

FIRST-PRINCIPLES CALCULATIONS OF STRUCTURAL, ELASTIC,
ELECTRONIC AND TRANSPORT PROPERTIES OF VANADIUM-DOPED
ZIRCONIUM LEAD CARBIDE MAX PHASE

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UNIVERSITI TEKNOLOGI MALAYSIA

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DEDICATION

This thesis is dedicated to my wife Mary, Benedict my son and my daughter Esther for their love, prayers, and patience, throughout the Ph.D. journey. They have been a source of inspiration. Remembering my late parents Mr. Bai Summanuwa and Mrs. Kush'da Bai both of blessed memory; may their souls rest in peace with the Lord; their prayers and love have taught me to be sincere and courageous in life. I missed their company and will forever remain indebted to them.

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ABSTRACT

First-principles calculations have been used to systematically investigate the structural, electronic, elastic, thermodynamic, and transport properties of Zr_2PbC MAX phase and its alloys. The V-based alloys $(V_xZr_{1-x})_2PbC$, $0 \leq x \leq 1$ were synthesized by substituting V on the M-site of the MAX phase at a concentration of 0.25, 0.50, 0.75, and 1.00 respectively. Within the density functional theory (DFT), density functional perturbation theory (DFPT), and Boltzmann transport theory, the generalized gradient approximation (GGA: PBE, PBEsol, PW91), the local density approximation (LDA: PZ) exchange-correlation functionals, and the plane-wave pseudopotential method were used. The examined materials crystallized into a hexagonal shape of space group P63/mmc in relaxed and optimized configurations. The calculated electronic bands and density of states show that the studied MAX phases are conductors. The elastic constants show that all studied materials are mechanically stable based on the Born stability criteria for hexagonal crystals, and structurally stable based on the total minimum energy of the relaxed structures. The 100 % replacement of the Zr atoms shows a significant increase in the Seebeck coefficient and the thermoelectric figure of merit of the terminal MAX phase (V_2PbC). Structurally all the studied materials are hard, brittle, and of high directional anisotropy. Calculated properties have been compared with available experimental data and are in good agreement. All four alloys show a significant increase in the electronic, elastic and thermodynamic properties with a decrease in the lattice parameters as the V concentration increases. The terminal alloy V_2PbC has a lower total energy compared to the Zr_2PbC . The transport properties have been calculated in a temperature range of 200 to 800 K by applying GGA (PBE). For the material at 800 K, a rapid decrease in the thermal conductivity with a slow decrease in electrical conductivity leads to an increase in the figure of merit.

ABSTRAK

Pengiraan prinsip pertama telah digunakan untuk menyiasat secara sistematik sifat struktur, elektronik, elastik, termodinamik dan pengangkutan bagi fasa Zr_2PbC MAX dan aloinya. Aloi berasaskan V (V_xZr_{1-x}) $_2PbC$, $0 \leq x \leq 1$ telah disintesis dengan menggantikan V pada tapak-M fasa MAX pada kepekatan 0.25, 0.50, 0.75, dan 1.00 masing-masing. Dalam teori fungsian ketumpatan (DFT), teori usikan fungsian ketumpatan (DFPT), dan teori pengangkutan Boltzmann, penghampiran kecerunan umum (GGA: PBE, PBEsol, PW91), penghampiran ketumpatan tempatan (LDA: PZ) fungsian korelasi pertukaran, dan kaedah pseudo-keupayaan gelombang satah telah digunakan. Bahan yang diperiksa menghablur menjadi bentuk heksagon kumpulan ruang P63/mmc dalam konfigurasi santai dan teroptimum. Jalur elektronik yang dikira dan ketumpatan keadaan menunjukkan bahawa fasa MAX yang dikaji adalah konduktor. Pemalar anjal menunjukkan semua bahan yang dikaji adalah stabil secara mekanikal berdasarkan kriteria kestabilan Born untuk kristal heksagon, dan stabil secara struktur berdasarkan jumlah tenaga minimum struktur santai. Penggantian 100% atom Zr menunjukkan peningkatan ketara dalam pekali Seebeck dan angka merit termoelektrik fasa MAX terminal (V_2PbC). Secara struktur semua bahan yang dikaji adalah keras, rapuh, dan anisotropi berarah tinggi dalam arah yang sama. Sifat yang dikira telah dibandingkan dengan data eksperimen sedia ada dan berada dalam persetujuan yang baik. Keempat-empat aloi menunjukkan peningkatan ketara dalam sifat elektronik, elastik dan termodinamik dengan penurunan dalam parameter kekisi apabila kepekatan V meningkat. Aloi terminal V_2PbC mempunyai jumlah tenaga minimum berbanding dengan Zr_2PbC . Sifat pengangkutan telah dikira dalam julat suhu 200 hingga 800 K dengan menggunakan GGA (PBE). Untuk bahan pada 800 K, penurunan pesat dalam kekonduksian haba dengan penurunan perlahan dalam kekonduksian elektrik membawa kepada peningkatan dalam angka merit.

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LIST OF ABBREVIATIONS

DFT	Density functional theory
GGA	Generalized gradient approximation
LDA	Local density approximation
PBE	Perdew Burke Ernzerhof
ecut	Cut-off kinetic energy
QE	Quantum Espresso
pw	Plane-wave
CB	Conduction band
VB	Valence band
a.m.u.	Atomic mass unit
DFPT	density-functional perturbation theory
K-S	Kohn-Sham
H-K	Hohenberg-Kohn
TDOS	Total density of states
PDOS	Partial density of states
CRTA	Constant Relaxation Time Approximation

LIST OF SYMBOLS

\AA	Angstrom
σ_τ	Electrical conductivity
τ	Time
σ	Stress tensor
ϵ	Strain tensor
C_{ij}	Elastic constant tensor
ϵ_0	Ground state energy
ϵ_{XC}	Exchange correlation energy
eV	Electron volt
P	Pressure
S_τ	Seebeck coefficient
ϵ_f	Chemical potential
k_0	Thermal conductivity
k_B	Boltzmann constant
F	Free energy
M	Molecular weight
N_A	Avogadro number
ZT	Figure of merit
BTP2	BoltzTraP2
Ψ	Many electrons wave function
ψ	Single-electron wave function
v	Velocity
Ry	Rydberg
H_v	Vicker's hardness
Γ	Gamma
n	Density
n	Number of atoms
θ_D	Debye temperature
C_v	Heat capacity at constant volume
b	Energy bands
S	Entropy
T	Temperature
K	Kelvin

v_l	Longitudinal sound velocity
v_t	Transverse sound velocity
v_m	Mean sound velocity
N_A	Avogadro number

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CHAPTER 1

INTRODUCTION

1.1 Research Background

The growing interest and demand for high-efficiency clean energy sources are critical to developing renewable energy instruments which will alleviate and resolve the energy issue of the world's ever-expanding population [1]–[4]. Traditional energy sources are either depleting or endangering the environment [5]–[7]. The majority of materials developed for thermoelectric generators nowadays are bandgap materials, which are prone to wear and tear at high temperatures[8]–[10].

Thermoelectric materials are a kind of new energy material that can directly convert heat energy into electric energy, and vice versa without needing any other energy conversion via the Seebeck effect of solid materials. Improving the thermoelectric performances of existing thermoelectric materials and exploring new thermoelectric materials with excellent performance are eternal research topics in the thermoelectricity field. The performance of thermoelectric materials can be evaluated by the figure of merit $ZT = (S_{\tau}^2 \sigma_{\tau} / k_{el})T$, where S_{τ} is the Seebeck coefficient, σ_{τ} is the electrical conductivity, k_{el} is the thermal conductivity and T is the temperature [11]. These three parameters for fundamental physical properties of solid are, however, dependent upon each other as a function of the carrier concentration. In particular, S_{τ} and σ_{τ} generally vary reciprocally, making a dramatic improvement in the power factor, $S_{\tau}^2 \sigma_{\tau}$, difficult. Accordingly, semiconducting alloys and compounds with high carrier mobilities have been intensively studied as thermoelectric materials, e.g., Bi_2Te_3 [12], PbTe [13], and Si-Ge [14] alloys. A highly covalent character of these materials appears to be responsible for their high mobilities. The efficiency of thermoelectric energy conversion also improves with increasing temperature differences over which the thermoelectric device operates. In recent years, the MAX phases and their derived MXene phases have gradually received the attention of

researchers as promising thermoelectric materials due to their unique microstructures and properties [15].

The term MAX phases (also called nanolaminates because of their layered structures, which consist of hexagonal carbide or nitride blocks $M_{n+1}X_n$ refers to ternary nitrides and carbides classified into families of $M_{n+1}AX_n$ ($n = 1$ to 3) with over 150 different structures that have been widely studied over the last two and a half decades [16]–[25]. They have a layered structure and a distinct set of features bridging the gap between metals and ceramics characteristics [26]–[29]. The group crystallizes to the hexagonal shape of the $P6_3/mmc$ space group, with M representing the early transition d metals like Cr, Hf, Zr, Ta, Nb, Mo, Ti, and V. The A is the p elements from group 12 to 16 in the periodic table such as Al, Sn, P, Ge, S, In and Si, while the X is nitrogen (N) and, or carbon (C)[30]–[33].

Due to the metal like characteristics they are categorized as good thermal and electrical conductors, plastically deformable at room temperature, damage tolerant, resistant to thermal shock, and relatively soft whereas as ceramics materials, they are often distinguished by high temperature elastic and mechanical properties, as well as superior corrosion resistance [34]–[38]. They are a new class of materials that have intrigued the academic and commercial interests of many since their reappearance in the 1990s [39], [40]. Several MAX phases, emanating from the integer $n = 1$ also known as the 211 MAX phases with examples like Zr_2PbC , Cr_2AlC [41], Cr_2GaC , Hf_2InC , Hf_2SC , Mn_2GaC [42], Mo_2GaC , Nb_2AlC , Nb_2AsC [43]–[46], have been synthesized. Others are the 312 MAX phases with $n = 2$ like Ti_3AlC_2 , Ti_3GeC_2 , Ti_3SnC_2 , Ta_3AlC_2 , Ti_3ZnC_2 , Zr_3AlC_2 , and the 413 MAX phases with $n = 3$ like Ti_4GaC_3 , Ti_4SiC_3 , Ti_4GeC_3 , Nb_4AlC_3 , Ta_4AlC_3 [47], $(Mo,V)_4AlC_3$ are some examples of the various classes of MAX phases to mention just a few that have been synthesized and studied theoretically and experimentally to date[43], [48]–[52].

The well-known potential applications of MAX phase materials include high-temperature applications in the aviation and automobile industries[28] [53], protective and bond-coatings for gas turbines, accident-tolerant fuel cladding in nuclear power plants, solar receivers in concentrated solar power systems, and

electrical contacts for catalysis [37], [50], and joining material[28]. The robustness of the compounds to external forces also called their mechanical stability is a property to ensure its sustainability in any application[54]. Stability and hardness are crucial for industrial applications[54], [55].

The pursuit of advanced technologically high-impact qualities, as well as the opportunity to develop materials with new functionalities, has spurred interest in MAX phases formed by substitution or replacement on the M, A, or X sublattices [41], [46], [56]–[62]. The ability to alloy MAX complexes not only helps to improve their properties but also helps to lower the total energy that would otherwise be metastable in their pure condition [63]. There have been quite a number of both theoretical and experimental research aimed at investigating the different ground and elevated state properties like the structural, electronic, thermodynamic, optical, and transport properties of MAX families of compounds [41], [46], [56]–[58].

According to the literature, the majority of studies on bulk Zr_2PbC focus on the effect of pressure and temperature changes on the properties. While several accessible publications on Zr_2PbC and selected literature on similar crystals have explored the effects of substituting different atoms on the M, A, and X sites on the properties of Zr_2PbC and other selected MAX phases [59]–[62], [64]. The majority of the study has concentrated on the structural, electrical, elastic, and thermodynamic properties of the pure and alloy of Zr_2PbC MAX phase. A handful of such works available on the parent Zr_2PbC MAX phase, and the selected literature on similar nanolaminates have, however, studied the effects of substituting different elements like Mn [65], Cr, Mo [66] [19], Ti, V, Mn, Fe, Mo [67] on the M-site; Sn, Bi, Sb[68], Bi[69], and Pb[70] on the A-site., and C, N [46], respectively, on the X site to investigate the properties.

1.2 Problem Statement

The current state of research on thermoelectric materials limits their engineering uses in thermoelectric devices. Discovering novel thermoelectric materials are continual research areas that need closer attention [71]. MAX phases and their MXene phase counterparts have all it takes to be the alternative thermoelectric materials due to their enormous high-temperature adaptability, and structural and mechanical stability. To harness the thermoelectric potentials in MAX phases, relevant substitution elements of the early transition metals like vanadium V on the M site will enhance the thermoelectric figure of merit by maximizing the Seebeck coefficient, electrical conductivity, and minimizing the electronic thermal conductivity to meet the ever-expanding global energy challenges. Several MAX phases, most notably Ti_3SiC_2 [72], [73], Cr_2AlC , Nb_2AlC [74], Ti_3AlC_2 [75], have been found to possess very low Seebeck coefficients [76], [77]. Previous works of literature on Zr_2PbC [24], [78], Ti_2AlC , V_2AlC , Cr_2AlC , and Nb_2AlC [74] [74], and other MAX phases have focussed mostly on the use of different substitution elements on the M, A, and X-site which has raised the hope to investigate the thermoelectric properties of MAX phases.

Therefore, this has motivated expansive research in the past decade toward the development of novel thermoelectric MAX phases. However, despite many recent studies of these MAX phases several of their physical properties like the Seebeck coefficient issue have not been encouraging due to low values [72], [73][74][75]. This work aims at enhancing the electrical, elastic, thermodynamic, and thermoelectric properties of the parent Zr_2PbC MAX phase using V as a replacement element to enhance the Seebeck coefficient, and electrical conductivity and simultaneously lower the electronic thermal conductivity. Understanding these fundamental properties is of great importance for thermoelectric performance and future technological applications.

1.3 Research Objectives

This thesis investigates the properties of the parent Zr_2PbC MAX phase by employing V as the substituting element on the M-site at a concentration of $(V_xZr_{1-x})_2PbC$, $0.25 \leq x \leq 1.00$, using GGA and LDA exchange functionals, respectively. The specific objectives are to:

- (a) Determine the structural, electronic, elastic, and thermoelectric properties of Zr_2PbC MAX phase.
- (b) Evaluate the structural and electronic properties of $(V_xZr_{1-x})_2PbC$.
- (c) Compute the elastic properties of $(V_xZr_{1-x})_2PbC$
- (d) Calculate the thermoelectric coefficients of $(V_xZr_{1-x})_2PbC$ using GGA XC functional within a temperature range of 200 to 800 K.

1.4 Scope of Research

The structural, electronic, elastic, thermodynamic, and transport properties of $(V_xZr_{1-x})_2PbC$, $0 \leq x \leq 1$, MAX phase have been investigated using the density functional theory (DFT)[79] plane-wave method, density functional perturbation theory(DFPT) [80], and Boltzmann transport theory (BTE) [81], [82], respectively. The GGA (PBE, PBEsol, PW91) [83], and LDA(PZ) [84] exchange functionals, respectively have been used to study the structural, electronic, elastic, and thermodynamic properties of the bulk Zr_2PbC , the V-containing MAX phase using an average converged k-mesh of $14 \times 14 \times 14$ and cut-off energy of 500 Ry. While thermoelectric properties: Seebeck, power factor, electrical and thermal conductivities have been calculated using GGA(PBE) using cut-off energy of 500 Ry and a dense k-mesh of $30 \times 30 \times 30$ within a temperature of 200 to 800 K.

1.5 Significance of Study

Thermoelectricity has recently gained recognition due to its capacity to transform waste heat into energy while emitting no toxic toxins. The rebirth of research into thermoelectric materials began in the early 1990s [15]. Thermoelectric materials are environmentally friendly for power generation and refrigeration, thus providing a solution for energy crisis and pollution; however, the thermoelectric conversion efficiency is low and mainly limited by the performance of thermoelectric materials. New concepts and technologies were applied recently to enhance ZT, but accompanied difficulties need to be solved. There is still a long way between high thermoelectric performance and high thermoelectric conversion efficiency. Building a device that could reach the theoretical efficiency is not a trivial pursuit, but a huge development project by itself considering the tremendous practical challenges involved, suitable low resistance hot side, and cold side metal contacts, optimizing assembly of modules, etc. The development of thermoelectric materials and devices needs connected efforts involving physicists, chemists, materials scientists, and theory scientists[11].

This work is an effort to enhance the thermoelectric coefficients and other properties of Zr_2PbC by substituting the Zr atoms with V atoms. research on MAX phases has intrigued the interest of many people and continues to do so due to their unusual hybrid metal/ceramic capabilities coming from their structure and atomic arrangement, and they are being evaluated for several applications[85], [86]. The simplicity with which MAX phases may be machined to high tolerance is also a key factor to consider when designing tools for dry drilling concrete surfaces in civil engineering tasks. This may bring the materials close to the market. Other applications include electrodes, free-cutting elements[26], exhaust gas filters for automobiles, corrosion-resistant materials, surface coatings materials, and microelectronics[87].

REFERENCES

- [1] A. Jain *et al.*, “Commentary: The materials project: A materials genome approach to accelerating materials innovation,” *APL Mater.*, vol. 1, no. 1, 2013, doi: 10.1063/1.4812323.
- [2] H. J. Goldsmid, *Optimisation and selection of semiconductor thermoelements*, vol. 121. 2016.
- [3] D. J. Singh, “Thermopower of SnTe from Boltzmann transport calculations,” *Funct. Mater. Lett.*, vol. 3, no. 4, pp. 223–226, 2010, doi: 10.1142/S1793604710001299.
- [4] D. Li *et al.*, “Recent Progress of Two-Dimensional Thermoelectric Materials,” *Nano-Micro Lett.*, vol. 12, no. 1, p. 36, Dec. 2020, doi: 10.1007/s40820-020-0374-x.
- [5] S. Ahmad, R. Ahmad, M. Bilal, and N. U. Rehman, “DFT studies of thermoelectric properties of R–Au intermetallics at 300 K,” *J. Rare Earths*, vol. 36, no. 2, pp. 197–202, 2018, doi: 10.1016/j.jre.2017.08.004.
- [6] A. Gaul and T. Advisor, “A Theoretical Study of Enhancing Thermoelectric Efficiency in Pnictogen-Chalcogen Alloys via Doping , Strain , and Nanostructuring,” Rensselaer Polytechnic Institute Troy, New York March, 2018.
- [7] M. Ohtaki, T. Tsubota, K. Eguchi, and H. Arai, “High-temperature thermoelectric properties of $(\text{Zn}_{1-x}\text{Al}_x)\text{O}$,” *J. Appl. Phys.*, vol. 79, no. 3, pp. 1816–1818, 1996, doi: 10.1063/1.360976.
- [8] H. Salehi, P. Amiri, and R. Z. Hasanabad, “Ab-initio Study of Electronic , Optical , Dynamic and Thermoelectric Properties of CuSbX_2 (X = S , Se) Compounds,” vol. 3, no. 2, 2018.
- [9] M. Hammi, O. El Rhazouani, M. Arejdal, and A. Slassi, “Ab initio study of semi-classic transport coefficients of SnO_2 thermoelectric material,” *Chinese J. Phys.*, vol. 55, no. 1, pp. 187–194, 2017, doi: 10.1016/j.cjph.2016.10.016.
- [10] S. D. Guo, “Thermoelectric properties of half-Heusler ZrNiPb by using first principles calculations,” *RSC Adv.*, vol. 6, no. 53, pp. 47953–47958, 2016, doi: 10.1039/c6ra08461c.

- [11] X. Zhang and L. Zhao, “Thermoelectric materials: Energy conversion between heat and electricity,” *J. Mater.*, vol. 1, no. 2, pp. 92–105, Jun. 2015, doi: 10.1016/j.jmat.2015.01.001.
- [12] M. Scheele, N. Oeschler, K. Meier, A. Kornowski, C. Klinke, and H. Weller, “Synthesis and thermoelectric characterization of Bi₂Te₃ nanoparticles,” *Adv. Funct. Mater.*, vol. 19, no. 21, pp. 3476–3483, 2009, doi: 10.1002/adfm.200901261.
- [13] Y. Pei, A. Lalonde, S. Iwanaga, and G. J. Snyder, “High thermoelectric figure of merit in heavy hole dominated PbTe,” *Energy Environ. Sci.*, vol. 4, no. 6, pp. 2085–2089, 2011, doi: 10.1039/c0ee00456a.
- [14] Y. Shiraki and N. Usami, *Silicon – germanium nanostructures*. Woodhead Publishing Limited, 2011.
- [15] J. Mao, Z. Liu, and Z. Ren, “Size effect in thermoelectric materials,” *npj Quantum Mater.*, vol. 1, no. November, pp. 1–9, 2016, doi: 10.1038/npjquantmats.2016.28.
- [16] M. W. Barsoum *et al.*, “Thermal properties of Ti₃SiC₂,” *J. Phys. Chem. Solids*, vol. 60, pp. 429–439, 1999.
- [17] X. He, Y. Bai, Y. Li, C. Zhu, and M. Li, “Ab initio calculations for properties of MAX phases Ti₂InC, Zr₂InC, and Hf₂InC,” vol. 149, pp. 564–566, 2009, doi: 10.1016/j.ssc.2008.12.047.
- [18] Z. J. Lin, M. J. Zhuo, Y. C. Zhou, M. S. Li, and J. Y. Wang, “Structural characterization of a new layered-ternary Ta₄AlC₃ ceramic,” *J. Mater. Res.*, vol. 21, no. 10, pp. 2587–2592, 2006, doi: 10.1557/jmr.2006.0310.
- [19] P. A. Burr, D. Horlait, and W. E. Lee, “Experimental and DFT investigation of (Cr,Ti)₃AlC₂ MAX phases stability,” *Mater. Res. Lett.*, vol. 5, no. 3, pp. 144–157, May 2017, doi: 10.1080/21663831.2016.1222598.
- [20] X. K. Qian, X. D. He, Y. B. Li, Y. Sun, H. Li, and D. L. Xu, “Cyclic oxidation of Ti₃AlC₂ at 1000 – 1300 ° C in air,” vol. 53, pp. 290–295, 2011, doi: 10.1016/j.corsci.2010.09.033.
- [21] F. Sultana, M. M. Uddin, M. A. Ali, M. M. Hossain, S. H. Naqib, and A. K. M. A. Islam, “First principles study of M₂InC (M = Zr, Hf and Ta) MAX phases: The effect of M atomic species,” *Results Phys.*, vol. 11, no. May, pp. 869–876, 2018, doi: 10.1016/j.rinp.2018.10.044.
- [22] W. J. Chen *et al.*, “Experimental realization of photonic topological insulator

- in a uniaxial metacrystal waveguide,” *Nat. Commun.*, vol. 5, pp. 1–8, 2014, doi: 10.1038/ncomms6782.
- [23] W. Jeitschko and H. Nowotny, “Die Kristallstruktur von Ti_3SiC_2 ein neuer Komplexearbid-Typ,” *Monatshefte für Chemie - Chem. Mon.*, vol. 98, no. 2, pp. 329–337, Mar. 1967, doi: 10.1007/BF00899949.
- [24] W. Jeitschko, H. Nowotny, and F. Benesovsky, “ Ti_2AlN , eine stickstoffhaltige H-Phase,” *Monatshefte für Chemie*, vol. 94, no. 6, pp. 1198–1200, 1963, doi: 10.1007/BF00905710.
- [25] P. Eklund, M. Beckers, U. Jansson, H. Högberg, and L. Hultman, “The $M_{n+1}AX_n$ phases: Materials science and thin-film processing,” *Thin Solid Films*, vol. 518, no. 8, pp. 1851–1878, Feb. 2010, doi: 10.1016/j.tsf.2009.07.184.
- [26] R. Arróyave, A. Talapatra, T. Duong, W. Son, and M. Radovic, “Out-of-plane ordering in quaternary MAX alloys: an alloy theoretic perspective,” *Mater. Res. Lett.*, vol. 6, no. 1, pp. 1–12, Jan. 2018, doi: 10.1080/21663831.2017.1380723.
- [27] W. Yu *et al.*, “Solid solution effects in the $Ti_2Al(C_xN_y)$ MAX phases: Synthesis, microstructure, electronic structure and transport properties,” *Acta Mater.*, vol. 80, pp. 421–434, Nov. 2014, doi: 10.1016/j.actamat.2014.07.064.
- [28] J. Gonzalez-Julian, “Processing of MAX phases: From synthesis to applications,” *Journal of the American Ceramic Society*, vol. 104, no. 2, pp. 659–690, 2021, doi: 10.1111/jace.17544.
- [29] P. Eklund, J. Rosen, and P. O. Å. Persson, “Layered ternary $M_{n+1}AX_n$ phases and their 2D derivative MXene: An overview from a thin-film perspective,” *J. Phys. D. Appl. Phys.*, vol. 50, no. 11, 2017, doi: 10.1088/1361-6463/aa57bc.
- [30] Y. Bai, N. Srikanth, C. K. Chua, and K. Zhou, “Density Functional Theory Study of $M_{n+1}AX_n$ Phases : A Review Density Functional Theory Study of $M_{n+1}AX_n$ Phases : A Review,” *Crit. Rev. Solid State Mater. Sci.*, vol. 0, no. 0, pp. 1–52, 2019, doi: 10.1080/10408436.2017.1370577.
- [31] A. S. Ingason *et al.*, “A Nanolaminated Magnetic Phase: Mn_2GaC ,” *Mater. Res. Lett.*, vol. 2, no. 2, pp. 89–93, Apr. 2014, doi: 10.1080/21663831.2013.865105.
- [32] P. Eklund, J. Rosen, and P. O. Å. Persson, “Layered ternary $M_{n+1}AX_n$ phases and their 2D derivative MXene: an overview from a thin-film perspective,” *J.*

- Phys. D. Appl. Phys.*, vol. 50, no. 11, p. 113001, Mar. 2017, doi: 10.1088/1361-6463/aa57bc.
- [33] R. Khatun, M. A. Rahman, K. M. Hossain, M. Z. Hasan, M. Rasheduzzaman, and S. Sarker, “Physical properties of MAX phase Zr_2PbC under pressure: Investigation via DFT scheme,” *Phys. B Condens. Matter*, vol. 620, no. July, p. 413258, Nov. 2021, doi: 10.1016/j.physb.2021.413258.
- [34] I. Zhirkov, L. Landälv, E. Göthelid, M. Ahlgren, P. Eklund, and J. Rosen, “Effect of Si on DC arc plasma generation from Al-Cr and Al-Cr-Si cathodes used in oxygen,” *J. Appl. Phys.*, vol. 121, no. 8, 2017, doi: 10.1063/1.4976862.
- [35] M. W. Barsoum, “The $M_{n+1}AX_n$ phases: A new class of solids,” *Prog. Solid State Chem.*, vol. 28, no. 1–4, pp. 201–281, Jan. 2000, doi: 10.1016/S0079-6786(00)00006-6.
- [36] T. El-Raghy, M. W. Barsoum, A. Zavaliangos, and S. R. Kalidindi, “Processing and mechanical properties of Ti_3SiC_2 : II, Effect of grain size and deformation temperature,” *J. Am. Ceram. Soc.*, vol. 82, no. 10, pp. 2855–2860, 1999, doi: 10.1111/j.1151-2916.1999.tb02167.x.
- [37] W. H. K. Ng *et al.*, “The Ti_3AlC_2 MAX Phase as an Efficient Catalyst for Oxidative Dehydrogenation of n-Butane,” pp. 1–7, 2018, doi: 10.1002/anie.201702196.
- [38] M. Yoshida, Y. Hoshiyama, J. Ommyoji, and A. Yamaguchi, “Reaction mechanism for the synthesis of Ti_3AlC_2 through an intermediate carbide of Ti_3AlC from elemental Ti, Al, and C powder mixture,” *J. Ceram. Soc. Japan*, vol. 118, no. 1373, pp. 37–42, 2010, doi: 10.2109/jcersj2.118.37.
- [39] M. Barsoum, “The MAX Phases: Unique New Carbide and Nitride Materials,” no. November, 2001, doi: 10.1511/2001.28.736.
- [40] J. Wang and Y. Zhou, “Recent Progress in Theoretical Prediction, Preparation, and Characterization of Layered Ternary Transition-Metal Carbides,” *Annu. Rev. Mater. Res.*, vol. 39, no. 1, pp. 415–443, Aug. 2009, doi: 10.1146/annurev-matsci-082908-145340.
- [41] J. C. Schuster, H. Nowotny, and C. Vaccaro, “The ternary systems: $CrAlC$, $VAlC$, and $TiAlC$ and the behavior of H-phases (M_2AlC),” *J. Solid State Chem.*, vol. 32, no. 2, pp. 213–219, Apr. 1980, doi: 10.1016/0022-4596(80)90569-1.

- [42] A. S. Ingason *et al.*, “A Nanolaminated Magnetic Phase : Mn_2GaC ,” vol. 3831, 2014, doi: 10.1080/21663831.2013.865105.
- [43] C. Anirudh, V. K. A. V, and U. N. Kempaiah, “Max Phase Materials- Review of an Exciting Class of Ternary Carbides and Nitrides,” *Int. J. Emerg. Tech. Adv. Eng.*, vol. 4, no. 8, pp. 624–630, 2014, [Online]. Available: www.ijetae.com.
- [44] V. A. Gorshkov, A. V. Karpov, D. Y. Kovalev, and A. E. Sychev, “Synthesis, Structure and Properties of Material Based on V_2AlC MAX Phase,” *Phys. Met. Metallogr.*, vol. 121, no. 8, pp. 765–771, 2020, doi: 10.1134/S0031918X20080037.
- [45] B. Velasco, E. Gordo, L. Hu, M. Radovic, and S. A. Tsipas, “Influence of porosity on elastic properties of Ti_2AlC and Ti_3SiC_2 MAX phase foams,” *J. Alloys Compd.*, vol. 764, pp. 24–35, Oct. 2018, doi: 10.1016/j.jallcom.2018.06.027.
- [46] M. W. Barsoum, M. Ali, and T. El-Raghy, “Processing and characterization of Ti_2AlC , Ti_2AlN , and $Ti_2AlC_{0.5}N_{0.5}$,” *Metall. Mater. Trans. A Phys. Metall. Mater. Sci.*, vol. 31, no. 7, pp. 1857–1865, 2000, doi: 10.1007/s11661-006-0243-3.
- [47] B. Manoun, S. K. Saxena, T. El-Raghy, and M. W. Barsoum, “High-pressure x-ray diffraction study of Ta_4AlC_3 ,” *Appl. Phys. Lett.*, vol. 88, no. 20, pp. 2004–2007, 2006, doi: 10.1063/1.2202387.
- [48] B. Anasori *et al.*, “Experimental and theoretical characterization of ordered MAX phases Mo_2TiAlC_2 and $Mo_2Ti_2AlC_3$,” *J. Appl. Phys.*, vol. 118, no. 9, p. 094304, 2015, doi: 10.1063/1.4929640.
- [49] L. J. Liu, L. Lian, and J. Yu, “Stability, mechanical properties and anisotropic elastic properties of Ga_xMg_y compounds,” *Mater. Res.*, vol. 22, no. 2, pp. 1–11, 2019, doi: 10.1590/1980-5373-MR-2018-0624.
- [50] M. Atikur Rahman, “Study on Structural, Electronic, Optical and Mechanical Properties of MAX Phase Compounds and Applications Review Article,” *Am. J. Mod. Phys.*, vol. 4, no. 2, p. 75, 2015, doi: 10.11648/j.ajmp.20150402.15.
- [51] M. Dahlqvist, J. Lu, R. Meshkian, Q. Tao, L. Hultman, and J. Rosen, “Prediction and synthesis of a family of atomic laminate phases with Kagomé-like and in-plane chemical ordering,” *Science Advances*, vol. 3, no. 7. 2017, doi: 10.1126/sciadv.1700642.

- [52] A. S. Ingason *et al.*, “Magnetic MAX phases from theory and experiments ; a review,” vol. 28, no. 43, 2016.
- [53] Z. M. Sun, “Progress in research and development on MAX phases: A family of layered ternary compounds,” *Int. Mater. Rev.*, vol. 56, no. 3, pp. 143–166, 2011, doi: 10.1179/1743280410Y.0000000001.
- [54] G. Surucu, “Investigation of structural, electronic, anisotropic elastic, and lattice dynamical properties of MAX phases borides: An Ab-initio study on hypothetical M_2AB ($M = Ti, Zr, Hf$; $A = Al, Ga, In$) compounds,” *Mater. Chem. Phys.*, vol. 203, pp. 106–117, 2018, doi: 10.1016/j.matchemphys.2017.09.050.
- [55] A. Gencer and G. Surucu, “Electronic and Lattice Dynamical Properties of Ti_2SiB MAX Phase,” no. July, 2018.
- [56] M. W. Barsoum *et al.*, “Thermal and electrical properties of $Nb_2AlC, (Ti, Nb)_2AlC$ and Ti_2AlC ,” *Metall. Mater. Trans. A Phys. Metall. Mater. Sci.*, vol. 33, no. 9, pp. 2775–2779, 2002, doi: 10.1007/s11661-002-0262-7.
- [57] B. Manoun, S. K. Saxena, G. Hug, A. Ganguly, E. N. Hoffman, and M. W. Barsoum, “Synthesis and compressibility of $Ti_3(Al, Sn_{0.2})C_2$ and $Ti_3Al(C_{0.5}, N_{0.5})_2$,” *J. Appl. Phys.*, vol. 101, no. 11, pp. 0–7, 2007, doi: 10.1063/1.2733644.
- [58] H. Yang, B. Manoun, R. T. Downs, A. Ganguly, and M. W. Barsoum, “Crystal chemistry of layered carbide, $Ti_3(Si_{0.43}Ge_{0.57})C_2$ Hexiong,” *J. Phys. Chem. Solids*, vol. 67, no. 12, pp. 2512–2516, 2006, doi: 10.1016/j.jpcs.2006.07.009.
- [59] M. Atikur Rahman, “Study on Structural, Electronic, Optical and Mechanical Properties of MAX Phase Compounds and Applications Review Article,” *Am. J. Mod. Phys.*, vol. 4, no. 2, p. 75, 2015, doi: 10.11648/j.ajmp.20150402.15.
- [60] R. Khatun, “Structural, elastic, electronic, optical and thermodynamic properties of MAX phase compound Zr_2PbC under pressure: DFT based investigation,” vol. 6, no. April, 2020.
- [61] X. Qian *et al.*, “First-principle studies of properties of ternary layered M_2PbC ($M=Ti, Zr$ and Hf),” *Comput. Mater. Sci.*, vol. 65, pp. 377–382, Dec. 2012, doi: 10.1016/j.commatsci.2012.08.004.
- [62] M. B. Kanoun, S. Goumri-Said, A. H. Reshak, and A. E. Merad, “Electro-

- structural correlations, elastic and optical properties among the nanolaminated ternary carbides Zr_2AC ,” *Solid State Sci.*, vol. 12, no. 5, pp. 887–898, 2010, doi: 10.1016/j.solidstatesciences.2010.01.035.
- [63] D. Horlait, S. Grasso, A. Chroneos, and W. E. Lee, “Attempts to synthesise quaternary MAX phases $(Zr,M)_2AlC$ and $Zr_2(Al,A)C$ as a way to approach Zr_2AlC ,” *Mater. Res. Lett.*, vol. 4, no. 3, pp. 137–144, 2016, doi: 10.1080/21663831.2016.1143053.
- [64] T. El-Raghy, S. Chakraborty, and M. . Barsoum, “Synthesis and characterization of Hf_2PbC , Zr_2PbC and M_2SnC ($M=Ti, Hf, Nb$ or Zr),” *J. Eur. Ceram. Soc.*, vol. 20, no. 14–15, pp. 2619–2625, Dec. 2000, doi: 10.1016/S0955-2219(00)00127-8.
- [65] C. M. Hamm *et al.*, “Structural, magnetic and electrical transport properties of non-conventionally prepared MAX phases V_2AlC and $(V/Mn)_2AlC$,” *Mater. Chem. Front.*, vol. 2, no. 3, pp. 483–490, 2018, doi: 10.1039/c7qm00488e.
- [66] Y. F. Li, Y. C. Ding, B. Xiao, and Y. H. Cheng, “Anisotropic electrical and lattice transport properties of ordered quaternary phases Cr_2TiAlC_2 and Mo_2TiAlC_2 : A first principles study,” *Phys. Lett. Sect. A Gen. At. Solid State Phys.*, vol. 380, no. 44, pp. 3748–3755, 2016, doi: 10.1016/j.physleta.2016.09.015.
- [67] S. Lin *et al.*, “Alloying effects on structural, magnetic, and electrical/thermal transport properties in MAX-phase $Cr_{2-x}M_xGeC$ ($M = Ti, V, Mn, Fe$, and Mo),” *J. Alloys Compd.*, vol. 680, pp. 452–461, 2016, doi: 10.1016/j.jallcom.2016.04.197.
- [68] M. A. Ali, M. M. Hossain, N. Jahan, A. K. M. A. Islam, and S. H. Naqib, “Newly synthesized Zr_2AlC , $Zr_2(Al_{0.58}Bi_{0.42})C$, $Zr_2(Al_{0.2}Sn_{0.8})C$, and $Zr_2(Al_{0.3}Sb_{0.7})C$ MAX phases: A DFT based first-principles study,” *Comput. Mater. Sci.*, vol. 131, pp. 139–145, Apr. 2017, doi: 10.1016/j.commatsci.2017.01.048.
- [69] D. Horlait, S. C. Middleburgh, A. Chroneos, and W. E. Lee, “Synthesis and DFT investigation of new bismuth-containing MAX phases,” *Nat. Publ. Gr.*, no. January, pp. 1–9, 2016, doi: 10.1038/srep18829.
- [70] D. Horlait *et al.*, “Attempts to synthesise quaternary MAX phases $(Zr,M)_2AlC$ and $Zr_2(Al,A)C$ as a way to approach Zr_2AlC ,” vol. 3831, 2016, doi: 10.1080/21663831.2016.1143053.

- [71] C. Liu, Y.-Y. Yang, C.-W. Nan, and Y.-H. Lin, “Thermoelectric properties and prospects of MAX phases and derived MXene phases,” *Acta Phys. Sin.*, vol. 70, no. 20, p. 206501, 2021, doi: 10.7498/aps.70.20211050.
- [72] M. Magnuson, M. Mattesini, N. Van Nong, P. Eklund, and L. Hultman, “The electronic-structure origin of the anisotropic thermopower of nanolaminated Ti_3SiC_2 determined by polarized x-ray spectroscopy and Seebeck measurements,” vol. 195134, pp. 1–13, 2012.
- [73] M. Magnuson, M. Mattesini, N. Van Nong, P. Eklund, and L. Hultman, “Electronic-structure origin of the anisotropic thermopower of nanolaminated Ti_3SiC_2 determined by polarized x-ray spectroscopy and Seebeck measurements,” vol. 195134, pp. 1–8, 2012, doi: 10.1103/PhysRevB.85.195134.
- [74] J. D. Hettinger *et al.*, “Electrical transport , thermal transport , and elastic properties of M_2AlC ($\text{M}=\text{Ti}$, Cr , Nb , and V),” no. May 2014, pp. 2–7, 2005, doi: 10.1103/PhysRevB.72.115120.
- [75] T. Scabarozzi *et al.*, “Electronic and thermal properties of $\text{Ti}_3\text{Al}(\text{C}_{0.5}\text{N}_{0.5})_2$, $\text{Ti}_2\text{Al}(\text{C}_{0.5}\text{N}_{0.5})$ and Ti_2AlN ,” *J. Appl. Phys.*, vol. 104, no. 7, pp. 1–7, 2008, doi: 10.1063/1.2979326.
- [76] M. W. Barsoum, “Physical Properties of the MAX Phases,” *Encycl. Mater. Sci. Technol.*, no. Table 1, pp. 1–11, 2006, doi: 10.1016/b0-08-043152-6/02058-1.
- [77] B. M. Radovic and M. W. Barsoum, “MAX phases : Bridging the gap between metals and ceramics,” *Am. Ceram. Soc. Bull.*, vol. 92, no. 3, pp. 20–27, 2012, [Online]. Available: www.ceramics.org.
- [78] W. Jeitschko, H. Nowotny, and F. Benesovsky, “Carbides of formula T_2MC ,” *J. Less-Common Met.*, vol. 7, no. 2, pp. 133–138, 1964, doi: 10.1016/0022-5088(64)90055-4.
- [79] W. Son, T. Duong, A. Talapatra, H. Gao, R. Arróyave, and M. Radovic, “Ab-initio investigation of the finite-temperatures structural, elastic, and thermodynamic properties of Ti_3AlC_2 and Ti_3SiC_2 ,” *Comput. Mater. Sci.*, vol. 124, pp. 420–427, 2016, doi: 10.1016/j.commatsci.2016.08.015.
- [80] A. Champagne *et al.*, “Phonon dispersion curves in Cr_2AlC single-crystals,” *Mater. Res. Lett.*, vol. 6, no. 7, pp. 378–383, Jul. 2018, doi: 10.1080/21663831.2018.1463298.

- [81] G. K. H. Madsen, J. Carrete, and M. J. Verstraete, “BoltzTraP2, a program for interpolating band structures and calculating semi-classical transport coefficients,” *Comput. Phys. Commun.*, vol. 231, pp. 140–145, 2018, doi: 10.1016/j.cpc.2018.05.010.
- [82] G. K. H. Madsen and D. J. Singh, “BoltzTraP. A code for calculating band-structure dependent quantities,” *Comput. Phys. Commun.*, vol. 175, no. 1, pp. 67–71, 2006, doi: 10.1016/j.cpc.2006.03.007.
- [83] J. P. Perdew, K. Burke, and M. Ernzerhof, “Generalized Gradient Approximation Made Simple,” *Phys. Rev. Lett.*, vol. 77, no. 18, pp. 3865–3868, Oct. 1996, doi: 10.1103/PhysRevLett.77.3865.
- [84] V. I. Anisimov, F. Aryasetiawan, and A. I. Lichtenstein, “First-principles calculations of the electronic structure and spectra of strongly correlated systems: The LDA + U method,” *J. Phys. Condens. Matter*, vol. 9, no. 4, pp. 767–808, 1997, doi: 10.1088/0953-8984/9/4/002.
- [85] P. Eklund, M. Beckers, U. Jansson, H. Högberg, and L. Hultman, “Linköping University Post Print The $M_{n+1}AX_n$ phases: Materials science and thin-film processing,” *Mater. Sci. thin-film Process.*, no. 518, pp. 1851–1878, 2010, doi: 10.1016/j.tsf.2009.07.184.
- [86] M. W. Barsoum and M. Radovic, “Elastic and Mechanical Properties of the MAX Phases,” *Annu. Rev. Mater. Res.*, vol. 41, no. 1, pp. 195–227, 2011, doi: 10.1146/annurev-matsci-062910-100448.
- [87] M. N. Abdelmalak, “A New Family of Two-Dimensional Materials and its Application as Electrodes for Li-ion Batteries,” *Thesis*, no. April, 2014.
- [88] M. Zhang, W. Tian, P. Zhang, J. Ding, Y. Zhang, and Z. Sun, “Microstructure and properties of Ag – Ti_3SiC_2 contact materials prepared by pressureless sintering,” *Int. J. Miner. Metall. Mater.*, vol. 25, no. 7, pp. 810–816, 2018.
- [89] Y. Hadji *et al.*, “Microstructure and microindentation of Ti_3SiC_2 – Titanium filler brazed joints by tungsten inert gas (TIG) process,” *Ceram. Int.*, vol. 43, no. 9, pp. 7290–7294, 2017, doi: 10.1016/j.ceramint.2017.03.028.
- [90] C. Xu *et al.*, “First-principles calculations of Ti_3SiC_2 and Ti_3AlC_2 with hydrogen interstitial,” *J. Nucl. Mater.*, vol. 488, pp. 261–266, 2017, doi: 10.1016/j.jnucmat.2016.09.025.
- [91] A. S. Ingason, M. Dahlqvist, and J. Rosen, “Magnetic MAX phases from theory and experiments; A review,” *J. Phys. Condens. Matter*, vol. 28, no. 43,

- 2016, doi: 10.1088/0953-8984/28/43/433003.
- [92] C. Dhakal, S. Aryal, R. Sakidja, and W. Y. Ching, “Approximate lattice thermal conductivity of MAX phases at high temperature,” *J. Eur. Ceram. Soc.*, vol. 35, no. 12, pp. 3203–3212, 2015, doi: 10.1016/j.jeurceramsoc.2015.04.013.
- [93] K. Lambrinou, T. Lapauw, and J. Vleugels, “Exploring the Potential of MAX Phases for Select Applications in Extreme Environments,” pp. 1–32, 2015.
- [94] D. J. Tallman, “On the Potential of MAX phases for Nuclear Applications,” no. June, 2015.
- [95] W. H. K. Ng *et al.*, “The Ti_3AlC_2 MAX Phase as an Efficient Catalyst for Oxidative Dehydrogenation of n-Butane,” *Angew. Chemie - Int. Ed.*, vol. 57, no. 6, pp. 1485–1490, 2018, doi: 10.1002/anie.201702196.
- [96] A. S. Ingason *et al.*, “A nanolaminated magnetic phase: Mn_2GaC ,” *Mater. Res. Lett.*, vol. 2, no. 2, pp. 89–93, 2017, doi: 10.1080/21663831.2013.865105.
- [97] I. Persson *et al.*, “On the organization and thermal behavior of functional groups on Ti_3C_2 MXene surfaces in vacuum,” *2D Mater.*, 2017, doi: 10.1088/2053-1583/aa89cd.
- [98] M. Khazaei *et al.*, “Novel Electronic and Magnetic Properties of Two-Dimensional Transition Metal Carbides and Nitrides,” pp. 2185–2192, 2013, doi: 10.1002/adfm.201202502.
- [99] M. Atikur Rahman, “Study on Structural, Electronic, Optical and Mechanical Properties of MAX Phase Compounds and Applications Review Article,” *American Journal of Modern Physics*, vol. 4, no. 2, p. 75, 2015, doi: 10.11648/j.ajmp.20150402.15.
- [100] V. Gray, E. H. Kisi, O. Kirstein, and A. P. J. Stamp, “Elastic constants of polycrystalline Ti_3AlC_2 and Ti_3SiC_2 measured using coherent inelastic neutron scattering,” no. September, pp. 705–712, 2016, doi: 10.1111/jace.14600.
- [101] M. K. Drulis, H. Drulis, S. Gupta, M. W. Barsoum, M. K. Drulis, and H. Drulis, “On the heat capacities of M_2AlC ($\text{M} = \text{Ti}, \text{V}, \text{Cr}$) ternary carbides,” vol. 093502, 2006, doi: 10.1063/1.2191744.
- [102] L. Chaput, G. Hug, P. Pécheur, and H. Scherrer, “Anisotropy and thermopower in Ti_3SiC_2 ,” *Phys. Rev. B - Condens. Matter Mater. Phys.*, vol. 71, no. 12, pp. 2–4, 2005, doi: 10.1103/PhysRevB.71.121104.

- [103] S. Fukada-tanaka and Y. Inagaki, “Colour-enhancing protein in blue petals,” vol. 407, no. October, pp. 581–582, 2000.
- [104] Y. Zhao, S. Deng, H. Liu, J. Zhang, Z. Guo, and H. Hou, “First-principle investigation of pressure and temperature influence on structural, mechanical and thermodynamic properties of Ti_3AC_2 ($A = Al$ and,” *Comput. Mater. Sci.*, vol. 154, no. July, pp. 365–370, 2018, doi: 10.1016/j.commatsci.2018.07.007.
- [105] M. A. Ali and M. W. Qureshi, “Newly synthesized MAX phase Zr_2SeC : DFT insights into physical properties towards possible applications,” *RSC Adv.*, vol. 11, no. 28, pp. 16892–16905, 2021, doi: 10.1039/D1RA02345D.
- [106] M. W. Barsoum, T. H. Scabarozi, S. Amini, J. D. Hettinger, and S. E. Lofland, “Electrical and thermal properties of Cr_2GeC ,” *J. Am. Ceram. Soc.*, vol. 94, no. 12, pp. 4123–4126, 2011, doi: 10.1111/j.1551-2916.2011.04886.x.
- [107] S. Lin *et al.*, “Magnetic and electrical/thermal transport properties of Mn-doped $M_{n+1}AX_n$ phase compounds $Cr_{2-x}Mn_x GaC$ ($0 \leq x \leq 1$),” *J. Appl. Phys.*, vol. 113, no. 5, p. 053502, Feb. 2013, doi: 10.1063/1.4789954.
- [108] S. Aryal, R. Sakidja, M. W. Barsoum, and W. Y. Ching, “A genomic approach to the stability, elastic, and electronic properties of the MAX phases,” *Phys. Status Solidi Basic Res.*, vol. 251, no. 8, pp. 1480–1497, 2014, doi: 10.1002/pssb.201451226.
- [109] O. Kirstein, J. F. Zhang, E. H. Kisi, D. P. Riley, M. J. Styles, and A. Paradowska, “Single crystal elastic constants of the MAX phase Ti_3AlC_2 determined by neutron diffraction,” *Mater. Sci. Forum*, vol. 638–642, pp. 2417–2422, 2010, doi: 10.4028/www.scientific.net/MSF.638-642.2417.
- [110] Veronica Gray *et al.*, “ $M_{n+1}AX_n$ Phonon Density of States: Ti_3AlC_2 and Ti_3SiC_2 Simulation and Experimentation Results,” *J. Chem. Chem. Eng.*, vol. 9, no. 4, 2015, doi: 10.17265/1934-7375/2015.04.003.
- [111] H. Wang *et al.*, “First-principles study of vacancies in Ti_3SiC_2 and Ti_3AlC_2 ,” *Materials (Basel)*, vol. 10, no. 2, 2017, doi: 10.3390/ma10020103.
- [112] H. Wang, Z. Song, D. Qiao, D. Feng, and J. Lu, “Tribological Behavior of Si_3N_4 / Ti_3SiC_2 Contacts Lubricated by Lithium-Based Ionic Liquids,” vol. 2014, 2014.
- [113] A. G. Zhou and M. W. Barsoum, “Kinking nonlinear elastic deformation of Ti_3AlC_2 , Ti_2AlC , $Ti_3Al(C_{0.5},N_{0.5})_2$ and $Ti_2Al(C_{0.5},N_{0.5})$,” *J. Alloys Compd.*, vol.

- 498, no. 1, pp. 62–70, 2010, doi: 10.1016/j.jallcom.2010.03.099.
- [114] K. Wang, Y. Zhou, C. Yu, M. Xiang, D. Huang, and W. Xu, “Synthesis and strengthening of Ti_3AlC_2 ,” *J. Alloys Compd.*, vol. 654, pp. 120–125, 2016, doi: 10.1016/j.jallcom.2015.09.091.
- [115] K. R. Whittle, M. G. Blackford, R. D. Aughterson, S. Moricca, and G. R. Lumpkin, “Radiation tolerance of $\text{M}_{n+1}\text{AX}_n$ phases, Ti_3AlC_2 and Ti_3SiC_2 ,” vol. 58, pp. 4362–4368, 2010, doi: 10.1016/j.actamat.2010.04.029.
- [116] J. C. Schuster, H. Nowotny, and C. Vaccaro, “The ternary systems: Cr–Al–C, V–Al–C, and Ti–Al–C and the behavior of H-phases (M_2AlC),” *J. Solid State Chem.*, vol. 32, no. 2, pp. 213–219, 1980, doi: 10.1016/0022-4596(80)90569-1.
- [117] A. Azzouz-Rached, H. Rached, I. Ouadha, D. Rached, and A. Reggad, “The Vanadium-doping effect on physical properties of the Zr_2AlC MAX phase compound,” *Mater. Chem. Phys.*, vol. 260, no. September 2020, p. 124189, 2021, doi: 10.1016/j.matchemphys.2020.124189.
- [118] F. L. Meng, Y. C. Zhou, and J. Y. Wang, “Strengthening of Ti_2AlC by substituting Ti with V,” *Scr. Mater.*, vol. 53, no. 12, pp. 1369–1372, 2005, doi: 10.1016/j.scriptamat.2005.08.030.
- [119] S. Jin, T. Su, Q. Hu, and A. Zhou, “Thermal conductivity and electrical transport properties of double-A-layer MAX phase $\text{Mo}_2\text{Ga}_2\text{C}$,” *Mater. Res. Lett.*, vol. 8, no. 4, pp. 158–164, 2020, doi: 10.1080/21663831.2020.1724204.
- [120] M. W. Barsoum, H. Yoo, I. K. Polushina, V. Y. Rud, and Y. V Rud, “The phases with the general formula M_nAX_m ,” *Phys. Rev. B*, vol. 62, no. 15, pp. 194–198, 2000.
- [121] X. Li, H. Cui, and R. Zhang, “Structural , optical , and thermal properties of MAX-phase Cr_2AlB_2 ,” vol. 13, no. 2, pp. 1–9, 2018.
- [122] H. Mebtouche *et al.*, “First-principles calculations of the structural, electronic, mechanical and thermodynamic properties of MAX phase $\text{Mo}_{n+1}\text{GeC}_n$ ($n = 1–3$) compounds,” *Mater. Today Commun.*, vol. 25, no. July, p. 101420, 2020, doi: 10.1016/j.mtcomm.2020.101420.
- [123] G. Surucu, K. Colakoglu, E. Deligoz, and N. Korozlu, “First-Principles Study on the MAX Phases $\text{Ti}_{n+1}\text{GaN}_n$ ($n = 1, 2, \text{ and } 3$),” 2016, doi: 10.1007/s11664-016-4607-1.
- [124] A. Gencer and G. Surucu, “Electronic and lattice dynamical properties of

- Ti₂SiB MAX phase,” *Mater. Res. Express*, vol. 5, no. 7, 2018, doi: 10.1088/2053-1591/aace7f.
- [125] T. H. Scabarozzi *et al.*, “Electrical, thermal, and elastic properties of the MAX-phase Ti₂SC,” *J. Appl. Phys.*, vol. 104, no. 3, pp. 1–6, 2008, doi: 10.1063/1.2959738.
- [126] J. D. Hettinger, S. E. Lofland, P. Finkel, U. States, and J. Palma, “Electrical and Thermal Transport Properties of M₂AlC , where M = Ti , Cr , Nb and V,” no. May 2014.
- [127] P. A. Burr, D. Horlait, and W. E. Lee, “Experimental and DFT investigation of (Cr,Ti)₃AlC₂ MAX phases stability,” *Mater. Res. Lett.*, vol. 5, no. 3, pp. 144–157, May 2017, doi: 10.1080/21663831.2016.1222598.
- [128] T. Heine, “Grand Challenges in Computational Materials Science: From Description to Prediction at all Scales,” *Front. Mater.*, vol. 1, no. July, pp. 1–3, 2014, doi: 10.3389/fmats.2014.00007.
- [129] A. Thore, “A theoretical investigation of Ti_{n+1}AlC_n and Mn₂GaC MAX phases: phase stability and materials properties,” no. 1693, 2012, doi: 10.1016/j.spmi.2012.04.014.
- [130] K. Capelle, “A Bird ’ s-Eye View of Density-Functional Theory,” vol. 36, no. 4, pp. 1318–1343, 2006.
- [131] N. M. Harrison, “An Introduction to Density Functional Theory,” *Technology*, vol. 2, no. 1, pp. 1–26, 1995, doi: 10.1016/S1380-7323(05)80031-7.
- [132] R. F. W. Bader *et al.*, “A very short introduction to density functional theory (DFT),” *J. Chem. Phys.*, vol. 2013, no. 26, pp. 1–26, 2013, doi: 10.1002/9783527610709.
- [133] Y. Asadi and Z. Nourbakhsh, “First principle study of the structural, electronic, vibrational, thermodynamic, linear and nonlinear optical properties of zinc-blende ZnSe and ZnTe semiconductors,” *Comput. Condens. Matter*, vol. 19, p. e00372, Jun. 2019, doi: 10.1016/j.cocom.2019.e00372.
- [134] V. I. Anisimov, “First-Principles Calculations of the Electronic Structure and Spectra of Strongly Correlated Systems: LDA + U Method,” *Spectrosc. Mott Insul. Correl. Met.*, vol. 9, pp. 106–116, 1995, doi: 10.1088/0953-8984/9/4/002.
- [135] J. L. Bretonnet, “Basics of the density functional theory,” *AIMS Mater. Sci.*, vol. 4, no. 6, pp. 1372–1405, 2017, doi: 10.3934/MATERSCI.2017.6.1372.

- [136] L. Gagliardi, D. G. Truhlar, G. L. Manni, R. K. Carlson, C. E. Hoyer, and J. L. Bao, “Multiconfiguration Pair-Density Functional Theory: A New Way To Treat Strongly Correlated Systems,” 2017, doi: 10.1021/acs.accounts.6b00471.
- [137] B. N. Plakhutin and B. N. Plakhutin, “Koopmans’ theorem in the Hartree-Fock method. General formulation Koopmans’ theorem in the Hartree-Fock method. General formulation,” vol. 094101, 2018, doi: 10.1063/1.5019330.
- [138] N. Argaman and G. Makov, “Density functional theory: An introduction,” *Am. J. Phys.*, vol. 68, no. 1, pp. 69–79, 2000, doi: 10.1119/1.19375.
- [139] T. H. Theorem, “Density Functional Theory,” *Comput. Chem.*, pp. 42–48, 1964, doi: 10.1002/0471220655.ch5.
- [140] R. Baer, “VI. The Hohenberg-Kohn density functional theory,” pp. 1–19.
- [141] T. H. Theorem, “Density Functional Theory,” pp. 6–18, 1964.
- [142] A. Non, “The Kohn-Sham method,” pp. 1–36.
- [143] F. Sultana, M. M. Uddin, M. A. Ali, M. M. Hossain, S. H. Naqib, and A. K. M. A. Islam, “First principles study of $M_2\text{InC}$ ($M = \text{Zr, Hf and Ta}$) MAX phases: The effect of M atomic species,” *Results Phys.*, vol. 11, no. August, pp. 869–876, 2018, doi: 10.1016/j.rinp.2018.10.044.
- [144] H. I. Faraoun, F. Z. Abderrahim, and C. Esling, “First principle calculations of MAX ceramics Cr_2GeC , V_2GeC and their substitutional solid solutions,” *Comput. Mater. Sci.*, vol. 74, pp. 40–49, 2013, doi: 10.1016/j.commatsci.2013.03.005.
- [145] N. Kaur, V. Srivastava, and S. A. Dar, “GGA based study on electronic structure and thermoelectric properties of Mn_2PtCo full-Heusler compound,” *Indian J. Phys.*, Jan. 2021, doi: 10.1007/s12648-020-01941-x.
- [146] Z. P. Yin, A. Kutepov, and G. Kotliar, “Correlation-enhanced electron-phonon coupling: Applications of GW and screened hybrid functional to bismuthates, chloronitrides, and other high-Tc superconductors,” *Phys. Rev. X*, vol. 3, no. 2, 2013, doi: 10.1103/PhysRevX.3.021011.
- [147] W. Setyawan and S. Curtarolo, “High-throughput electronic band structure calculations: Challenges and tools,” *Comput. Mater. Sci.*, vol. 49, no. 2, pp. 299–312, 2010, doi: 10.1016/j.commatsci.2010.05.010.
- [148] R. O. Jones, “PhD: Introduction to Density Functional Theory and Exchange-Correlation Energy Functionals,” vol. 31, 2006.

- [149] M. A. L. Marques and E. K. U. Gross, “Time-dependent density functional theory,” *Annu. Rev. Phys. Chem.*, vol. 55, no. November, pp. 427–455, 2004, doi: 10.1146/annurev.physchem.55.091602.094449.
- [150] J. P. Perdew, Y. Wang, and M. Carlo, “r), (2),” vol. 46, no. 20, pp. 947–954, 1992.
- [151] W. Kohn, A. D. Becke, and R. G. Parr, “Density Functional Theory of Electronic Structure,” *J. Phys. Chem*, vol. 100, no. 96, pp. 12974–12980, 1996.
- [152] F. Bloch, “Über die Quantenmechanik der Elektronen in Kristallgittern,” *Zeitschrift für Phys.*, vol. 52, no. 7–8, pp. 555–600, 1929, doi: 10.1007/BF01339455.
- [153] H. J. Monkhorst and J. D. Pack, “Special points for Brillouin-zone integrations* Hendrik,” *Phys. Rev. B*, vol. 13, no. 12, pp. 5188–5192, 1976, doi: 10.1103/physrevb.13.5188.
- [154] K. C. Bimal, “Quasi-Harmonic And Anharmonic Entropies In Transition Metals University of Texas at El Paso ScholarWorks @ UTEP Open Access Theses & Dissertations Quasi-Harmonic And Anharmonic Entropies In Transition Metals University of Texas at El Paso,” no. January 2019, 2020, doi: 10.13140/RG.2.2.31226.24006.
- [155] Charles Kittel, “Introduction to Solid State Physics, 8th Edition,” p. 704, 2004, [Online]. Available: <http://as.wiley.com/WileyCDA/WileyTitle/productCd-EHEP000803.html>.
- [156] G. Vasseur, Y. Fagot-Revurat, B. Kierren, M. Sicot, and D. Malterre, “Effect of symmetry breaking on electronic band structure: Gap opening at the high symmetry points,” *Symmetry (Basel)*, vol. 5, no. 4, pp. 344–354, 2013, doi: 10.3390/sym5040344.
- [157] G. Shukla, “Impossibility of the simple cubic metal in nature,” no. March, 2020.
- [158] T. Lapauw *et al.*, “The double solid solution (Zr, Nb)₂(Al, Sn)C MAX phase: a steric stability approach,” *Sci. Rep.*, vol. 8, no. 1, pp. 1–13, 2018, doi: 10.1038/s41598-018-31271-2.
- [159] E. Kogan and V. U. Nazarov, “Symmetry classification of energy bands in graphene,” *Phys. Rev. B - Condens. Matter Mater. Phys.*, vol. 85, no. 11, pp. 1–5, 2012, doi: 10.1103/PhysRevB.85.115418.

- [160] J. D. Pack and H. J. Monkhorst, “Special points for Brillouin-zone integrations,” *Phys. Rev. B*, vol. 16, no. 4, pp. 1748–1749, Aug. 1977, doi: 10.1103/PhysRevB.16.1748.
- [161] E. Fermi, “Sopra lo Spostamento per Pressione delle Righe Elevate delle Serie Spettrali,” *Nuovo Cim.*, vol. 11, no. 3, pp. 157–166, Mar. 1934, doi: 10.1007/BF02959829.
- [162] P. Schwerdtfeger, “The pseudopotential approximation in electronic structure theory,” *ChemPhysChem*, vol. 12, no. 17, pp. 3143–3155, 2011, doi: 10.1002/cphc.201100387.
- [163] F. Han, *Fundamentals of Semiconductors*. 2011.
- [164] L. Lu *et al.*, “Experimental observation of Weyl points,” *Science (80-.)*, vol. 349, no. 6248, pp. 622–624, 2015, doi: 10.1126/science.aaa9273.
- [165] A. Bouhemadou, “Structural, electronic and elastic properties of MAX phases M_2GaN ($M = Ti, V$ and Cr),” *Solid State Sci.*, vol. 11, no. 11, pp. 1875–1881, 2009, doi: 10.1016/j.solidstatesciences.2009.08.002.
- [166] F. D. Murnaghan, “The Compressibility of Media Under Extreme Pressure,” *Physics (College. Park. Md.)*, vol. 30, pp. 244–247, 1944.
- [167] P. Bhardwaj and S. Singh, “First principle calculation of structural, electronic and elastic properties of rare earth nitride,” *Mater. Sci. Pol.*, vol. 34, no. 4, pp. 715–720, 2016, doi: 10.1515/msp-2016-0123.
- [168] S. Coh, “Electronic Structure Theory: Applications and Geometrical Aspects,” 2011, [Online]. Available: <http://mss3.libraries.rutgers.edu/dlr/showfed.php?pid=rutgers-lib:35951>.
- [169] W. D. Callister, “Materials science and engineering: An introduction (2nd edition),” *Mater. Des.*, vol. 12, no. 1, p. 59, 1991, doi: 10.1016/0261-3069(91)90101-9.
- [170] K. Schwarz, P. Blaha, and G. K. H. Madsen, “Electronic structure calculations of solids using the WIEN2k package for material sciences,” *Comput. Phys. Commun.*, vol. 147, no. 1–2, pp. 71–76, 2002, doi: 10.1016/S0010-4655(02)00206-0.
- [171] D. S. Ginley, *Handbook of Transparent Conductors*. 2011.
- [172] F. Wooten, “Optical Properties of Solids,” 1972.
- [173] A. Debernardi and S. Baroni, “Third-order density-functional perturbation theory: A practical implementation with applications to anharmonic couplings

- in Si,” *Solid State Commun.*, vol. 91, no. 10, pp. 813–816, 1994, doi: 10.1016/0038-1098(94)90654-8.
- [174] A. Urru and A. Dal Corso, “Density functional perturbation theory for lattice dynamics with fully relativistic ultrasoft pseudopotentials: The magnetic case,” *Phys. Rev. B*, vol. 100, no. 4, pp. 1–6, 2019, doi: 10.1103/PhysRevB.100.045115.
- [175] A. Dal and C. Sissa, “Thermo pw : Thermodynamics of crystals.”
- [176] T. H. K. Barron and M. L. Klein, “Second-order elastic constants of a solid under stress,” *Proc. Phys. Soc.*, vol. 85, no. 3, pp. 523–532, 1965, doi: 10.1088/0370-1328/85/3/313.
- [177] G. Grimvall, B. Magyari-Köpe, V. Ozoliņš, and K. A. Persson, “Lattice instabilities in metallic elements,” *Rev. Mod. Phys.*, vol. 84, no. 2, pp. 945–986, Jun. 2012, doi: 10.1103/RevModPhys.84.945.
- [178] D. H. Chung and W. R. Buessem, “The voigt-reuss-hill (vrh) approximation and the elastic moduli of polycrystalline zno, tio₂ (rutile), and α -al₂o₃,” *J. Appl. Phys.*, vol. 39, no. 6, pp. 2777–2782, 1968, doi: 10.1063/1.1656672.
- [179] I. R. Shein and A. L. Ivanovskii, “Elastic properties of superconducting MAX phases from first-principles calculations,” *Phys. Status Solidi Basic Res.*, vol. 248, no. 1, pp. 228–232, 2011, doi: 10.1002/pssb.201046163.
- [180] S. F. Pugh, “XCII. Relations between the elastic moduli and the plastic properties of polycrystalline pure metals,” *London, Edinburgh, Dublin Philos. Mag. J. Sci.*, vol. 45, no. 367, pp. 823–843, 1954, doi: 10.1080/14786440808520496.
- [181] U. F. Ozyar, E. Deligoz, and K. Colakoglu, “Systematic study on the anisotropic elastic properties of tetragonal XYSb (X Combining double low line Ti, Zr, Hf; Y Combining double low line Si, Ge) compounds,” *Solid State Sci.*, vol. 40, pp. 92–100, 2015, doi: 10.1016/j.solidstatesciences.2015.01.001.
- [182] C. S. Man and M. Huang, “A simple explicit formula for the Voigt-Reuss-Hill average of elastic polycrystals with arbitrary crystal and texture symmetries,” *J. Elast.*, vol. 105, no. 1–2, pp. 29–48, 2011, doi: 10.1007/s10659-011-9312-y.
- [183] D. Freyer and W. Voigt, “Crystallization and Phase Stability of CaSO₄ and CaSO₄ - Based Salts,” *Monatshefte fur Chemie*, vol. 134, no. 5, pp. 693–719, 2003, doi: 10.1007/s00706-003-0590-3.
- [184] R. Hill, “The elastic behaviour of a crystalline aggregate,” *Proc. Phys. Soc.*

- Sect. A*, vol. 65, no. 5, pp. 349–354, 1952, doi: 10.1088/0370-1298/65/5/307.
- [185] A. J. G. Lunt *et al.*, “Calculations of single crystal elastic constants for yttria partially stabilised zirconia from powder diffraction data,” *J. Appl. Phys.*, vol. 116, no. 5, 2014, doi: 10.1063/1.4891714.
- [186] D. H. Chung and W. R. Buessem, “The Voigt-Reuss-Hill approximation and elastic moduli of polycrystalline MgO, CaF₂, β-ZnS, ZnSe, and CdTe,” *J. Appl. Phys.*, vol. 38, no. 6, pp. 2535–2540, 1967, doi: 10.1063/1.1709944.
- [187] S. I. Ranganathan and M. Ostojca-Starzewski, “Universal elastic anisotropy index,” *Phys. Rev. Lett.*, vol. 101, no. 5, pp. 3–6, 2008, doi: 10.1103/PhysRevLett.101.055504.
- [188] M. A. S. Ali, M. A. S. Ali, and M. M. Uddin, “Structural, elastic, electronic and optical properties of metastable MAX phase Ti₅SiC₄ compound,” *Indian J. Pure Appl. Phys.*, vol. 54, no. 6, pp. 386–390, 2016, [Online]. Available: <http://op.niscair.res.in/index.php/IJPAP/article/view/4502>.
- [189] J. G. Lee, *Computational Materials Science*, Second Edi. Boca Raton : CRC Press, Taylor & Francis, 2017., 2017.
- [190] R. V. Goldstein, V. A. Gorodtsov, M. A. Komarova, and D. S. Lisovenko, “Extreme values of the shear modulus for hexagonal crystals,” *Scr. Mater.*, vol. 140, pp. 55–58, 2017, doi: 10.1016/j.scriptamat.2017.07.002.
- [191] M. M. Wu, L. Wen, B. Y. Tang, L. M. Peng, and W. J. Ding, “First-principles study of elastic and electronic properties of MgZn₂ and ScZn₂ phases in Mg-Sc-Zn alloy,” *J. Alloys Compd.*, vol. 506, no. 1, pp. 412–417, 2010, doi: 10.1016/j.jallcom.2010.07.018.
- [192] R. Yu, X. F. Zhang, L. L. He, and H. Q. Ye, “Topology of charge density and elastic anisotropy of Ti₃SiC₂ polymorphs,” *J. Mater. Res.*, vol. 20, no. 5, pp. 1180–1185, 2005, doi: 10.1557/JMR.2005.0145.
- [193] A. Candan, S. Akbudak, Ş. Uğur, and G. Uğur, “Theoretical research on structural, electronic, mechanical, lattice dynamical and thermodynamic properties of layered ternary nitrides Ti₂AN (A = Si, Ge and Sn),” *J. Alloys Compd.*, vol. 771, no. August, pp. 664–673, Jan. 2019, doi: 10.1016/j.jallcom.2018.08.286.
- [194] M. A. Ali, M. M. Hossain, N. Jahan, A. K. M. A. Islam, and S. H. Naqib, “Newly synthesized Zr₂AlC, Zr₂(Al_{0.58}Bi_{0.42})C, Zr₂(Al_{0.2}Sn_{0.8})C, and Zr₂(Al_{0.3}Sb_{0.7})C MAX phases: A DFT based first-principles study,”

- Comput. Mater. Sci.*, vol. 131, pp. 139–145, Apr. 2017, doi: 10.1016/j.commatsci.2017.01.048.
- [195] P. K. Jha, “Phonon spectra and vibrational mode instability of MgCNi₃,” *Phys. Rev. B - Condens. Matter Mater. Phys.*, vol. 72, no. 21, pp. 1–6, 2005, doi: 10.1103/PhysRevB.72.214502.
- [196] G. Surucu, K. Colakoglu, E. Deligoz, and N. Korozlu, “First-Principles Study on the MAX Phases Ti_{n+1}GaN_n (n = 1 , 2 , and 3),” 2016, doi: 10.1007/s11664-016-4607-1.
- [197] P. Wachter, M. Filzmoser, and J. Rebizant, “Electronic and elastic properties of the light actinide tellurides,” *Phys. B Condens. Matter*, vol. 293, no. 3–4, pp. 199–223, 2001, doi: 10.1016/S0921-4526(00)00575-5.
- [198] S. Al-Qaisi *et al.*, “Structural, elastic, mechanical and thermodynamic properties of Terbium oxide: First-principles investigations,” *Results Phys.*, vol. 7, pp. 709–714, 2017, doi: 10.1016/j.rinp.2017.01.027.
- [199] X. K. Qian, H. Y. Wu, H. P. Zhu, S. H. Ma, and T. Jiang, “First-principles study of a new higher-order max phase of Ti₅Al₂C₃,” *J. Ceram. Sci. Technol.*, vol. 7, no. 1, pp. 47–52, 2016, doi: 10.4416/JCST2015-00027.
- [200] M. A. Hadi, M. A. Rayhan, S. H. Naqib, A. Chroneos, and A. K. M. A. M. A. Islam, “Structural, elastic, thermal and lattice dynamic properties of new 321 MAX phases,” *Comput. Mater. Sci.*, vol. 170, no. May, p. 109144, 2019, doi: 10.1016/j.commatsci.2019.109144.
- [201] J. Singh, “Boltzmann Transport Theory,” *Mod. Phys. Eng.*, 2004.
- [202] P. Engineering, “Electronic transport properties of thermoelectric Zintl,” 2016.
- [203] and D. J. S. T. C. Collins, R. N. Euwema, “Self-Consistent Orthogonalized-Plane-Wave Calculations,” 1971.
- [204] D. D. Koelling and J. H. Wood, “On the interpolation of eigenvalues and a resultant integration scheme,” *J. Comput. Phys.*, vol. 67, no. 2, pp. 253–262, 1986, doi: 10.1016/0021-9991(86)90261-5.
- [205] E. Assmann, P. Wissgott, J. Kuneš, A. Toschi, P. Blaha, and K. Held, “Woptic: Optical conductivity with Wannier functions and adaptive k-mesh refinement,” *Comput. Phys. Commun.*, vol. 202, pp. 1–11, 2016, doi: 10.1016/j.cpc.2015.12.010.
- [206] J. Scheidemantel, C. Ambrosch-Draxl, T. Thonhauser, V. Badding, and O. Sofo, “Transport coefficients from first-principles calculations,” *Phys. Rev. B -*

- Condens. Matter Mater. Phys.*, vol. 68, no. 12, pp. 1–6, 2003, doi: 10.1103/PhysRevB.68.125210.
- [207] G. K. H. Madsen, J. Carrete, and M. J. Verstraete, “BoltzTraP2, a program for interpolating band structures and calculating semi-classical transport coefficients,” *Comput. Phys. Commun.*, vol. 231, no. January, pp. 140–145, 2018, doi: 10.1016/j.cpc.2018.05.010.
- [208] A. Jayaraman, A. Bhat Kademane, and M. Molli, “DFT Study on the Carrier Concentration and Temperature-Dependent Thermoelectric Properties of Antimony Selenide,” *Indian J. Mater. Sci.*, vol. 2016, pp. 1–7, 2016, doi: 10.1155/2016/7296847.
- [209] C. F. Zinola, “Density functional theory,” *Electrocatal. Comput. Exp. Ind. Asp.*, pp. 117–138, 2010, doi: 10.1201/9781420045451.
- [210] A. Bouhemadou, “Calculated structural and elastic properties of M_2InC ($M = Sc, Ti, V, Zr, Nb, Hf, Ta$),” *Mod. Phys. Lett. B*, vol. 22, no. 22, pp. 2063–2076, 2008, doi: 10.1142/S0217984908016807.
- [211] P. Giannozzi *et al.*, “QUANTUM ESPRESSO: a modular and open-source software project for quantum simulations of materials,” *J. Phys. Condens. Matter*, vol. 21, no. 39, p. 395502, Sep. 2009, doi: 10.1088/0953-8984/21/39/395502.
- [212] C. Malica and A. Dal Corso, “Temperature-dependent atomic B factor: An ab initio calculation,” *Acta Crystallogr. Sect. A Found. Adv.*, vol. 75, no. 1996, pp. 624–632, 2019, doi: 10.1107/S205327331900514X.
- [213] O. Motornyi, M. Raynaud, A. Dal Corso, and N. Vast, “Simulation of electron energy loss spectra with the turboEELS and thermo-pw codes,” *J. Phys. Conf. Ser.*, vol. 1136, no. 1, 2018, doi: 10.1088/1742-6596/1136/1/012008.
- [214] Y. Zhou, J. R. Chelikowsky, X. Gao, and A. Zhou, “On the Preconditioning Function Used in Planewave DFT Calculations and its Generalization,” *Commun. Comput. Phys.*, vol. 18, no. 1, pp. 167–179, 2015, doi: 10.4208/cicp.060314.120215a.
- [215] G. I. Miletić and A. Drašner, “DFT study of the cohesive and structural properties of YNi_5H_x compounds,” *J. Alloys Compd.*, vol. 622, pp. 1041–1048, 2015, doi: 10.1016/j.jallcom.2014.10.106.
- [216] A. Dal Corso, “Elastic constants of beryllium: A first-principles investigation,” *J. Phys. Condens. Matter*, vol. 28, no. 7, 2016, doi:

10.1088/0953-8984/28/7/075401.

- [217] A. D. Corso, “Thermo _ pw a FORTRAN driver for Quantum ESPRESSO routines : progress report 2018,” 2018.
- [218] P. Fuentealba, L. v. Szentpály, H. Stoll, F. X. Fraschio, and H. Preuss, “Pseudopotential calculations including core-valence correlation: Alkali compounds,” *J. Mol. Struct. THEOCHEM*, vol. 93, no. C, pp. 213–219, 1983, doi: 10.1016/0166-1280(83)80104-3.
- [219] G. K. H. Madsen, J. Carrete, and M. J. Verstraete, “BoltzTraP2, a program for interpolating band structures and calculating semi-classical transport coefficients,” *Comput. Phys. Commun.*, vol. 231, pp. 140–145, 2018, doi: 10.1016/j.cpc.2018.05.010.
- [220] G. Pizzi, D. Volja, B. Kozinsky, M. Fornari, and N. Marzari, “thermoelectric and electronic transport properties with,” 2013, pp. 1–19.
- [221] G. K. H. Madsen, J. Carrete, and M. J. Verstraete, “BoltzTraP2 , a program for interpolating band structures and calculating semi-classical transport coefficients ☆,” *Comput. Phys. Commun.*, vol. 231, pp. 140–145, 2018, doi: 10.1016/j.cpc.2018.05.010.
- [222] M. Dahlqvist, A. Thore, and J. Rosen, “Electronic structure, bonding characteristics, and mechanical properties in $(W_{2/3}Sc_{1/3})_2AlC$ and $(W_{2/3}Y_{1/3})_2AlC$ i-MAX phases from first-principles calculations,” *J. Phys. Condens. Matter*, 2018, doi: 10.1088/1361-648X/aacc19.
- [223] J. P. Perdew, K. Burke, and M. Ernzerhof, “Generalized Gradient Approximation Made Simple,” *Phys. Rev. Lett*, vol. 77, no. 18, pp. 3865–3868, 1996.
- [224] N. A. Phatak, S. K. Saxena, Y. Fei, and J. Hu, “Synthesis and structural stability of Ti_2GeC ,” *J. Alloys Compd.*, vol. 474, no. 1–2, pp. 174–179, 2009, doi: 10.1016/j.jallcom.2008.06.073.
- [225] M.- Protocol, C. Cao, Y. Chen, Y. Wu, and E. Deumens, “OPAL : A Multiscale Multicenter Simulation Package Based on MPI-2 Protocol,” *Int. J. Quantum Chem.*, vol. 111, no. May, pp. 4020–4029, 2011, doi: 10.1002/qua.
- [226] W. Sailuam, K. Phacheerak, Atipong bootchanont, I. Fongkaew, and S. Limpijumnong, “Elastic and mechanical properties of hydroxyapatite under pressure: A first-principles investigation,” *Comput. Condens. Matter*, vol. 24, p. e00481, 2020, doi: 10.1016/j.cocom.2020.e00481.

- [227] M. Li *et al.*, “Element Replacement Approach by Reaction with Lewis Acidic Molten Salts to Synthesize Nanolaminated MAX Phases and MXenes,” *J. Am. Chem. Soc.*, vol. 141, no. 11, pp. 4730–4737, 2019, doi: 10.1021/jacs.9b00574.
- [228] A. K. M. A. I. M. A. Rayhan, M. A. Ali, S. H. Naqib, “First-principles Study of Vickers Hardness and Thermodynamic Properties of Ti_3SnC_2 Polymorphs,” *J. Sci. Res.*, vol. 7, no. 3, pp. 53–64, 2015, doi: 10.3329/jsr.v7i3.23182.
- [229] F. Mouhat and F. X. Coudert, “Necessary and sufficient elastic stability conditions in various crystal systems,” *Phys. Rev. B - Condens. Matter Mater. Phys.*, vol. 90, no. 22, pp. 0–3, 2014, doi: 10.1103/PhysRevB.90.224104.

LIST OF PUBLICATIONS

- 1. S. T. Ahams**, A. Shaari, R. Ahmed, N. F. Abdul Pattah, M. C. Idris & B. U. Haq. (2021) Ab initio Study of the Structure, Elastic, and Electronic Properties of $Ti_3(Al_{1-n}Si_n)C_2$ Layered Ternary Compounds. *Scientific Reports*.
<https://doi.org/10.1038/s41598-021-84466-5>
- 2. S. T. Ahams***, A. Shaari, R. Ahmed, N. F. Abdul Pattah, M. C. Idris, B. U Haq. (2021) Theoretical investigation of Zr_2PbC , $(V_{0.25}Zr_{0.75})_2PbC$, $(V_{0.5}Zr_{0.5})_2PbC$, $V_{0.75}Zr_{0.25})_2PbC$, and V_2PbC MAX phases: A DFT based study. *Materials Today Communications* **27** 102397.
<https://doi.org/10.1016/j.mtcomm.2021.102397>.
- 3. S. T. Ahams***, A. Shaari, R. Ahmed, N. F. Abdul Pattah, M. C. Idris, B. U Haq. (2021) A DFT investigation of the structural, mechanical, thermodynamic, and electronic properties of Zr_2PbC MAX phase.
Solid State Technology **64(2)**4863-4874.
- 4. S. T. Ahams**, A. Shaari, R. Ahmed, N. F. Abdul Pattah, M. C. Idris. (2020) Ab-initio Calculations of the Structural and Electronic Properties of Zr_2AC .
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