

CHARACTERISTICS OF NEODYMIUM-MODIFIED BORATE CRYSTALS
AND GLASSES GROWN VIA POLYCRYSTALLINE SEED MEDIATION

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DEDICATION

This thesis is dedicated to my beloved mothers (Raja Maharom & Sh. Zaiton), fathers (Kamaruddin & Sy. Abd. Azam), wife (Sh. Rabi'atul Adawiyyah), kids (Wan Nur Humairah, Wan Uzair Faaris, Wan Umar Hariz and Wan Uthman Daris) and all my families, who always give support and pray for my success.

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ABSTRACT

A series of glass samples with composition $(90-x)\text{Li}_2\text{B}_4\text{O}_7+10\text{Nb}_2\text{O}_5+x\text{Nd}_2\text{O}_3$, where $x = 1, 5, 10, 15$, and 20 mol% were prepared by the melt-quenching technique. For polycrystal seed preparation as well as crystal growth process, 1 mol% of Nd_2O_3 was chosen. Nd_2O_3 was used as a modifier to investigate the impact on the physical, structural, and optical properties of the glasses and crystals. Polycrystal seed as well as platinum wire and lithium niobate single crystal seed were used to grow crystals by Czochralski technique. The growth and growth mechanism of the crystals were studied. The glass samples and crystals were subjected to various characterisation techniques and the physical, structural, and optical properties were determined. XRD pattern confirmed the amorphous nature of the glass system while several peaks corresponded to the phases of LiNbO_3 , LiNb_3O_8 , $\text{Li}_2\text{B}_4\text{O}_7$, and $\text{Nd}(\text{BO}_2)_3$ were recorded for heat-treated glass (polycrystal). The microstructural of the samples (glass \rightarrow polycrystal) showed that these crystallites grew, further suggesting that the nucleation and growth process were responsible for the development of this kind of microstructure. The optimum growth conditions of pulling rates and rotation speeds were $0.2\text{--}1.5\text{ mmh}^{-1}$ and $5\text{--}7$ rpm, respectively. To stabilise the growth rate and achieve a good growth interface, the temperature gradient was gently optimised. FTIR spectra exhibited four vibration bands around 700 , $800\text{--}1200$, $1200\text{--}1600$, and $3200\text{--}3600\text{ cm}^{-1}$ assigned to the bending of B–O–Nd, B–O stretching of tetrahedral $[\text{BO}_4]$ units, stretching relaxation of B–O in $[\text{BO}_3]$ units, and stretching of O–H group in both NdLNB crystal and glass system. The UV–Vis spectra showed ten absorption peaks and the transition at ${}^4\text{G}_{5/2}$ (583 nm) was the most prominent. As the concentration of Nd^{3+} ions for glass system increased, both optical band gap and Urbach energies decreased. Additionally, the optical band gap of NdLNB crystal was significantly lower than that of NdLNB glass which were 2.84 and 3.75 eV , respectively. The Judd–Ofelt intensity parameters of all the glass samples followed the trend of $\Omega_6 > \Omega_4 > \Omega_2$ except for 5NdLNB glass, which followed the trend of $\Omega_4 > \Omega_6 > \Omega_2$ which similar to NdLNB crystal. The radiative lifetime of the glass system increased with increasing Nd^{3+} concentration. Meanwhile, radiative lifetime for NdLNB crystal was slightly higher than for NdLNB glass. Luminescence spectra showed two prominent infrared emissions at 903 and 1059 nm which were due to ${}^4\text{F}_{3/2}\rightarrow{}^4\text{I}_{9/2}$ and ${}^4\text{F}_{3/2}\rightarrow{}^4\text{I}_{11/2}$ transitions, respectively while NdLNB crystal exhibited four emissions at 883 , 943 , 1064 , and 1115 nm with the same transition. SHG tests for both NdLNB crystal and NdLNB glass recorded an emission peak at 532 and 529.5 nm , respectively. The achieved results promise the use of these optimised glasses and crystals in frequency conversion, nonlinear optical application, and laser production. The new approach of using the polycrystal seed to grow a crystal with modified properties from its own glass-forming melt was also successful.

ABSTRAK

Satu siri sampel kaca dengan komposisi $(90-x) \text{Li}_2\text{B}_4\text{O}_7 + 10\text{Nb}_2\text{O}_5 + x\text{Nd}_2\text{O}_3$, dengan $x = 1, 5, 10, 15$ dan 20 mol\% telah disediakan menggunakan kaedah pelindapan leburan. Bagi penyediaan benih polihablur dan juga proses pertumbuhan hablur, $1 \text{ mol\% Nd}_2\text{O}_3$ telah dipilih. Nd_2O_3 telah digunakan sebagai pengubah untuk mengkaji kesan terhadap sifat fizikal, struktur dan optik kaca dan hablur. Benih polihablur serta wayar platinum dan hablur tunggal litium niobate digunakan untuk menumbuhkan hablur dengan kaedah Czochralski. Mekanisma pertumbuhan hablur telah dikaji. Hablur dan kaca NdLNB telah melalui pelbagai kaedah pencirian dan sifat fizikal, struktur dan optik telah diperolehi. Corak XRD telah mengesahkan sifat amorfus sistem kaca manakala beberapa puncak yang merujuk kepada fasa LiNbO_3 , LiNb_3O_8 , $\text{Li}_2\text{B}_4\text{O}_7$ dan $\text{Nd}(\text{BO}_2)_3$ telah direkodkan untuk kaca terawat haba (polihablur). Struktur mikro sampel (kaca \rightarrow kaca terawat haba) menunjukkan habluran telah membesar, yang mana proses nukleasi dan pertumbuhan berperanan dalam pengembangan struktur mikro ini. Tetapan pertumbuhan optimum bagi kadar tarikan ialah $0.2\text{--}1.5 \text{ mmj}^{-1}$ dan kelajuan putaran ialah $5\text{--}7 \text{ rpm}$. Bagi menstabilkan kadar pertumbuhan dan mendapatkan sempadan permukaan yang baik, kecerunan suhu perlu dikawal dengan baik. Spektrum FTIR telah mempamerkan tiga jalur getaran di sekitar $700, 800\text{--}1200, 1200\text{--}1600, \text{ and } 3200\text{--}3600 \text{ cm}^{-1}$ yang merujuk kepada lenturan ikatan B–O–Nd, regangan ikatan B–O unit tetrahedron $[\text{BO}_4]$ dan kelonggaran regangan ikatan B–O unit $[\text{BO}_3]$ untuk kedua-dua NdLNB hablur dan sistem kaca. Spektrum UV-Vis-NIR telah menunjukkan sepuluh puncak penyerapan dan transisi pada ${}^4\text{G}_{5/2}$ (583 nm) adalah yang paling ketara. Apabila kepekatan ion Nd^{3+} untuk sistem kaca meningkat, kedua-dua jurang tenaga optik dan tenaga Urbach menurun. Selain itu, didapati jurang tenaga optik hablur NdLNB jauh lebih rendah daripada kaca NdLNB yang masing-masing ialah 2.84 dan 3.75 eV . Telah diperhatikan bahawa parameter keamatan Judd–Ofelt bagi semua sampel kaca mengikuti trend $\Omega_6 > \Omega_4 > \Omega_2$ kecuali kaca bagi 5NdLNB yang mengikuti trend $\Omega_4 > \Omega_6 > \Omega_2$ sama seperti hablur NdLNB. Jangka hayat pendarcahaya untuk sistem kaca meningkat apabila kepekatan ion Nd^{3+} meningkat. Manakala, jangka hayat pendarcahaya untuk hablur NdLNB adalah sedikit tinggi berbanding kaca NdLNB. Spektrum fotopendarcahaya menunjukkan dua pancaran inframerah yang ketara pada 903 dan 1059 nm yang disebabkan oleh transisi ${}^4\text{F}_{3/2} \rightarrow {}^4\text{I}_{9/2}$ dan ${}^4\text{F}_{3/2} \rightarrow {}^4\text{I}_{11/2}$, manakala hablur NdLNB menunjukkan empat pancaran pada $883, 943, 1064, \text{ dan } 1115 \text{ nm}$ dengan transisi yang sama. Ujian SHG bagi kedua-dua hablur dan kaca NdLNB telah mencatatkan puncak pancaran masing-masing pada 532 dan 529.5 nm , yang mana ianya sesuai digunakan untuk penggandaan frekuensi, aplikasi optik tidak linear dan penghasilan laser. Pendekatan baru untuk menggunakan benih polihablur bagi menumbuhkan hablur dari pembentukan leburan kacanya sendiri telah berjaya.

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LIST OF ABBREVIATIONS

| | | |
|---------|---|--|
| ADC-CGS | - | Automatic diameter control - crystal growth system |
| BBO | - | Beta-barium borate |
| BF | - | Bright field |
| BO | - | Bridging oxygen |
| BSE | - | Back-scattered electrons |
| CB | - | Conduction band |
| CBO | - | Cesium triborate |
| CLBO | - | Cesium lithium borate |
| CR | - | Cross relaxation |
| DF | - | Dark field |
| DP | - | Diffraction pattern |
| DTA | - | Differential thermal analysis |
| EDX | - | Energy dispersive X-ray |
| FTIR | - | Fourier transform infrared |
| FWHM | - | Full width half maximum |
| HT | - | Heat treatment |
| IR | - | Infrared |
| JCPDS | - | Joint Committee on Powder Diffraction Standards |
| J-O | - | Judd-Ofelt |
| KAB | - | Potassium aluminum borate |
| KDP | - | X-Ray diffraction |
| KTP | - | Potassium titanyl phosphate |
| LBO | - | Lithium triborate |
| LCOB | - | Lithium calcium oxoborate |
| LFB | - | Lithium Fluoro Borate |
| Li | - | Lithium |
| LTB | - | Lithium tetraborate |
| Nb | - | Niobium |
| NBO | - | Non-bridging oxygen |
| Nd | - | Neodymium |

| | | |
|------------|---|------------------------------------|
| NdLNB | - | Neodymium lithium niobium borate |
| NLO | - | Nonlinear optical |
| NR | - | Non-radiative |
| O | - | Oxygen |
| PL | - | Photoluminescence |
| PR | - | Pulling rate |
| R | - | Radiative |
| RE | - | Rare earth |
| RECOB | - | Rare earth calcium oxyborate |
| RS | - | Rotation speed |
| SADP | - | Selected area diffraction pattern |
| SAED | - | Selected area electron diffraction |
| SAW | - | Surface acoustic waves |
| SE | - | Secondary electrons |
| SEM | - | Scanning electron microscopy |
| SFD | - | Self-frequency doubling |
| SHG | - | Second harmonic generation |
| TEM | - | Transmission electron microscopy |
| TM | - | Transition metal |
| TSSG | - | Top seeded solution growth |
| UV-Vis -IR | - | Ultraviolet visible infrared |
| VB | - | Valence band |
| XRD | - | X-Ray diffraction |
| YAB | - | Yttrium aluminum borate |
| YAG | - | Yttrium aluminium garnet |
| YCOB | - | Yttrium calcium oxoborate |

LIST OF SYMBOLS

| | | |
|-----------------------|---|-------------------------------------|
| A | - | Absorbance |
| A_{ed} | - | Electric-dipole |
| A_{md} | - | Magnetic-dipole |
| α | - | Absorption coefficient |
| B | - | Constant |
| β | - | Nephelauxetic ratio |
| $\bar{\beta}$ | - | Average nephelauxetic ratio |
| c | - | Speed of light |
| d | - | Thickness of the sample |
| D | - | Diameter |
| E_u | - | Urbach energy |
| E_f | - | Energy of electron of final state |
| E_i | - | Energy of an electron at lower band |
| E_g^D | - | Direct optical energy band gap |
| E_g^I | - | Indirect optical energy band gap |
| $\varepsilon(\nu)$ | - | Molar extinction coefficient |
| e | - | Electron charge |
| F | - | Force constant |
| f_{exp} | - | Experimental oscillator strength |
| f_{cal} | - | Oscillator strength |
| ΔG | - | Gain bandwidth |
| $h\nu$ | - | Photon energy |
| H_R | - | Hruby parameter |
| λ | - | Wavelength |
| $\Delta\lambda_{eff}$ | - | Effective gain bandwidth emission |
| Ω_λ | - | Judd-Ofelt parameters |
| m | - | Electron mass |
| m_r | - | Atomic weights in kg of cation |
| m_o | - | Atomic weights in kg of anion |
| μ | - | Reduced mass |

| | | |
|---------------|---|---|
| n | - | Refractive index |
| N_A | - | Avogadro number |
| OPD | - | Oxygen packing density |
| θ | - | Angle |
| ρ | - | Density |
| ρ_L | - | Density of distilled water |
| P | - | Pulling rate |
| ρ_s | - | Crystal density |
| ρ_l | - | Melt density |
| Q | - | Quality factor |
| rms | - | Root-mean-square |
| R | - | Crucible radius |
| σ_P^E | - | Stimulated emission cross-section |
| S_{ed} | - | Line-strength for electric |
| δ | - | Bonding parameter |
| t | - | Crystallite size |
| T_c | - | Glass crystallization temperature |
| T_g | - | Glass transition temperature |
| T_m | - | Glass melting temperature |
| Γ_M | - | Decay rate |
| Γ_{NR} | - | Non-radiative decay |
| Γ_R | - | Intrinsic radiative decay rate |
| τ_{rad} | - | Radiative lifetime |
| ΔT | - | Glass thermal stability |
| U_κ | - | Values of reduced matrix elements |
| ν | - | Wavenumber |
| ν_g | - | Wavenumber of particular transition to an ion |
| ν_a | - | Wavenumber of same transition for aquo ion |
| V_m | - | Molar volume |
| W_1 | - | Weight of sample in air |
| W_2 | - | Weight in distilled water |
| x_i | - | Mole fraction |

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CHAPTER 1

INTRODUCTION

1.1 Background of the Study

Solid state materials can be classified as crystals, polycrystals (ceramics), or amorphous depending on their arrangement of constituent molecules, atoms, or ions. Several materials can form a crystal with the same composition but in different arrangements of the atoms which create different three-dimensional structures (Myerson, 2002). Non-crystalline solid, often referred to as amorphous solid, is the opposite extreme of a crystal. This type of solids has neither reticular nor granular structure. Amorphous solids possess short-range order but are devoid of any organised structure over longer distances; in this respect they resemble liquids. However, their rigidity and cohesiveness allow them to retain a definite shape, so for most practical purposes, they can be considered solids. In fact, the only obvious distinction between amorphous materials, such as glass, and liquids is the high viscosity (resistance to flow) of the amorphous solids.

The glass composition is extremely important in the formation of glass. Generally, glass is composed of different oxides that serve various functions, particularly in research and industry. The presence of strongly bonded large networks of atoms in the liquid is a basic condition for glass formation. A superior glass must have many bonds or linkages of the types with high bond strengths like B–O–B, Si–O, Ge–O, and P–O as glass formers. Some oxides, such as borate (B_2O_3), silicate (SiO_2), germanate (GeO_2), and phosphate (P_2O_5) are defined as glass formers because they are capable to form glassy network by themselves under normal quenching conditions but act as glass formers when mixed with others, such as ZnO, PbO, MgO, CaO, and BaO (Gautam et al., 2012).

A material can be transformed into a crystal by slow and gradual transformation from solid, liquid/solution, or vapour phase. Several techniques adopting one of these three basic methods can be employed for growing from small to big crystals (Brice, 1986). Selection of appropriate techniques of crystal growth is highly dependent on the chemical and physical properties of the material to be grown and the suitability of the technique to grow the required crystal (Seevakan and Bharanidharan, 2018). This is because each technique to be used has certain parameters to be considered for the success of crystal growth such as growth kinetics, chemical reactivity, temperature gradient, solubility, shape, purity, decomposition, melting point, and the cost involved (Brian, 1980).

Presently, borate glass and crystal are receiving much more attention due to their applications in technologies such as solid-state lasers, nonlinear optics, and solar energy (Alajerami et al., 2012). Borate is one of the popular glass formers because it can easily combine with a variety of oxides to form a binary glass system (Kaur et al., 2014). Borate network building units can be either triangles (BO_3) with non-bridging atoms or tetrahedral (BO_4) with all bridging oxygen atoms, making it an interesting system. In comparison with other conventional glasses, borate glass is most recently analysed due to the superior optical and mechanical properties, chemical durability, stability against atmospheric moisture, good solubility of rare earth ions, low melting temperature, and good corrosion resistance (Wu et al., 2011). Moreover, borate glasses and crystals are suitable hosts for various transition metals and rare earth ions, making them suitable for optical materials.

Several attempts have been made to improve the values of borate glass and crystal. It is possible to change the coordination geometry of boron from BO_3 to BO_4 by adding alkali oxides (Balachander et al., 2013). As a modifier, lithium which is more electropositive, will cause essential changes in the binary lithium borate system, such as enhancing the bonding strength by forming ionic bonds with oxygen (non-bridging oxygen) and decreasing the hygroscopic nature of borate (Bhogi et al., 2015). Lithium borate is one of the most useful nonlinear optical materials for ultraviolet and visible laser applications. Laser systems with lithium borate as a key component in their design are widely utilised in applications like ophthalmology, materials

processing, marking, optical data storage, and semiconductor manufacturing. Several transition metals including Nb, V, Ti, and Cr are employed as second modifiers to change the glass structure, create non-bridging oxygen (NBO), enhance bond strength, and reduce glass stickiness (Bhagat, 2003). The chemical and physical properties of the glasses are altered when NBO is created. However, the modifications continue to rely on their connectivity to change these properties.

Rare earth (RE)-doped glasses and crystal materials have been extensively studied for fabricating lasers, semiconductors, and amplifiers (Kumar et al., 2013a). RE ions are used to investigate local structural changes in the materials due to their unique spectroscopic properties upon the optical transitions in the intra 4f-shell (Semwal and Bhatt, 2013). Since the demonstration of laser activity in neodymium single crystals made by Krupke (1971), Nd^{3+} has been widely used as a laser-active ion in amorphous and crystalline states (Kumar et al., 2013a). The two host materials most commonly used for this laser ion are yttrium aluminium garnet (YAG) and glass. When doped in YAG, the Nd:YAG crystal produces laser output primarily at 1.064 μm . However, when doped in glass, the Nd:glass medium lases at wavelengths ranging from 1.054 to 1.056 μm , depending upon the type of glass used. In glass, Nd can be doped to extremely high concentration, but it is limited to a maximum concentration of 1.0%–1.5% in YAG crystals (Kuhn, 1998). Furthermore, owing to the commercial importance of Nd^{3+} doped glasses and crystal lasers, many studies on the optical properties and structural role of Nd^{3+} ion and its interaction with other ions have been carried out (Anjaiah et al., 2015; Mhareb et al., 2014; Pal et al., 2013; Vijaya and Suresh, 2012).

Looking at the great prospect of lithium niobium borate containing neodymium in the field of material science, the physical, structural, and optical properties of these materials need to be further investigated.

1.2 Problem Statement

Several studies have been reported regarding the preparation of lithium borate glasses (Prasad and Varma, 2005a; Graca et al., 2008; Kashif et al., 2012; Khalek et al., 2012; Kashif et al., 2013; Farouk et al., 2015). The physical, structural, optical, and electrical properties of the glass systems show a great potential host for nonlinear optical (NLO) and lasing materials. Looking at the great prospect of rare earth doped lithium borate crystal for NLO and laser applications, it is necessary to develop an accurate method for their growth.

Based on the previous studies, Mhareb et al. (2014) has reported the influence of Nd on lithium-magnesium-borate glasses, Kashif et al. (2013) has studied the effect of V, Mn, Co, and Cu on lithium-niobium-borate glasses, and Wu et al. (2011) has reported the influence of Nd on lithium-borate glasses. Those studies only reported the preparation of the glasses using the melt-quench technique, and not much effort has been given to grow a polycrystal seed from melt-quench. Usually, the crystal seeds are made from the Czochralski technique, which is the common practise of crystal growth. (Kashchiev, 2000). However, neodymium-doped lithium-niobium-borate via seeding technique has not been performed yet. Besides that, most of the reported work is limited to the presence of LiNbO_3 nanocrystals by a controlled heat treatment process but emphasis has not been laid down on the formation of crystals by using a seeding technique. Thus, it had become desirable to undertake systematic studies to fully explore the impact of seeding on the modified properties. Moreover, a well-controlled crystal growth process with appropriate growth parameters involving temperature gradient, growth rate, pulling rate, and rotating speed must be achieved in order to produce a good crystal. The physical, thermal, structural, optical, and SHG properties of the crystals and glasses will also be reported.

1.3 Research Objectives

In order to solve the problem as stated in Section 1.2, several objectives have been outlined as follows:

- (a) To grow neodymium modified lithium-niobium-borate polycrystal seed via melt-quench assisted glass route and thermal annealing.
- (b) To determine the crystallisation behaviour and microstructural features of the neodymium-lithium-niobium-borate glass and polycrystal.
- (c) To optimise the growth mechanism of neodymium-lithium-niobium-borate crystals by employing polycrystal, platinum and lithium niobate crystal seeds.
- (d) To determine the improvement in physical, thermal, structural, optical and SHG properties of the crystals and glasses.

1.4 Scope of the Study

In this study, the composition of $(90-x)\text{Li}_2\text{B}_4\text{O}_7+10\text{Nb}_2\text{O}_5+x\text{Nd}_2\text{O}_3$ glass system, where $x = 1, 5, 10, 15,$ and 20 were prepared by conventional melt-quenching technique. The borate glass is chosen as a host due to the high stability and high forming ability of glass. For the preparation of polycrystal seed and crystal growth, the primary focus was only on 1 mol% of Nd_2O_3 . Neodymium oxide was used as a modifier in order to investigate the impact on the physical, structural and optical properties of the glasses and crystals. Consequently, the heat treatment procedures were executed to crystallise the nucleated glass to produce NdLNB polycrystal seed. X-ray diffraction (XRD) and Scanning Electron Microscopy (SEM) were then used to ascertain the crystallisation behaviour and microstructural features of the glass and polycrystal before it will be applied for the next crystal growth process. NdLNB crystals were grown by various types of seed such as polycrystal seed that was prepared in the previous stage, platinum wire and lithium niobate single crystal seed with Automatic Diameter Control - Crystal Growth System (ADC-CGS). The growth and growth mechanism were described, along with the challenges likely to be encountered during growth of the crystal. NdLNB crystals and glasses were subjected to various characterisation techniques and their properties were determined. The densities were measured by Archimedes method. X-ray diffraction (XRD), Differential thermal analyser (DTA), Fourier transformed infrared (FTIR), Ultraviolet-Visible (UV-Vis), Photoluminescence (PL) spectroscopy, Scanning Electron

Microscopy (SEM) and Transmission Electron Microscopy (TEM) characterisations were made to examine the physical, thermal, structural and optical properties. Measurements of the SHG of the crystal and glass samples were performed using the SHG powder technique test set up based on a method of Kurtz and Perry (1968).

1.5 Significance of the Study

This research will give a novelty to the crystal growth process by using a polycrystal seed from an amorphous material and thus inventing a new NLO and lasing material as well. This invention also has the potential to be applied in growing other brand-new glass forming melt crystal such as phosphate, tellurite and silicate which are used for efficient harmonic generation, frequency conversion, laser production, and in other important applications related to nonlinear optics and its associated technologies. Besides, it can nurture knowledge and create new specialists in this area. Moreover, this crystal growth technology is progressively in advance in almost all developed countries. But, unfortunately not much effort has been made in Malaysia to exploit this technology specifically in Czochralski crystal growth technique. Considering its significance, this study has been initiated.

1.6 Thesis Organisation

In this thesis, there are five chapters. Chapter 1 presents the background, problem statement, objectives, and significance of the study. A literature review is presented in Chapter 2. It provides the basis of general information about borate glasses, crystallisation of borate glasses and borate crystals. In addition, a brief description of the crystal growth is highlighted. Nevertheless, the reviews of measurement methods and analysis of each characterisation instrument are also reported. Chapter 3 describes the experimental and measurement techniques, which include sample preparation of glass by the melt-quenching technique, polycrystal by the heat-treatment process, and crystal by the Czochralski technique, and the apparatus used for characterisation and analysis such as DTA, XRD, FTIR, UV-VIS, PL, SEM-

EDX, TEM, and SHG tests. Chapter 4 consists of the experimental results and discussions. This chapter is divided into four sub-sections: the first part describes the synthesis and characterisation of the glass system, the second part explains the characterisation of polycrystal seeds, the third part describes the growth and growth mechanism of the crystals, and the last part presents the comparison of crystal and glass properties. Finally, Chapter 5 summarises the main findings achieved through this research, and suggests several recommendations for future studies. Additional information associated with the thesis is given in the Appendices.

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1. **Kamaruddin, W. H. A.**, Rohani, M. S., Sahar, M. R., Liu, H. and Yuanhua, S. (2016) Synthesis and characterisation of lithium niobium borate glasses containing neodymium, *J. of Rare Earths*, 34(12), pp. 1199-1205. (**Q2, IF 2.188**)

Indexed Journal

1. **Kamaruddin, W. H. A.**, Rohani, M. S., Sahar, M. R. and Liu, H. (2017) Effect of Nd^{3+} on the properties of lithium niobium borate crystal and glass, *Solid State Phenomena*, 268, pp. 210-216. (**Indexed by SCOPUS**)

Non-Indexed Conference Proceedings

1. **Kamaruddin, W. H. A.**, Rohani, M. S., Sahar, M. R. and Liu, H. (2018) Glass formation and crystalline phase in the ternary $\text{Nd}_2\text{O}_3\text{-Li}_2\text{B}_4\text{O}_7\text{-Nb}_2\text{O}_5$ system, 30th Regional Conference on Solid State Science and Technology, Melaka, Malaysia.
2. **Kamaruddin, W. H. A.**, Rohani, M. S., Sahar, M. R. and Liu, H. (2017) Polycrystalline seed-mediated growth of neodymium-lithium-niobium-borate crystal by czochralski technique, 1st International Malaysia-Indonesia-Thailand Symposium, UiTM Perlis, Malaysia. pp. 295-299.
3. **Kamaruddin, W. H. A.**, Rohani, M. S., Sahar, M. R. and Liu, H. (2016) The growth of neodymium-doped lithium niobium borate crystal by czochralski technique, 16th International Graduate Conference on Engineering Science and Humanity 2016, UTM, Johor, Malaysia.
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