borosilicate Crown Glass.
- Sabadel, J. C., Armand, P. D., Cachau-Herreillat, Baldeck, P., Doclot, O., Ibanez, A. and Philippot, E. (1997). Structural and Nonlinear Optical Characterizations of Tellurium Oxide-Based Glasses: TeO₂-BaO-TiO₂. *J.Solid State Chemistry* **132**, 411-419.

- Sahar (1990). *A Study on Oxyhalide Glasses*. PhD Thesis, University of Warwick.
- Sahar M. R., Jehbu A. K and Karim M. M. (1997). TeO₂-ZnO-ZnCl₂ Glasses For IR Transmission. *J.Non-Cryst.Solid*. **213-214**, 164-167.
- Sahar (1998). *Sains Kaca*. Universiti Teknologi Malaysia Skudai.
- Sahar, M. R. (2000). *Fizik Bahan Amorfus*. 1st Ed. UTM Skudai: DBP.
- Sahar, M. R. and Noordin, N. (1995). Oxychloride Glasses Based on the TeO₂-ZnO-ZnCl₂ System. *J.Non-Cryst.Solid*, Vol 184, 137.
- Sahar, M. R. and Sazali, E. S. (2011). *Physical and Thermal Properties of TeO₂-Na₂O-MgO Doped Eu₂O₃ Glass*. *UMTAS*.
- Santos, I. A., Eiras J. A., and Araujo E. B. (2001). A Dta Study of Activation Energy for Crystallization in Low Silica Content Calcium Aluminosilicate Glasses. *J. Mat. Sci. Lett.* **20**, 1815-1817.
- Sekiye, T., Mochida, N. and Soejima, A. (1992). Raman Spectra of MO_{1/2}-TeO₂ (M= Li, Na, Rb, Cs and Ti) Glasses. *J.Non-Cryst.Solids*, **144**, 128-144.
- Shaik Abdul Saleem, Thammisetty Sasikala, Asanapuram Mohan Babu, Lalapet Rama Moorthy, Bungala Chinna Jamalaiah and Mula Jayasimhadri (2011). Erbium-Doped Fluoroborate Glasses for Near Infrared Broadband Amplifiers. *International Journal of Applied Glass Science* 2 [3] 215-221.
- Shelby, J. E. (2005). *Introduction to Glass Science and Technology*. Royal Society of Chemistry.
- Shengyu Jiang, Jiancheng Zhang, Feng Gu, Yue Shen and Hua Wang (2006). Crystallization Kinetics of Nanophase Glass-Ceramics as Magnetic Disk Substrate. *J. Matter Sci. Technol.*, **22** No.2.

- Sherman Hsu C. P. (1997). Handbook of Instrumental Techniques for Analytical Chemistry, Prentice-Hall, NJ, p. 254.
- Shixun Dai, Jialu Wu, Junjie Zhang, Guonian Wang and Zhonghong Jiang (2005). The Spectroscopic Properties of Er^{3+} -Doped TeO_2 -Nb Glasses with High Mechanical Strength Performance. *Spectrochimica Acta Part A*, **62**, 431–437
- Sidek, H. A. A., Halimah, M. K. and Faizal, M. N. (2006). Synthesis and Elastic Behaviour of Borate Glass Doped with High Te Content. *J.App. Sci.* **6** (2), 274-279.
- Singh A.K. and Rai S. B. (2007). *Up conversion and optical thermometry in $\text{HO}^{3+}:\text{TeO}_2$ glass, effect of addition of PbO_2 and BaCO_3* . Laser and Spectroscopy Laboratory, Department of Physics BHU, Varanasi 221005, India.
- Sphoorti Srivastava, Zulfequar, M. and Kumar, A. (2009). Study of Glass Transition Kinetics in Glassy Alloys of Se. *Chalcogenide Letters*, **6**, 403 – 414.
- Stanworth, J. E., (1952). Tellurite Glasses, *J. of Nature*, **169**, 581-82.
- Stehlik, B. and Balak, L. (1948). *Chem. Zvesti*, **2**, 69.
- Strnad, Z. (1986). *Glass Ceramic Materials*, Elsevier, Amsterdam, The Netherlands.
- Sun, K. (1988). *Preparation and Characterization of Rare Earth Glasses*. Thesis, Brown University.
- Surendra Babu, S., Kiwan Jang, Eun Jin Cho, Hoseop Lee and Jayasankar C. K. (2007). Thermal, Structural and Optical Properties of Eu^{3+} -Doped Zinc-Tellurite Glasses. *J. Phys. D: Appl. Phys.* **40** 5767–5774.
- Tagg, S. L., (1997). *Structural Investigation of Alkali Tellurite Crystals and Glasses Utilizing Diffraction and Spectroscopic Techniques*. Ph. D. Thesis, Dept. of Chemistry, Indiana University, Indiana.

- Tanabe, S., Hirao, K. and Soga, N. (1990). *J.Non-Cryst.Solids*, **122**, 79.
- Tanaka, K., Yoko, T., Yamada, H. and Kamiya, K., (1998). Structure and Ionic Conductivity of LiCl-Li₂O-TeO₂ Glasses. *J. Non-Cryst. Solids*, **103**, 250-56.
- Theil, C. W., Sun, Y. and Cone, R. L. (2002). Progress in Relating Rare Earth ion 4f and 5d Energy Levels to Host Bands in Optical Materials for Hole Burning, Quantum Information and Phosphors. *J.of Modern Optics*, **49**, 2399.
- Uhlmann, D. R. and Kreidl, N. J. (1980). *Glass Science and Technology*. New York: Academic Press.
- Wang, J. S., Vogel, E.M. and Snitzer, E. (1994). 1.3 μ Emission of Neodymium and Praseodymium in Tellurite-Based Glasses, *J. Non-Cryst. Solids*, **178** 109-113.
- Watanabe, T., Benino, Y., Ishizaki, K., and Komatsu, T. (2000). *J.Ceram. Soc. Japan*. **107** 1140.
- Weber, M. J. (1990). Science and Technology Of Laser Glass. *J.Non-Cryst.Solids*, **123**, No. 1-3, 208-222.
- Wei, S. J., Zhu, H. F. and Chen, K. (2011). Phase Change Behavior in Titanium-Doped Ge₂Sb₂Te₅ Films. *J. of App. Phys. Lett.*, **98**, 231910.
- Yang Li, Jianhu Yang, Shiqing Xu, Guonian Wang and Lili Hu, (2005). *J. Mat. Sci. Tech*. **21**, No.3, 391-394.
- Zachariasen W. H. (1932). The Atomic Arrangement in Glass. *J. Am. Chem. Soc.*, **54** **10**, 3841–3851.
- Zarifah, N. A., Halimah, M. K., Hashim, M., Azmi, B. Z. and Daud, W. M. (2010). Magnetic Behaviour of (Fe₂O₃)_x (TeO₂)_{1-x} Glass System Due to Iron Oxide. *Chalcogenide Lett.*, 565-571.

Zhang Zhiying and Cao Zhenlin, (1990). Kinetics of Non-Isothermal Crystallization. *Chinese J. of Poly.Sci.* **8** No. 2.

Zishan H. Khan (2011). Non-Isothermal Crystallization in Amorphous $\text{Ga}_x\text{Se}_{100-x}$ Nanorods. *Japanese Journal of Applied Physics.* **50**, 105603.