ALLOYING AND AGING TREATMENT EFFECT ON THE SHAPE MEMORY AND DAMPING PROPERTIES OF Cu-13Al-4Ni ALLOYS

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DEDICATION

To my parents for raising me to believe that everything was possible and to my husband for making everything possible

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ABSTRACT

Copper-based shape memory alloys (SMAs) are gaining attention as materials that require a good damping property in high temperature applications because they exhibit high damping properties during martensitic transformation and have an effective energy dissipation. However, copper-based SMAs such as the ternary alloy Cu-Al-Ni are not easily deformed in the lower temperature martensitic phase which can be attributed to brittleness induced by coarse grain size, high degree of order and elastic anisotropy. Hence, this study aimed to improve the properties of Cu-13Al-4Ni SMAs by addition of fourth alloying element and aging treatment that provides a significant effect on the microstructures and properties of the alloys. In this research, Cu-13Al-4Ni-X alloys with the addition of the fourth additional elements (X=titanium, cobalt or boron) were prepared by casting. The as-cast alloys were then homogenized at 900°C and followed by an aging treatment at 150°C, 200°C and 250°C for 24 hours. The transformation temperatures and microstructure characteristics of Cu-Al-Ni-X SMAs were investigated via differential scanning calorimetry (DSC), scanning electron microscopy (SEM) coupled with energy dispersive spectroscopy (EDS) and x-ray diffraction (XRD). The hardness of these alloys was determined using Vicker's hardness tester. The shape memory effect was determined by compression test using an Instron universal testing machine. The damping property was measured by dynamic mechanical analysis (DMA) technique. The results revealed that the addition of titanium to the Cu-Al-Ni alloy led to the formation of X-phase which consists of intermetallic compounds of NiTi and AlTi₂ that refined the microstructures. On the other hand, addition of cobalt changed the morphologies of the phases with formation of γ_2 phase which improved the ductility of the quaternary alloy. Addition of boron to Cu-Al-Ni alloy led to the formation of secondary phases which also refined the microstructures. Among the three element additions, Co, B and Ti, it was found that the alloy with 0.7% of Co addition at aging treatment of 150°C for 24 hours showed the best shape memory effect with 100% recovery followed by Cu-13Al-4Ni-0.7Ti at aging temperature of 200°C with 97.5% recovery. However, the Cu-13Al-4Ni-0.7Ti at aging temperature of 200°C has the best damping properties with 0.18 internal friction followed by Cu-13Al-4Ni-0.7Co at aging temperature of 150°C with 0.1 internal friction. The findings showed that both of these alloys are good candidates for damping application.

ABSTRAK

Aloi ingatan bentuk (SMA) berasaskan kuprum mendapat perhatian sebagai bahan yang memerlukan sifat redaman yang baik dalam aplikasi suhu tinggi kerana ia menunjukkan sifat redaman yang tinggi semasa transformasi martensitik dan berkesan dalam penyebaran tenaga. Walau bagaimanapun, SMA berasaskan kuprum seperti aloi ternari Cu-Al-Ni tidak mudah berubah bentuk pada fasa martensitik suhu rendah yang dapat dikaitkan dengan kerapuhan disebabkan oleh saiz bijian besar, tahap tertib tinggi dan anisotropi elastik. Oleh itu, kajian ini bertujuan untuk meningkatkan sifat-sifat SMA Cu-13Al-Ni dengan penambahan unsur keempat dan rawatan penuaan yang memberi kesan yang ketara terhadap struktur mikro dan sifat-sifat aloi. Dalam penyelidikan ini, aloi Cu-13Al-4Ni-X dengan penambahan unsur tambahan keempat (X = titanium, kobalt atau boron) disediakan melalui proses tuangan. Aloi tuangan kemudian dihomogenkan pada suhu 900°C dan diikuti dengan rawatan penuaan pada suhu 150°C, 200°C dan 250°C selama 24 jam. Suhu transformasi dan ciri mikrostruktur Cu-13Al-4Ni-X SMA disiasat melalui permeteran kalori pengimbasan kebezaan (DSC), mikroskop elektron pengimbas (SEM), spektrometri serakan tenaga (EDS) dan pembelauan sinar-x (XRD). Kekerasan aloi ini ditentukan dengan menggunakan penguji kekerasan Vicker. Keingatan bentuk telah ditentukan dengan ujian mampatan menggunakan mesin Instron. Sifat redaman diukur dengan analisis mekanikal dinamik (DMA). Hasil kajian menunjukkan bahawa kesan penambahan titanium pada aloi Cu-13Al-4Ni menyebabkan pembentukan fasa-X yang terdiri daripada sebatian intermetalik NiTi dan AlTi₂ yang menghaluskan saiz struktur mikro. Sebaliknya, penambahan kobalt mengubah morfologi fasa dengan pembentukan fasa γ_2 yang meningkatkan kemuluran aloi kuaternari. Penambahan boron kepada Cu-13Al-4Ni menyebabkan pembentukan fasa sekunder yang juga menghaluskan saiz struktur mikro. Di antara tiga penambahan unsur, Co, B dan Ti, didapati bahawa aloi dengan penambahan Co 0.7% yang telah menjalani rawatan penuaan pada suhu 150°C selama 24 jam menunjukkan kesan ingatan bentuk terbaik dengan pemulihan 100 % diikuti oleh Cu-13Al-4Ni-0.7Ti pada suhu penuaan 200°C dengan pemulihan 97.5%. Walau bagaimanapun, Cu-13Al-4Ni-0.7Ti pada suhu penuaan 200°C mempunyai sifat redaman terbaik dengan geseran dalaman 0.18 diikuti oleh Cu-13Al-4Ni-0.7Co pada suhu penuaan 150°C dengan geseran dalaman 0.1. Hasil kajian menunjukkan bahawa kedua-dua aloi ini merupakan calon aloi yang sesuai untuk aplikasi redaman.

TABLE OF CONTENTS

TITLE

D	DECL	ARATION	i
D)EDI	CATION	ii
А	CKN	OWLEDGEMENT	iii
А	BSTI	RACT	iv
А	BSTI	RAK	v
Т	ABL	E OF CONTENTS	vi
L	LIST (OF TABLES	xii
L	LIST (OF FIGURES	xiv
L	LIST (OF ABBREVIATIONS	xxi
L	LIST (OF SYMBOLS	xxii
CHAPTER	1	INTRODUCTION	1
1	.1	Background of the Research	1
1	.2	Problem Statement of the Research	4
1	.3	Objectives of the Research	5
1	.4	Scopes of the Research	6
1	.5	Significance of the Research	6
CHAPTER 2	2	LITERATURE REVIEW	9
2	.1	Introduction	9
2	.2	Development of Shape Memory Alloys	10
2	.3	Martensitic Transformation	11
2	.4	Functional Properties of Shape Memory Alloys	17
		2.4.1 Shape Memory Effect Mechanism	19
		2.4.2 Mechanism of Superelasticity Pseudoelasticity	or 21
		2.4.3 Damping Characteristic (Internal Friction) of Shape Memory Alloys	of 22

3.1	Introd	uction		67
CHAPTER 3	RESE	EARCH M	IETHODOLOGY	67
		2.6.6.3	Damping Characteristics of Cu-Al- Ni SMAs	62
		2.6.6.2	Mechanical Properties of Cu-Al-Ni SMAs	59
		2.6.6.1	Martensitic Structure	55
	2.6.6	Effect of Memory	Aging Treatment on Cu-Al-Ni Shape	55
		E .0.0.0	Ni SMAs	52
		2.6.5.3	Damping Characteristics of Cu-Al-	50
		2.6.5.2	Mechanical Properties of Cu-Al-Ni Shape Memory Alloys	50
		2.6.5.1	Martensitic Structure of Cu-Al-Ni Shape Memory Alloy	46
	2.6.5	Influenc Shape M	e of Alloying Element on Cu-Al-Ni Iemory Alloy	46
		2.6.4.3	Powder Metallurgy	46
		2.6.4.2	Melt Spinning	44
		2.6.4.1	Conventional Casting	43
	2.6.4	Producti Memory	on Methods for Cu-Al-Ni Shape Alloy	43
	2.6.3	Mechani SMAs	ism of Internal Friction of Cu-Al-Ni	42
	2.6.2	Transfor SMAs	mation Temperatures of Cu-Al-Ni	39
	2.6.1	Cu-Al-N Microstr	li Shape Memory Alloys ucture	33
2.6	Cu-Al	-Ni Shape	e Memory Alloys: An Overview	33
	2.5.2	Copper-l	based Shape Memory Alloys	31
	2.5.1	Nickel T	itanium Shape Memory Alloys	29
2.5	Types	of Shape	Memory Alloys (SMAs)	29
		2.4.3.2	Measurement Methods for Internal Friction (IF)	29
		2.4.3.1	Damping Mechanisms of Shape Memory Alloys	27

3.2	Materials	69
3.3	Production of Materials	70
3.4	Homogenisation	72
3.5	Aging Treatment	72
3.6	Sample Preparation for Materials Analysis and Testing	73
	3.6.1 Cutting	73
	3.6.2 Grinding and Polishing	74
	3.6.3 Etching	75
3.7	Materials Characterisation	75
	3.7.1 Inductively Coupled Plasma Mass Spectrometry (ICP-MS)	75
	3.7.2 X-ray Diffraction (XRD)	76
	3.7.3 Optical Microscopy (OM)	77
	3.7.4 Scanning Electron Microscope (SEM) and Energy Dispersive Spectrometry (EDS)	77
3.8	Thermal Analysis	78
	3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined.	Error!
3.9	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 	Error! 79
3.9	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 	Error! 79 79
3.9	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 	Error! 79 79 80
3.9	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 	Error! 79 79 80 80
3.9	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test 	Error! 79 79 80 80 81
3.9 CHAPTER 4	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test RESULTS AND DISCUSSIONS	Error! 79 79 80 80 81 83
3.9 CHAPTER 4 4.1	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test RESULTS AND DISCUSSIONS Introduction	Error! 79 79 80 80 81 83 83
3.9 CHAPTER 4 4.1 4.2	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test RESULTS AND DISCUSSIONS Introduction Microstructures, Phase Transformation Characteristics and Mechanical Properties of Cu-Al-Ni Shape Memory Alloys 	Error! 79 79 80 80 81 83 83 83
3.9 CHAPTER 4 4.1 4.2	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test RESULTS AND DISCUSSIONS Introduction Microstructures, Phase Transformation Characteristics and Mechanical Properties of Cu-Al-Ni Shape Memory Alloys 4.2.1 Microstructure Characteristics 	Error! 79 79 80 80 81 83 83 83 83
3.9 CHAPTER 4 4.1 4.2	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test RESULTS AND DISCUSSIONS Introduction Microstructures, Phase Transformation Characteristics and Mechanical Properties of Cu-Al-Ni Shape Memory Alloys 4.2.1 Microstructure Characteristics 4.2.2 Phase Transformation 	Error! 79 79 80 80 81 83 83 83 83
3.9 CHAPTER 4 4.1 4.2	 3.8.1 Differential Scanning Calorimetry (DSC) Bookmark not defined. Mechanical Properties Test 3.9.1 Hardness 3.9.2 Compression Test and Microhardness Test 3.9.3 Shape Memory Effect Test 3.9.4 Damping Test RESULTS AND DISCUSSIONS Introduction Microstructures, Phase Transformation Characteristics and Mechanical Properties of Cu-Al-Ni Shape Memory Alloys 4.2.1 Microstructure Characteristics 4.2.2 Phase Transformation 4.2.3 Mechanical properties 	Error! 79 79 80 81 83 83 83 83 83 83 83 83 83 83 83 83 83

		4.2.3.2	Shape Memory Effect Test	90
		4.2.3.3	Damping Properties Test	92
4.3	Micro and M Memo	ostructures, Iechanical ory Alloys	, Phase Transformation Characteristics Properties of Cu-13Al-4Ni-xTi Shape	93
	4.3.1	Microstr	uctures Characterization	94
	4.3.2	Phase Tr	ansformation	100
	4.3.3	Mechani	cal Properties	103
		4.3.3.1	Compression and Microhardness Test	103
		4.3.3.2	Shape Memory Effect	105
		4.3.3.3	Damping Properties	107
4.4	Micro and M Memo	ostructures, Iechanical	, Phase Transformation Characteristics Properties of Cu-13Al-4Ni-Co Shape	108
	<i>4 4</i> 1	Microstr	ucture Characteristics	100
	442	Phase Tr	ansformation	113
	л.т.2 Л Л З	Mechani	cal Properties of Cu 13A1 4Ni Co	115
	н.н. 3	Shape M	lemory Alloys	115
		4.4.3.1	4.4.3.1 Compression and Microhardness Test results	116
		4.4.3.2	Shape Memory Effect Tesy	118
		4.4.3.3	Damping Properties Test	120
4.5	Micro and N Memo	ostructures, Aechanical	Phase Transformation Characteristics Properties of Cu-13Al-4Ni-B Shape	121
	4.5.1	Microstr	uctures Characterization	121
	4.5.2	Phase Tr	ransformation	126
	4.5.3	Mechani	cal Properties of Cu-Al-Ni-XB SMA	128
		4.5.3.1	Compression and Microhardness Test results	128
		4.5.3.2	Shape Memory Effect Test	130
		4.5.3.3	Damping Properties	132
4.6	Micro 13Al-	ostructures 4Ni Shape	and Mechanical Properties of Cu- Memory Alloy after Aging Treatment	

133

	4.6.1	Microstru	ucture Characterisation	133
	4.6.2	Mechani	cal Properties	136
		4.6.2.1	Compression and Microhardness Test	136
		4.6.2.2	Shape Memory Effect Test	139
		4.6.2.3	Damping Properties Test	141
4.7	Micro 13Al-4	structures 4Ni-0.7Ti	and Mechanical Properties of Cu- Alloys After Aging Treatment	144
	4.7.1	Microstru	ucture Characterization	144
	4.7.2	Mechani	cal Properties	146
		4.7.2.1	Compression and Microhardness Test	147
		4.7.2.2	Shape Memory Effect Test	149
		4.7.2.3	Damping Properties Test	151
4.8	Micro 13Al-4	structures 4Ni-0.7Co	and Mechanical Properties of Cu- Shape Memory Alloys After Aging	154
	1 eau	Microstr	usture Characterization	154
	4.8.1	Microstro		154
	4.8.2	Mechani	cal Properties	157
		4.8.2.1	Compression and Microhardness Test	157
		4.8.2.2	Shape Memory Effect Test	159
		4.8.2.3	Damping Properties Test	161
4.9	Micro 13Al-4	structures 4Ni-0.7B	and Mechanical Properties of Cu- Shape Memory Alloys After Aging	
	Treatr	nent		163
	4.9.1	Microstru	ucture Characterization	163
	4.9.2	Mechani	cal Properties	166
		4.9.2.1	Compression and Microhardness Test	166
		4.9.2.2	Shape Memory Effect Test	168
		4.9.2.3	Damping Properties Test	169
4.10	Summ	nary		171
	4.10.1	Microstru	ucture Characterization	172

	4.10.2 Microhardness of Cu-Al-Ni-X shape memory alloys 172	
	4.10.3 Shape memory behavior of Cu-Al-Ni-X shape memory alloys	173
	4.10.4 Damping properties of Cu-Al-Ni-X shape memory alloys	176
CHAPTER 5 FOR FUTURE V	CONCLUSIONS AND RECOMMENDATIONS WORK	179
5.1	Conclusions	179
5.2	Recommendations for Future Work	180
REFERENCES		181
LIST OF PUBLI	CATIONS	201

LIST OF TABLES

TABLE NO.	TITLE	PAGE
Table 2.1	Relation between Ni Concentration and Transformation Temperature of Cu-Al-Ni alloy [54]	41
Table 2.2	Summary of the Mechanism of Internal Friction of Cu-Al- Ni SMAs at various composition.	42
Table 3.1	Chemical Compositions of copper-based shape memory alloys (wt %)	69
Table 4.1	Transformation Temperatures of Cu-13Al-4Ni shape memory alloy	89
Table 4.2	Transformation temperatures of Cu-13Al-4Ni shape memory alloy with addition of 0.7wt%Ti and 1.3wt%Ti.	103
Table 4.3	The fracture stress, strain and microhardness of Cu-13Al-4Ni-0.7Ti and Cu-13Al-4Ni-1.3Ti alloys	104
Table 4.4	Transformation Temperature of Cu-13Al-4Ni-0.7Co and Cu-13Al-4Ni-1.3Co shape memory alloys	114
Table 4.5	Fracture stress, strain and microhardness of Cu-13Al-4Ni- 0.7Co and Cu-13Al-4Ni-1.3Co alloys	117
Table 4.6	Transformation temperatures of Cu-13Al-4Ni shape memory alloy with addition of 0.7wt%B and 1.3wt%B additions	128
Table 4.7	Fracture stress, strain and microhardness of Cu-13Al-4Ni- 0.7B and Cu-13Al-4Ni-1.3B alloys.	130
Table 4.8	Fracture stress and strain and microhardness of Cu-13Al- 4Ni SMA with different aging temperatures	138
Table 4.9	The shape memory effects of Cu-13Al-4Ni at different aging temperatures	140
Table 4.10	Fracture stress and strain and microhardness of Cu-13Al- 4Ni-0.7Ti with varying aging temperatures	148
Table 4.11	The shape memory effect of Cu-13Al-4Ni-0.7Ti at different aging temperatures	150
Table 4.12	Fracture stress and strain and microhardness of Cu-13Al- 4Ni-0.7Co at different aging temperatures	159
Table 4.13	The shape memory effect and internal friction of Cu-13Al- 4Ni-0.7Co at different aging temperatures	161

Table 4.14	Fracture stress, strain and microhardness of Cu-13Al-4Ni- 0.7B shape memory alloys at different aging temperatures	167
Table 4.15	The shape memory effect and internal friction of Cu-13Al- 4Ni-0.7B at different aging temperature s	169

LIST OF FIGURES

FIGURE NO	. TITLE	PAGE
Figure 2.1	Schematic drawing of structure alteration correlated with martensitic transformation: (a) single crystal original parent, (b) surface relief owning to transformation, (c) change in direction of pre-scratched straight line upon transformation of martensite [25].	12
Figure 2.2	Lattice deformation, and complementary slip shear and twinning shear [54].	13
Figure 2.3	Illustration of temperature-induced phase transformation of SMA without mechanical loading [57].	15
Figure 2.4	Schematic of the shape memory effect of an SMA illustrating the detwinning of the material under an applied stress [60].	16
Figure 2.5	Schematic of the shape memory effect of an SMA demonstrating the unloading and subsequent heating to austenite under no load condition [60].	16
Figure 2.6	Shape memory effect. At the end of a mechanical loading- unloading path (ABC) conducted at a persistent low temperature, the material presents residual deformation (AC), which can be recovered through a thermal cycle (CDA) [62].	18
Figure 2.7	Superelastic effect. At a persistent high temperature the material is capable to undergo large deformations with zero final permanent strain [62].	19
Figure 2.8	Mechanism of shape memory effect shown by (a) to (f) [58]	20
Figure 2.9	Mechanism of superelasticity shown by (a) to (e) [58].	22
Figure 2.10	The relative dynamic modulus and internal friction as function of temperature during the complete phase transformation: (a) cooling and (b) heating [68].	24
Figure 2.11	Schematic of the three contributions to IF during transformation of martensitic phase transformation: the transitory term $IFTr$, the phase transition, term $IFPT$ and the intrinsic term $IFInt$ [63]	26
Figure 2.12	Schematic of the SMA damper for a stay-cable bridge [91]	30

Figure 2.13	Stacking sequence of the planes in 2H and 18R martensites [96]	32
Figure 2.14	Six sets of planes of martensite shaped from DO ₃ type parent phase [96]	33
Figure 2.15	Phase diagram of the Cu-Al-Ni ternary alloy system with 3 wt% Ni [97]	35
Figure 2.16	Schematic phase diagram of Cu-Al-Ni alloy in temperature- stress coordinates [25, 99]	36
Figure 2.17	SEM micrograph illustating the martensite microstructure of the specimens: (a) typical zig-zag self-accommodating groups of $\beta 1$ ' martensite variants, (b) coarse variants of $\gamma 1$ ' martensite and (c) $\beta 1$ ' martensite self- accommodating variants [10]	37
Figure 2.18	SEM micrographs of (a) Cu-13.0Al-4Ni (b) Cu-13.5Al- 4Ni, (c) Cu-13.7Al-4Ni, and (d) Cu-14.0Al-4Ni SMAs [81]	38
Figure 2.19	Relationship between Al content and transformation temperatures in Cu-Al-Ni alloys [54, 115]	41
Figure 2.20	(a) Experimental set-up of a free jet melt spinner and (b) Microstructure achieved in the rapidly solidified Cu-Al-Ni alloy [96, 131]	45
Figure 2.21	Optical micrograph of (a) Cu-13wt.%Al-5wt.%Fe, (b) Cu- 13wt.%Al-5wt.%Fe-1wt.%Co and (c) Cu-13wt%Al- 5wt%Fe-2wt.%Co alloys [139]	48
Figure 2.22	Optical micrographs showing the recrystallized microstructures of (a) Cu-13.4Al-3.8Ni, (b) Cu-13.2Al-3.04Ni-0.36Ti, (c) Cu-13.0Al-2.97Ni-0.36Ti-0.22Mn, (d) Cu-13.4Al-3.05Ni-0.24Ti-0.63Zr [127]	49
Figure 2.23	Typical stress vs strain relationship [152]	51
Figure 2.24	Recovery characteristic curves of the Cu-13Al-4Ni with different B addition under prestrains of 10% [142].	52
Figure 2.25	(a) IF peaks of different Al composition. (b) The tan δ of the IF peaks and the ΔH values of the martensitic transformation [81].	53
Figure 2.26	Damping property of the Cu-Al-Ni SMAs refined by $1.0wt.\%$ Ti and various weight percent of $Cu_{51}Zr_{14}$ inoculant [156]	54
Figure 2.27	SEM image of aged Cu-14.1wt.%Al-4.0 wt.%Ni alloy at (a) 200 °C and (b) 300 °C [107].	56

Figure 2.28	Optical micrographs of Cu-13.4wt.%Al-4.0wt.%Ni (a) as- grown (b) aged sample at 250 °C conditions [31].	57
Figure 2.29	Microstructure of Cu-11.6Al-3.9Ni-2.5Mn after aging at different temperatures and time: (a) 300 °C for 20 min, (b) 300 °C for 2 hours, (c) 300 °C for 24 hours, (d) 400 °C for 24 hours and (e) 500 °C for 24 hours [162].	58
Figure 3.1	Summary of the Research Methodology	68
Figure 3.2	Procedures of casting process a) Materials to melt in the silicon carbide crucible (b) Stirring the molten materials (c) Pouring the molten materials into the mold. (d) as-cast ingot of the alloy	71
Figure 3.3	Temperature-time profile of homogenization and aging treatment	73
Figure 3.4	Specimens after wire cut: (a) Compression test and microstructural observation specimen (b) dynamic mechanical analysis DMA test specimen	74
Figure 3.5	Schematic of inductively coupled plasma mass spectrometry (ICP-MS) [169]	76
Figure 3.6	X-ray Diffractometer (XRD)	77
Figure 3.7	(a) Perkin Elmer Pyric 6 Differential Scanning Calorimeter and (b) schematic drawing of principle of DSC	79
Figure 3.8	Compression test (a) Universal testing machine (b) Close- up view of the test set-up.	80
Figure 3.9	Procedures of shape memory test.	81
Figure 3.10	Dynamic Mechanical Analyser (a) the equipment (b) the illustration of the DMA dual cantilever clamp measuring system	82
Figure 4.1	Microstructure of the Cu-13Al-4Ni shape memory alloy; (a) Optical Micrograph; (b) Scanning electron micrograph and (c) Optical micrograph (low magnification)	85
Figure 4.2	(a) VPSEM micrograph of Cu-13Al-4Ni alloy and corresponding EDS analysis of (b) lath longitudinal martensite ($\gamma 1'$) (Spectrum 1) and (c) self accomodation martensite ($\beta 1'$) (Spectrum 2)	86
Figure 4.3	X-ray diffraction pattern of Cu-13Al-4Ni shape memory alloy.	87
Figure 4.4	Transformation Temperatures of Cu-13Al-4Ni shape memory alloy (a) DSC curve (b) Heating and (c) Cooling	88
Figure 4.5	Compressive Stress- Strain Curve of Cu-13Al-4Ni alloys	90

Figure 4.6	Recovery characteristic curves of the Cu-13Al-4Ni SMA under prestrains of 10%.	91
Figure 4.7	The damping capacity of Cu-13Al-4Ni SMAs	93
Figure 4.8	Cu-13Al-4Ni with Ti addition (a) optical micrograph, 0.7 wt. %Ti (b) VPSEM micrograph, 0.7 wt. %Ti, (c) optical micrograph, 1.3 wt.% Ti and (d) VPSEM micrograph, 1.3 wt.% Ti	95
Figure 4.9	VPSEM micrograph of Cu-13Al-4Ni-1.3Ti and corresponding EDS analysis results of (b) X-phase along grain boundary (Spectrum 1) and (c) matrix alloy (Spectrum 2) (d) X-phase (Spectrum 3)	97
Figure 4.10	X-ray diffraction patterns of (a) Cu-13Al-4Ni-0.7Ti and (b) Cu-13Al-4Ni-1.3Ti	98
Figure 4.11	Elemental distribution map showing the distribution of various elements in the Cu-13Al-4Ni-1.3Ti shape memory alloy (a) SEM micrograph of the alloy, (b) element Al, (c) element Cu, (d) element Ni and (e) element Ti.	100
Figure 4.12	DSC curve of Cu-13Al-4Ni-0.7Co and Cu-13Al-4Ni-1.3Co shape memory alloy: (a) Heating curve and (b) Cooling curve	102
Figure 4.13	Compressive Stress-Strain Curve of Cu-13Al-4Ni-0.7Ti and Cu-13Al-4Ni-1.3Ti alloys.	104
Figure 4.14	Strain Recovery Characteristics under pre-strain 10% for (a) Cu-13Al-4Ni-0.7Ti (b) Cu-13Al-4Ni-1.3Ti shape memory alloys	106
Figure 4.15	Internal friction of Cu-13Al-4Ni-0.7Ti and Cu-13Al-4Ni- 1.3Ti	108
Figure 4.16	Cu-13Al04Ni with Co addition (a) optical micrograph, 0.7wt. %Co (b) VPSEM micrograph, 0.7wt.%Co, (c) optical micrograph, 1.3wt.% Co and (d) VPSEM micrograph, 1.3wt. %Co.	110
Figure 4.17	(a) VPSEM micrograph of Cu-13Al-4Ni-1.3Co shape memory alloy and corresponding EDS analysis results of(b) spectrum 1 and (c) spectrum 2	111
Figure 4.18	XRD patterns of the specimens (a) Cu-13Al-4Ni-0.7Co and (b) Cu-13Al-4Ni-1.3Co	112
Figure 4.19	Elemental distribution map showing the distribution of various elements in the Cu-13Al-4Ni-0.7Co shape memory alloy (a) SEM micrograph, (b) Cu, (c) Al, (d) Ni and (e) Co elements	113

Figure 4.20	DSC curve of Cu-13Al-4Ni-0.7Co and Cu-13Al-4Ni-1.3Co shape memory alloy: (a) Heating curve and (b) Cooling curve	115
Figure 4.21	Compressive Stress-Strain Curve of Cu-13Al-4Ni-0.7Co and Cu-13Al-Ni-1.3Ti Shape Memory Alloys	117
Figure 4.22	Strain Recovery Characteristics under pre-strain 10% for (a) Cu-13Al-4Ni-0.7wt%Co (b) Cu-Al-Ni-1.3wt%Co	119
Figure 4.23	Internal friction of Cu-13Al-4Ni-0.7Co and Cu-13Al-4Ni- 1.3Co shape memory alloys	121
Figure 4.24	Cu-13Al-4Ni with B addition (a) optical micrograph, 0.7wt%B (b) VPSEM micrograph, 0.7wt%B, (c) optical micrograph, 1.3wt%B and (d) VPSEM micrograph, 1.3wt%B.	123
Figure 4.25	VPSEM micrograph of Cu-13Al-4Ni-1.3B and corresponding EDS analysis results of Precipitates (b) Spectrum 1, (c) Spectrum 2 and matrix alloy (d) Spectrum 3	124
Figure 4.26	XRD patterns for (a) Cu-13Al-4Ni-0.7B and (b) Cu-13Al-4Ni-1.3B	125
Figure 4.27	Elemental distribution map showing the distribution of various elements in the Cu-13Al-4Ni-1.3B shape memory alloy (a) VPSEM micrograph of the alloy (b) element Al, (c) element Cu, (d) element Ni and (e) element B.	126
Figure 4.28	DSC curve of Cu-13Al-4-0.7B and Cu-13Al-4Ni-1.3B (a) Heating curve and (b) Cooling curve (b) Cooling curve	127
Figure 4.29	Compressive stress-strain curves for Cu-13Al-4Ni-0.7B and Cu-13Al-4Ni-1.3B alloys	129
Figure 4.30	Strain Recovery Characteristics under pre-strain 10% for (a) Cu-13Al-4Ni-0.7B (b) Cu-13Al-4Ni-1.3B	131
Figure 4.31	Internal friction of Cu-13Al-4Ni-0.7B and Cu-13Al-4Ni-1.3B shape memory alloys.	132
Figure 4.32	Optical and VPSEM micrographs of Cu-13Al-4Ni shape memory alloy after aging treatment at (a,b) 150°C, (c,d) 200 °C and (e,f) 250 °C	135
Figure 4.33	XRD results of Cu-13Al-4Ni alloys at different aging temperatures.	136
Figure 4.34	Compressive stress-strain curve for aged Cu-13Al-4Ni shape memory alloys after aging treatment.	138

Figure 4.35	Strain Recovery Characteristics under 10% pre-strain for Cu-13Al-4Ni SMA at different aging temperatures	140
Figure 4.36	(a) Damping capacity of the Cu-13Al-4Ni SMA aged at different temperatures (b) relationship between internal friction and aging temperature of aged Cu-Al-Ni alloy.	143
Figure 4.37	Optical and VPSEM micrograpsh for aged Cu-13Al-4Ni- 0.7Ti alloys after aging treatment at (a,b) 150°C, (c,d) 200 °C and (e,f) 250 °C	145
Figure 4.38	X-ray diffraction pattern for aged Cu-13Al-4Ni-0.7Ti alloys underwent aging treatment at different temperatures	146
Figure 4.39	Compressive stress-strain curve for aged Cu-13Al-4Ni- 0.7Ti shape memory alloys at different aging temperatures.	148
Figure 4.40	Strain Recovery Characteristics under pre-strain 10% for Cu-13Al-4Ni-0.7Ti SMA at different aging temperatures	150
Figure 4.41	(a) Damping capacity of the Cu-13Al-4Ni-0.7Ti alloy aged at different temperatures (b) relationship between internal friction and aging temperature of Cu-13Al-4Ni-0.7Ti alloy	153
Figure 4.42	Optical and VPSEM micrographs for aged Cu-13Al-4Ni- 0.7Co SMA at (a,b) 150°C, (c,d) 200 °C and (e,f) 250 °C	155
Figure 4.43	X-ray diffraction patterns for aged Cu-13Al-4Ni-0.7Co alloys at different aging temperatures	156
Figure 4.44	Compressive stress-strain curve for Cu-13Al-4Ni-0.7Co alloys after aging treatment	158
Figure 4.45	Strain Recovery Characteristics under pre-strain 10% for Cu-13Al-4Ni-0.7Co SMA at different aging temperatures	160
Figure 4.46	(a) Damping capacity of the Cu-13Al-4Ni-0.7Co SMA aged at different aging temperatures (b) relationship between internal friction and aging temperature of Cu-13Al-4Ni-0.7Co alloy after aging treatment	162
Figure 4.47	Optical and VPSEM micrographs for Cu-13Al-4Ni-0.7B shape memory alloy at aging temperatures: (a,b) 150°C, (c,d) 200 °C and (e,f) 250 °C	164
Figure 4.48	Analysis of Cu-13Al-4Ni-0.7B alloy (a) VPSEM micrograph (b) EDS analysis (Spectrum 1) and (c) EDS analysis (Spectrum 2)	165
Figure 4.49	X-ray diffraction pattern of aged Cu-Al-Ni-0.7wt%B alloy after aging treatment	166

Figure 4.50	Compressive stress-strain curve for aged Cu-13Al-4Ni- 0.7B SMAs	167
Figure 4.51	Strain Recovery Characteristics under pre-strain 10% for Cu-13Al-4Ni-0.7B shape memory alloy after aging at different aging temperatures.	168
Figure 4.52	 (a) Damping capacity of the Cu-13Al-4Ni-0.7B SMA aged at different aging temperatures (b) relationship between internal friction and aging temperature of Cu-13Al-4Ni- 0.7B alloy after aging treatment 	171
Figure 4.53	Hardness of Cu-13Al-Ni-X (X=Ti, Co or B) alloys at different aging temperature: a) without aging b) 150 $^{\circ}$ C c) 200 $^{\circ}$ C and d) 250 $^{\circ}$ C	173
Figure 4.54	Shape recovery characteristics under pre-strain for Cu- 13Al-4Ni-0.7X, (X=Ti, Co and B) shape memory alloy: a) without aging b) aging at 150 °C c) aging at 200 °C and d) aging at 250 °C temperature	175
Figure 4.54	Shape recovery characteristics under pre-strain for Cu- 13Al-4Ni-0.7X, (X=Ti, Co and B) shape memory alloy: a)without aging b) aging at 150 °C c) aging at 200 °C and d) aging at 250 °C temperature (continue)	176
Figure 4.55	Internal friction peak of Cu-13Al-4Ni-0.7X, (X=Ti, Co or B) shape memory alloy at different condition: a) without aging b) at aging at 150 °C c) aging at 200 °C and d) aging at 250°C temperature	178

LIST OF ABBREVIATIONS

Cu-Al-Ni	-	Copper Aluminium Nickel	
DMA	-	Dynamic Mechanical Analysis	
DSC	-	Differential scanning calorimetry	
EDS	-	Energy dispersive spectroscopy	
ICP-MS	-	Inductive coupled plasma mass spectroscopy	
IF	-	Internal Friction	
ОМ	-	Optical microscope	
SEM	-	Scanning electron microscopy	
SMAs	-	Shape memory alloys	
SME	-	Shape memory effect	
VPSEM	-	Variable pressure scanning electron micoscopy	
XRD	-	X-ray diffraction	

LIST OF SYMBOLS

δ	-	delta
λ	-	wavelength
R	-	Recovery ratio
ε_{SME}	-	Shape memory effect
M_s	-	Martensite start temperature
M_{f}	-	Martensite finish temperature
As	-	Austenite start temperature
A_{f}	-	Austenite finish temperature
d	-	Spacing distance
G	-	Gibbs free energy
Q^{-1}	-	factor
ΔW	-	Energy absorption
W	-	Maximum elastic stored energy
arphi	-	Specific damping capacity
ω	-	Oscillation frequency
Т	-	temperature
T ₀	-	Equilibrium temperature
σ_0	-	oscillation amplitude
V	-	Volume fraction
°C	-	Centrigrade degree
Wt%	-	Weight percentage
θ	-	Bragg's angle
Β, α	-	Lattice angles
Hv	-	Vicker's hardness
f	-	Frequency
σ	-	stress
3	-	Strain
IF _{Int}	-	Intrinsic internal friction
IF _m	-	intrinsic damping of martensite
IF_{PT}	-	Phase transformation internal friction

IF _{Tr}	-	Transient Internal Friction
IF _a	-	Intrinsic damping of austenite
GB	-	Grain boundary

CHAPTER 1

INTRODUCTION

1.1 Background of the Research

Environmental issues related to mechanical, structural and noise vibrations from seismic events such as excavation, construction, mining, and exploration activities have gained considerable attention [1]. Mild discomfort to human and general machinery inefficiencies are some of the detrimental effects of the vibration sources generated from the seismic events that may lead to structural failure, loss of lives, properties and investments [2, 3]. These unwanted vibrations maybe eliminated by selecting materials with good damping properties for engineering structures and systems [4, 5]. Research on these materials has created tremendous interest among scientists and engineers.

Shape memory alloys (SMAs) are materials which capable of regaining their original shape after a large inelastic deformation. This is due to the reversible transformation between two different phases namely martensite and austenite. The martensite phase is highly twinned crystalline structure with high transformation strain and it is stable in low energy level, and the austenite phase is stable in high energy level with low strain. Most SMAs show two unique properties which are known as shape memory effect. The SMAs has ability to recover its original structure upon heating and pseudoelastic effect by restoring the stress-induced deformation upon releasing the applied loads [6].

Copper based SMAs are notable for their easy production and application in addition to their lower price compared to NiTi alloys. Among Cu based SMAs, Cu-Al-Ni alloys have a higher themal stability. Therefore, Cu-Al-Ni alloys are being developed for high temperature applications due to their potential to be used as sensors and actuators at temperature around 200 °C. Also, it has emerged as a potential material for high damping material. Composition of aluminium and content of nickel is strongly affects the shape memory properties of Cu-Al-Ni SMAs. The shape memory properties of the alloys can be accomplished when the Al content is about 11wt.% -13wt.%and content of Ni is about 3-5wt.% [7]. Some other factors that influence the shape memory properties are martensite structure, composition and transformation temperatues [7, 8]. Various kinds of martensite such as β'_1 (18R), γ'_1 (2H) or α'_1 (6R) can be formed by transformation of β_1 (DO3) parent phase in Cu-Al-Ni SMAs, in which formation of these martensite phases depends on applied stress and temperature. 18R (β'_1) and 2H (γ'_1) are the two martensite structures that normally formed in Cu-Al-Ni SMAs, in which the content of Al and Ni as well as heat treatment affect their formation [8, 9]. Martensite formed on cooling is β'_1 in a low Al alloy and is γ'_1 in a high Al alloy, In alloys with composition with compositions near the phase boundary between these two martensites, both can coexist [10]. β'_1 and γ'_1 are two martensite structures that typically obtained in Cu-Al-Ni SMAs, in which their formation are strongly affected by content of Al and Ni and the requied characteristic of the application determines the content of combination of both Al and Ni.

Nevertheless, the single crystal in structure of the Cu-Al-Ni SMA is ductile however polycrystalline Cu-Al-Ni SMA is generally brittle and low recovery strain which has limitation in practical application [11, 12]. In fact, existence of β phase in Cu-Al-Ni alloy results in intergranular fracture as a result of high elastic anisotropy and large size grains. To solve this issue, several approaches such as grain refining by addition of alloying elements, precipitation hardening or heat treatment can be utilized to enhance the mechanical properties and improving its ductility as well as workability [13-16].

Addition of alloying elements such as Nb, V, B, Ti and Mn or various concentrations of Al and Ni to Cu based SMAs is one of the approaches to increase the ductility and refining these alloys [17-20]. Once the grain size is refined, the mode of the fracture alters from intergranular to transgranular with a ductile fracture mode during impact test [20]. Ti has a significant effect in decreasing grain size in Cu-Al-Ni ternary alloy [13, 19]. The grain refinement due to the Ti addition is attributed to

the development of finely dispersed Ti-rich precipitates of X-phase, which hinder the grain growth [21]. It has been proved that the grain refinement enhances the mechanical properties.

Martensitic phase transformation from austenite to martensite is crucial to achieve the superelasticity and shape memory effects. Therefore, the shape memory characteristics such as thermal hysteresis, transformation temperatures and recoverable strain depend on aging effects for both phases [7]. Based on the Canbay et.al [22] findings, during the different annealing treatment of Cu-Al-Ni SMAs, β_1 ' martensite formed as the main phase, in which based on the applied condition, various crystalline size of this phase is formed. The relative steadiness of the both martensite and austenite in SMAs is a function of heat treatments, as the Cu based SMAs are sensitive to heat treatments [22, 23].

Damping is the transformation of mechanical energy of a vibrating structure into thermal energy. Its mechanism is complex with the combination of structural and material damping. Material damping depend on many factors which include type of materials, stress amplitude, internal forces, quality of surfaces and temperature. There are many engineering materials that have good damping property and among them are shape memory alloys (SMAs) [24]. SMAs are well known alloys which have pseudoelastic and shape memory properties but they are also known to have excellent damping property which makes them desirable for the vibration control devices design [25]. High damping capacity of SMAs in austenitic and martensitic states is due to stress induced martensitic transformation and stress-induced martensite variant orientation respectively [26]. SMAs with high damping capacity have many applications in different industries.

One of the important functional properties of SMAs is desired damping capacity of these alloys. Its mechanism is complex with the combination of structural and material damping. It is because a large fraction of mechanical energy is dissipated in the internal interfaces between the phases during their formation and motion, which is due to shape memory effect and superlasticity deformation of SMAs [27, 28]. Martensite variant interfaces and twin boundaries as hysteresis movement of interfaces leads to high damping capability of the thermoelastic martensite phase. Kustov et. al [29] establish a high damping capacity in Cu-Al-Ni alloys which is because of the dislocations and their relations with other lattice effects. Hence, Cu-Al-Ni alloys is possible to be used as shape memory energy absorbers which can minimize structural damage during earthquake.

Numerous studies reported the influence of aging on the damping properties of Cu based SMAs [30-33]. Cu-Al-Ni alloys have tendency to ageing which leads to formation of precipitate and consecutive martensitic transitions from β'_1 to γ'_1 . Hereafter, their damping properties are also likely to alter significantly with aging. Suresh and Ramamurty [30] examined the influence of aging on the damping capacity of Cu-Al-Ni and found decreasing in the internal friction (IF) value of this alloy with aging treatment as a result of the precipitation formation (γ_2) at certain aging temperature. Shivaramu [32] reported that the damping limit of the Cu-Al-Be-Mn SMAs decreased with increasing the aging temperature and time due to the formation of precipitates which can acts as barrier for the movement of the interfaces. Hence, it is shown in numerous studies that damping capacity is strongly depends on the influence of addition elements and heat treatment process. Therefore, the aims of the present study are to elucidate the influence of various addition elements and heat treatment on the microstructure and properties such as hardness, shape memory behaviour and damping characteristics of Cu-Al-Ni-X alloys.

1.2 Problem Statement of the Research

Shape memory alloys have been drawing attention in current times for the vibration control systems design, seismic resistant design and retrofit of structure [4, 24, 34] due to their high energy absorption and dissipation capacities that results in high damping capacity. The well-known and established SMAs, Ni-Ti alloys have high superelasticity, large recoverable strain and high fatigue life which satisfy actual application requirements [35, 36]. Nevertheless, poor cold workability and high cost of material is the main hindrance against utilizing Ni-Ti alloys for such applications [37, 38]. In addition, the austenite-finish temperature (A_f) of Ni-Ti alloy is around

0°C. It is challenging to obtain 20°C by adjusting the composition and heat treatments. Hence, their outdoor engineering application is restricted by relatively high range of working temperature in practical limits. Alternatively, Cu-Al-Ni SMAs are considered as potential materials for damping application due to their martensitic transformation temperature are adjustable to above room temperature, low production cost, easy manufacturing, excellent shape memory function and high damping capacity. However, high brittleness and unfavourable mechanical properties due to coarse grains limit their further development and applications in certain fields. The undesirable properties may be improved by adding alloying elements followed by aging treatments. Therefore, addition of fourth element in the Cu-Al-Ni SMA together with aging treatment became the focal point of this research in order to achieve the optimum performance to the shape memory recovery and damping capacity.

1.3 Objectives of the Research

The main objective of the research is:

To modify the microstructure and shape memory property of Cu-Al-Ni shape memory alloys by addition of alloying elements and heat treatment

The specific objectives of the research are:

- 1. To investigate the effect of various alloying elements on the microstructure, mechanical properties and shape memory behaviour of Cu-Al-Ni-X alloys.
- 2. To determine the effect of aging treatment on the microstructure, mechanical properties, and shape memory behaviour of Cu-Al-Ni-X alloys.
- 3. To correlate the effect of alloying elements and aging treatment on the damping properties of Cu-Al-Ni-X alloys.

1.4 Scopes of the Research

Based on the objectives of the study, the scopes of the research are:

- 1. Production of Cu-Al-Ni shape memory alloys with and without addition of alloying elements (Ti, Co, and B) using casting process.
- 2. Performed the aging treatment process on the base Cu-Al-Ni SMAs with addition of different alloying elements.
- 3. Conducted the material characterization to investigate the properties using various testing equipments such as inductively coupled plasma mass spectrometry (ICP-MS), differential scanning calorimetry (DSC), optical microscopy (OM), variable pressure scanning electron microscopy (VPSEM) and x-ray diffraction (XRD).
- 4. Performed the compression test using universal testing machine to investigate the shape memory effect and compression properties.
- 5. Determination of the damping characteristic using dynamic mechanical analysis (DMA) with cooling rate = 3° C/min, frequency = 1 Hz, and applied strain = 5.0×10^{-5} .

1.5 Significance of the Research

Copper-based shape memory alloys specifically the ternary Cu-Al-Ni have gained attention among scientists and engineers seeking for alloys with superior properties. Many attempted to improve the properties of Cu-Al-Ni by modifying the alloy composition with addition of various elements or by heat treatment processes but with no major success to be used for industrial applications. The primary purpose of this research is to enhance the properties of Cu-Al-Ni by adding fourth alloying element, namely, titanium, cobalt and boron followed by aging treatment with specific application that require good damping property. Hardly any literature or previous research have reported on the role of these three alloying elements and aging treatment on the damping property. The findings of this study will provide better understanding on the relationship between microstructural modification and the enhancement of mechanical properties of Cu-Al-Ni-X alloys especially on the damping property for seismic resistant devices and structures.

REFERENCES

- 1. Guo, H. Understanding global natural disasters and the role of earth observation. International Journal of Digital Earth. 2010;3(3):221-30.
- Song, G., Ma, N. and Li, H. N. Applications of shape memory alloys in civil structures. Engineering Structures. 2006;28(9):1266-74.
- Santos, F. Vibration Control with Shape-Memory Alloys in Civil Engineering Structures: Nova University of Lisbon; 2011.
- Humbeeck, J. V. and Kustov, S. Active and passive damping of noise and vibrations through shape memory alloys: applications and mechanisms. Smart Materials and Structures. 2005;14(5):S171-S85.
- Qian, H., Li, H., Song, G. and Guo, W. Recentering Shape Memory Alloy Passive Damper for Structural Vibration Control. Mathematical Problems in Engineering. 2013;2013:13.
- 6. Lexcellent, C. Shape-memory Alloys Handbook. 2013:1-9.
- Tadaki, T. Cu based shape memory alloys, Shape memory materials. Cambridge: Cambridge University Press; 2002.
- Recarte, V., Pérez-Sáez, R. B., San Juan, J., Bocanegra, E. H. and Nó, M. L. Influence of Al and Ni concentration on the Martensitic transformation in Cu-Al-Ni shape-memory alloys. Metallurgical and Materials Transactions A. 2002;33(8):2581-91.
- Sarı, U. and Aksoy, İ. Electron microscopy study of 2H and 18R martensites in Cu–11.92 wt% Al–3.78 wt% Ni shape memory alloy. Journal of Alloys and Compounds. 2006;417(1–2):138-42.
- Sarı, U. and Aksoy, İ. Electron microscopy study of 2H and 18R martensites in Cu–11.92wt% Al–3.78wt% Ni shape memory alloy. Journal of Alloys and Compounds. 2006;417(1):138-42.
- Palánki, Z., Daróczi, L., Lexcellent, C. and Beke, D. L. Determination of the equilibrium transformation temperature (T0) and analysis of the non-chemical energy terms in a CuAlNi single crystalline shape memory alloy. Acta Materialia. 2007;55(5):1823-30.

- Gastien, R., Corbellani, C. E., Sade, M. and Lovey, F. C. Thermal and pseudoelastic cycling in Cu-14.1Al-4.2Ni (wt%) single crystals. Acta Materialia. 2005;53(6):1685-91.
- Sobrero, C. E., La Roca, P., Roatta, A., Bolmaro, R. E. and Malarría, J. Shape memory properties of highly textured Cu–Al–Ni–(Ti) alloys. Materials Science and Engineering: A. 2012;536:207-15.
- Adachi, K., Shoji, K. and Hamada, Y. Formation of X Phases and Origin of Grain Refinement Effect in Cu-Al-Ni Shape Memory Alloys Added with Titamium. ISIJ International. 1989;29(5):378-87.
- Saiji, M., Masami, H., Ryuichiro, O. and Francisco Eiichi, F. Improvement of Ductility of Melt Spun Cu–Al–Ni Shape Memory Alloy Ribbons by Addition of Ti or Zr. Japanese Journal of Applied Physics. 1983;22(8A):L528.
- Odusote, J., Adeleke, A. and Ajayi, P. Mechanical Properties and Microstructure of Precipitation-Hardening Al-Cu-Zn alloys. International Journal of Automotive and Mechanical Engineering. 2015;12:3033-42.
- Saud, S., Hamzah, E., Abubakar, T. and Hosseinian, R. A Review of Alloying Element on the Microstructure and Mechanical Properties of Cu-Al-Ni Shape Memory Alloys. Jurnal Teknologi. 2013;64(1):51-6.
- Sampath, V. Improvement of Shape-Memory Characteristics and Mechanical Properties of Copper–Zinc–Aluminum Shape-Memory Alloy with Low Aluminum Content by Grain Refinement. Materials and Manufacturing Processes. 2006;21(8):789-95.
- Lee, J. S. and Wayman, C. M. Grain Refinement of a Cu-Al-Ni Shape Memory Alloy by Ti and Zr Additions. Transactions of the Japan Institute of Metals. 1986;27(8):584-91.
- Lee, J. S. and Wayman, C. M. Grain refinement of Cu · Zn · Al shape memory alloys. Metallography. 1986;19(4):401-19.
- Hurtado, I., Humbeeck, J. V., Ratchev, P. and Delaey, L. Effect of X-Phase Precipitation on Elastic Modulus of Cu-Al-Ni-(Ti)-(Mn) Shape Memory Alloys. Materials Transactions, JIM. 1996;37(7):1371-8.
- Canbay, C. A. and Karagoz, Z. Effect of Annealing Temperature on Thermomechanical Properties of Cu-Al-Ni Shape Memory Alloys. International Journal of Thermophysics 2013;34(7):1325-35.

- Benke, M., Mertinger, V., Nagy, E. and Van Humbeeck, J. Investigation of Ageing Phenomena in CuAlNi Based Shape Memory Alloys. Materials Science Forum. 2007;537-538:129-36.
- Ozbulut, O. E., Hurlebaus, S. and Desroches, R. Seismic Response Control Using Shape Memory Alloys: A Review. Journal of Intelligent Material Systems and Structures. 2011;22(14):1531-49.
- 25. Otsuka, K. a. W., C. M. Shape Memory Materials. United Kingdom: Cambridge University Press; 1998.
- 26. Liu, Y., Xie, Z. and Van Humbeeck, J. Cyclic deformation of NiTi shape memory alloys. Materials Science and Engineering: A. 1999;273-275:673-8.
- Juan, J. S., Nó, M. L. and Schuh, C. A. Nanoscale shape-memory alloys for ultrahigh mechanical damping. Nature Nanotechnology. 2009;4:415.
- 28. Van Humbeeck, J. Damping capacity of thermoelastic martensite in shape memory alloys. Journal of Alloys and Compounds. 2003;355(1):58-64.
- Kustov, S., Golyandin, S., Sapozhnikov, K., Van Humbeeck, J. and De Batist,
 R. Low-temperature anomalies in Young's modulus and internal friction of Cu–
 Al–Ni single crystals. Acta Materialia. 1998;46(14):5117-26.
- Suresh, N. and Ramamurty, U. The effect of ageing on the damping properties of Cu–Al–Ni shape memory alloys. Smart Materials and Structures. 2005;14(5):N47.
- Neelakantan, S. and Ramamurty, U. Aging response and its effect on the functional properties of Cu–Al–Ni shape memory alloys2008. 113–8 p.
- 32. Shivaramu, L., Shivasiddaramaiah, A. G., Mallik, U. S. and Prashantha, S. Effect Of Ageing On Damping Characteristics Of Cu-Al-Be-Mn Quaternary Shape Memory Alloys. Materials Today: Proceedings. 2017;4(10):11314-7.
- Mallik, U. S. and Sampath, V. Effect of composition and ageing on damping characteristics of Cu–Al–Mn shape memory alloys. Materials Science and Engineering: A. 2008;478(1):48-55.
- 34. Alaneme, K. K., Okotete, E. A. and Anaele, J. U. Structural vibration mitigation – a concise review of the capabilities and applications of Cu and Fe based shape memory alloys in civil structures. Journal of Building Engineering. 2019;22:22-32.

- Kang, G. and Song, D. Review on structural fatigue of NiTi shape memory alloys: Pure mechanical and thermo-mechanical ones. Theoretical and Applied Mechanics Letters. 2015;5(6):245-54.
- Scirè Mammano, G. and Dragoni, E. Functional fatigue of Ni–Ti shape memory wires under various loading conditions. International Journal of Fatigue. 2014;69:71-83.
- Duerig, T. W., Melton, K. N., Stockel, D. and Wayman, C. M. Engineering Aspects of Shape Memory Alloys: Books on Demand; 1990.
- Wu, S. K., Lin, H. C. and Chen, C. C. A study on the machinability of a Ti49.6Ni50.4 shape memory alloy. Materials Letters. 1999;40(1):27-32.
- Mercier, O., Melton, K. N. and De Préville, Y. Low-frequency internal friction peaks associated with the martensitic phase transformation of NiTi. Acta Metallurgica. 1979;27(9):1467-75.
- 40. Coluzzi, B., Biscarini, A., Campanella, R., Trotta, L., Mazzolai, G., Tuissi, A., et al. Mechanical spectroscopy and twin boundary properties in a Ni50.8Ti49.2 alloy. Acta Materialia. 1999;47(6):1965-76.
- Yuan, B., Chung, C. Y. and Zhu, M. Microstructure and martensitic transformation behavior of porous NiTi shape memory alloy prepared by hot isostatic pressing processing. Materials Science and Engineering: A. 2004;382(1):181-7.
- 42. Bansiddhi, A., Sargeant, T. D., Stupp, S. I. and Dunand, D. C. Porous NiTi for bone implants: A review. Acta Biomaterialia. 2008;4(4):773-82.
- 43. Ölander, A. An Electrochemical Investigation of Solid Cadmium-Gold Alloys. Journal of the American Chemical Society. 1932;54(10):3819-33.
- Vernon Lester, B. and Vernon Harold, M., inventors; Vernon Benshoff Company, assignee. Process Of Manufacturing Articles Of Thermoplastic Synthetic Resins. US patent US 2234993 A. 1941 1937/02/06.
- 45. Bystrom, A. A., Karl Erik. X-ray investigation of gold-cadmium alloys rich in gold. Acta Chem Scand. 1947;1(1):76-89.
- 46. Chang, L. C. and Read, T. A. Plastic Deformation and Diffusionless Phase Changes in Metals the Gold-Cadmium Beta Phase. JOM. 1951;3(1):47-52.
- Kurdjumov, G. V. and Khandros, L. G. First reports of the thermoelastic behaviour of the martensitic phase of Au-Cd alloys. Doklady Akademii Nauk SSSB. 1949;66:211-3.

- Buehler, W. J., Gilfrich, J. V. and Wiley, R. C. Effect of Low Temperature Phase Changes on the Mechanical Properties of Alloys near Composition TiNi. Journal of Applied Physics. 1963;34(5):1475-7.
- Jackson, C. M., Wagner, H. J. and Wasilewski, R. K. The alloy with a memory,
 55-Nitinol: Its physical metallurgy, properties, and applications. NASA,
 Washington, United States; 1972.
- Kauffman, G. B. and Mayo, I. The Story of Nitinol: The Serendipitous Discovery of the Memory Metal and Its Applications. The Chemical Educator. 1997;2(2):1-21.
- Wu, M. H. and Schetky, L. M., editors. Industrial Applications for Shape Memory Alloys. Proceedings of the International Conference on Shape Memory and Superelastic Technologies; 2000; Pacific Grove, California.
- Callister, W. D. and Rethwisch, D. G. Material Science and Engineering: Wiley; 2011. 343-4 p.
- Otsuka, K. and Wayman, C. M. Shape Memory Materials. Otsuka KW, C. M., editor. Cambridge, United Kingdom: Cambridge University Press; 1998. 27-45 p.
- Funakubo, H. Precision Machinery and Robotics: Gordon and Breach Science Publishers; 1987.
- Van der Wijst, M. W. M. Shape Memory Alloys featuring Nitinol. Eindhoven: Technische Universiteit Eindhoven; 1992.
- 56. Otsuka, K. and Ren, X. Physical metallurgy of Ti–Ni-based shape memory alloys. Progress in Materials Science. 2005;50(5):511-678.
- Kumar, P. K. and Lagoudas, D. C. Introduction to Shape Memory Alloys. Shape Memory Alloys: Modeling and Engineering Applications. Boston, MA: Springer US; 2008. p. 1-51.
- 58. Brojan, M., Bombač, D., Kosel, F. and Videnič, T. Shape memory alloys in medicine 2008;55:173-89.
- Otsuka, K., Saxena, A., Deng, J. and Ren, X. Mechanism of the shape memory effect in martensitic alloys: an assessment. Philosophical Magazine. 2011;91(36):4514-35.
- Lagoudas, D. C. Shape Memory Alloys: Modeling and Engineering Applications: Springer US; 2008.

- Debbarma.S.R, S. S. Review of Shape Memory applications in civil structures, and analysis for its potential as reinforcement in concrete flexural members. International Journal of Civil and Structural Engineering. 2012;2(3):924 - 42.
- 62. Fugazza, D. shape memory alloy decives in earthquake engineering: mechanical properties, constitutive modelling and numerical simulations. Pavia2003.
- 63. Blanter, M. S., Neuhauser, H., Golovin, I. S. and Sinning, H. R. Internal Friction in Metallic Materials Springer; 2007.
- 64. Lazan, B. J. Damping of materials and members in structural mechanics. 1968.
- Pérez-Sáez, R. B., Recarte, V., Nó, M. L. and San Juan, J. Anelastic contributions and transformed volume fraction during thermoelastic martensitic transformations. Physical Review B. 1998;57(10):5684-92.
- 66. Wang, Q. Z., Lu, D. M., Cui, C. X., Xu, M. and Yang, J. Internal friction of a CuAlMn shape memory alloy studied as a function of frequency. Materials Science and Engineering: A. 2015;627:277-80.
- Kustov, S., Cesari, E. and Van Humbeeck, J. Effects of Pinning and Atomic Ordering on Damping Properties and Martensitic Transformation of Copper-Based Shape Memory Alloys. Solid State Phenomena. 2012;184:366-71.
- Gong, C. L., Han, F. S., Li, Z. and Wang, M. P. Two internal-friction peaks related to thermoelastic martensitic transformations in CuAlNiMnTi shapememory alloy. Physical Review B. 2004;70(9):094103.
- Bidaux, J. E., Schaller, R. and Benoit, W. Study of the h.c.p.-f.c.c. phase transition in cobalt by acoustic measurements. Acta Metallurgica. 1989;37(3):803-11.
- 70. Stoiber, J., Bidaux, J. E. and Gotthardt, R. The movement of single β 1 → β 1
 ' interfaces in Cu · Zn · Al as studied by a new technique of internal friction measurement. Acta Metallurgica et Materialia. 1994;42(12):4059-70.
- San Juan, J. and Nó, M. L. Damping behavior during martensitic transformation in shape memory alloys. Journal of Alloys and Compounds. 2003;355(1):65-71.
- 72. Dejonghe, W., De Batist, R. and Delaey, L. Factors affecting the internal friction peak due to thermoelastic martensitic transformation. Scripta Metallurgica. 1976;10(12):1125-8.

- Mercier, O. and Melton, K. N. The influence of an anisotrophic elastic medium on the motion of dislocations: Application to the martensitic transformation. Scripta Metallurgica. 1976;10(12):1075-80.
- Kustov, S. B., Golyandin, S. N., Hurtado, I., Van Humbeeck, J. and Batist, R.
 d. Structural Internal Friction in CuAlNi Crystals. J Phys IV France. 1995;05(C8):C8-943-C8-8.
- Scheil, E. and Müller, J. Die Dämpfung mechanischer Schwingungen während der Martensitbildung in Eisen-Nickel-Legierungen. Archiv für das Eisenhüttenwesen. 1956;27(12):801-5.
- Belko, V. N., B.M. Darinsky, V.S. Postnikov and Sharshakov, I. M. Internal Friction During Diffusionless Phase Transformation in Co-Ni alloys. Phys, Met Metallogr. 1969;27(140).
- 77. V.N. Belko, B. M. Darinsky, Postnikov, V. S. and Scharsakov, I. M. Internal Friction During Diffusionless Phase Transformation in Co-Ni Alloys. Fiz Met Metalloved. 1969;27(141).
- Van Humbeeck, J. Damping Properties of Shape Memory Alloys During Phase Transformation. J Phys IV France. 1996;06(C8):C8-371-C8-80.
- 79. Matsuzaki, Y., Ikeda, T. and Boller, C. New technological development of passive and active vibration control: analysis and test. Smart Materials and Structures. 2005;14(2):343-8.
- Hodgson, D. E. Damping Applications of Shape-Memory Alloys. Materials Science Forum. 2002;394-395:69-74.
- Chang, S. H. Influence of chemical composition on the damping characteristics of Cu-Al-Ni shape memory alloys. Materials Chemistry and Physics. 2011;125(3):358-63.
- Huang, S., Leary, M., Ataalla, T., Probst, K. and Subic, A. Optimisation of Ni-Ti shape memory alloy response time by transient heat transfer analysis. Materials & Design. 2012;35:655-63.
- Shabalovskaya, S., Anderegg, J. and Van Humbeeck, J. Critical overview of Nitinol surfaces and their modifications for medical applications. Acta Biomaterialia. 2008;4(3):447-67.
- Fadlallah, S. A., El-Bagoury, N., Gad El-Rab, S. M. F., Ahmed, R. A. and El-Ousamii, G. An overview of NiTi shape memory alloy: Corrosion resistance

and antibacterial inhibition for dental application. Journal of Alloys and Compounds. 2014;583(0):455-64.

- Bakar, T. A. A. Tribological Investigation of Nickel Titanium Shape Memory Alloy Coatings. Ireland: Dublin City University; 2010.
- Desroches, R. a. S., B. Shape memory alloys is seismic resistant design and retrofit: A critical review of their potential and limitations. Journal of Earthquake Engineering. 2004;8(3):415-29.
- Bolce, M. and Cardone, D. Mechanical behaviour of shape memory alloys for seismic applications - 1. Martensite and austenite NiTi bars subjected to torsion. International Journal of Mechanical Sciences. 2001;43(11):2631-56.
- 88. Duerig, T. W. Engineering Aspects of Shape Memory Alloys: Butterworth-Heinemann Limited; 1990.
- Ejik, C. V. D., Olsen, J. S. and Zhang, Z. L. Applications of NiTi Shape Memory Alloy Dampers in Civil Structures. Proceedings of the First International Conference of Self Healing Materials; Norway2007.
- Casciati, F., Faravelli, L. and Petrini, L. Energy Dissipation in Shape Memory Alloy Devices. Computer-Aided Civil and Infrastructure Engineering. 1998;13(6):433-42.
- Li, H., Liu, M. and Ou, J. Vibration mitigation of a stay cable with one shape memory alloy damper. Structural Control and Health Monitoring. 2004;11(1):21-36.
- Lojen, G., Gojić, M. and Anžel, I. Continuously cast Cu–Al–Ni shape memory alloy – Properties in as-cast condition. Journal of Alloys and Compounds. 2013;580:497-505.
- Yildiz, K. and Kok, M. Study of martensite transformation and microstructural evolution of Cu–Al–Ni–Fe shape memory alloys. J Therm Anal Calorim. 2013:1-6.
- 94. Tsuchiya, K. 1 Mechanisms and properties of shape memory effect and superelasticity in alloys and other materials: a practical guide. In: Yamauchi K, Ohkata I, Tsuchiya K, Miyazaki S, editors. Shape Memory and Superelastic Alloys: Woodhead Publishing; 2011. p. 3-14.
- Tarhan, E. Aging Characteristics of Copper Based Shape Memory Alloys: The Middle East Technical University; 2004.

- 96. Agrawal, A. and Dube, R. K. Methods of fabricating Cu-Al-Ni shape memory alloys. Journal of Alloys and Compounds. 2018;750:235-47.
- Tadaki, T. Cu based Shape Memory Alloys. Shape Memory Materials. United Kingdom: Cambridge University Press; 1998. p. 97-116.
- 98. Picornell, C., Pons, J. and Cesari, E. Effects of Thermal Ageing in β-Phase in Cu-Al-Ni Single Crystals. J Phys IV France. 1997;07(C5):C5-323-C5-8.
- 99. Vajpai, S. K., Dube, R. K. and Sangal, S. Application of rapid solidification powder metallurgy processing to prepare Cu–Al–Ni high temperature shape memory alloy strips with high strength and high ductility. Materials Science and Engineering: A. 2013;570:32-42.
- 100. Sakamoto, H. and Shimizu, K. i. Effect of Heat Treatments on Thermally Formed Martensite Phases in Monocrystalline Cu-Al-Ni Shape Memory Alloy. ISIJ International. 1989;29(5):395-404.
- Otsuka, K. and Wayman, C. M. Shape memory materials. Cambridge: Cambridge University Press; 2002.
- 102. Suru, M.-G., Lohan, N.-M., Pricop, B., Mihalache, E., Mocanu, M. and Bujoreanu, L.-G. Precipitation Effects on the Martensitic Transformation in a Cu-Al-Ni Shape Memory Alloy. Journal of Materials Engineering and Performance. 2016;25(4):1562-9.
- Friend, C. M. The effect of aluminium content on the martensite phase stabilities in metastable CuAlNi alloys. Scripta Metallurgica. 1989;23(10):1817-20.
- Miyazaki, S. Development and Characterization of Shape Memory Alloys. Shape Memory Alloys. Vienna: Springer Vienna; 1996. p. 69-147.
- 105. Recarte, V., Pérez-Sáez, R. B., Bocanegra, E. H., Nó, M. L. and San Juan, J. Dependence of the martensitic transformation characteristics on concentration in Cu–Al–Ni shape memory alloys. Materials Science and Engineering: A. 1999;273-275:380-4.
- Recarte, V., Pérez-Landazábal, J. I., Rodríguez, P. P., Bocanegra, E. H., Nó, M. L. and San Juan, J. Thermodynamics of thermally induced martensitic transformations in Cu–Al–Ni shape memory alloys. Acta Materialia. 2004;52(13):3941-8.

- Suresh, N. and Ramamurty, U. Effect of aging on mechanical behavior of single crystal Cu–Al–Ni shape memory alloys. Materials Science and Engineering: A. 2007;454-455:492-9.
- Zárubová, N., Gemperle, A. and Novák, V. Initial stages of γ2 precipitation in an aged Cu-Al-Ni shape memory alloy. Materials Science and Engineering: A. 1997;222(2):166-74.
- 109. Karagoz, Z. and Canbay, C. A. Relationship between transformation temperatures and alloying elements in Cu–Al–Ni shape memory alloys. Journal of Thermal Analysis and Calorimetry. 2013;114(3):1069-74.
- Miyazaki, S., Shibata, K. and Fujita, H. Effect of specimen thickness on mechanical properties of polycrystalline aggregates with various grain sizes. Acta Metallurgica. 1979;27(5):855-62.
- 111. Miyazaki, S. and Fujita, H. Effects of Grain Size and Specimen Thickness on Mechanical Properties of Polycrystalline Copper and Copper-Aluminum Alloy. Transactions of the Japan Institute of Metals. 1978;19(8):438-44.
- 112. Dagdelen, F., Gokhan, T., Aydogdu, A., Aydogdu, Y. and Adigüzel, O. Effects of thermal treatments on transformation behaviour in shape memory Cu–Al– Ni alloys. Materials Letters. 2003;57(5):1079-85.
- 113. Tatar, C. Gamma irradiation-induced evolution of the transformation temperatures and thermodynamic parameters in a CuZnAl shape memory alloy. Thermochimica Acta. 2005;437(1):121-5.
- 114. Balo, Ş. N. and Sel, N. Effects of thermal aging on transformation temperatures and some physical parameters of Cu-13.5wt.%Al-4wt.%Ni shape memory alloy. Thermochimica Acta. 2012;536:1-5.
- Miyazaki, S., Otsuka, K., Sakamoto, H. and Shimizu, K. The Fracture of Cu-Al-Ni Shape Memory Alloy. Transactions of the Japan Institute of Metals. 1981;22(4):244-52.
- 116. Kustov, S., B., Golyandin, S., N., Hurtado, I., Van Humbeeck, J. and de Batist,
 R. Internal Friction in Cu-Al-Ni Crystals in Martensitic Phase and During Temperature-Induced Martensitic Transformation. J Phys IV France. 1996;06(C8):C8-389-C8-92.
- 117. Kustov, S., Golyandin, S., Sapozhnikov, K., Pons, J., Cesari, E. and Van Humbeeck, J. Effect of off-stoichiometry on the mobility of point-like defects

and damping in binary Cu–Al martensites. Acta Materialia. 2006;54(8):2075-85.

- Kostrubiec, B., Rasek, J., Wiśniewski, R. and Morawiec, H. Mechanical Spectroscopy in Cu-Al-Zn and Cu-Al-Ni Alloys. Solid State Phenomena. 2003;89:287-92.
- Suzuki, K., Nakanishi, N. and Mitani, H. Effects of Cooling Rates on Internal Friction in Cu-Al-Ni Ternary Alloys. Japan Institute of Metals. 1980;44:43.
- Pérez-Landazábal, J. I., Recarte, V., Ezpeleta, J. M., Rodríguez, P. P., San Juan,
 J. and Nó, M. L. Vibrational behavior of the β phase near martensitic transformation in Cu–Al–Ni shape memory alloys. Materials Science and Engineering: A. 2004;378(1):243-7.
- Recarte, V., Herreros, J., Nó, M. L. and San Juan, J. Internal Friction and Microdeformation on Cu-Al-Ni Shape Memory Alloys. Materials Science Forum. 1993;119-121:323-30.
- 122. Ibarra, A., Rodríguez, P. P., Recarte, V., Pérez-Landazábal, J. I., Nó, M. L. and San Juan, J. Internal friction behaviour during martensitic transformation in shape memory alloys processed by powder metallurgy. Materials Science and Engineering: A. 2004;370(1):492-6.
- 123. Covarel, G., Pelosin, V. and Rivière, A. Influence of martensite variant orientation on their mobility in a CuAlNi alloy. J Phys IV France. 2001;11(PR8):Pr8-153-Pr8-8.
- 124. Pérez-Sáez, R. B., Recarte, V., Nó, M. L. and San Juan, J. Analysis of the internal friction spectra during martensitic transformation by a new temperature rate method. Journal of Alloys and Compounds. 2000;310(1):334-8.
- 125. Covarel, G., Pelosin, V. and Rivière, A. Cu–13.2Al–3Ni (wt%) single crystal studied by isothermal mechanical spectroscopy. Journal of Alloys and Compounds. 2000;310(1):330-3.
- Leu, S. S., Chen, Y. C. and Jean, R. D. Effect of rapid solidification on mechanical properties of Cu-Al-Ni shape memory alloys. Journal of Materials Science. 1992;27(10):2792-8.
- 127. Roh, D. W., Kim, J. W., Cho, T. J. and Kim, Y. G. Tensile properties and microstructure of microalloyed Cu-Al-Ni-X shape memory alloys. Materials Science and Engineering: A. 1991;136:17-23.

- Duerig, T. W., Albrecht, J. and Gessinger, G. H. A Shape-Memory Alloy for High-Temperature Applications. JOM. 1982;34(12):14-20.
- 129. Sure, G. N. and Brown, L. C. The mechanical properties of grain refined βcuaini strain-memory alloys. Metallurgical and Materials Transactions A. 1984;15(8):1613-21.
- Morris, M. A. Influence of boron additions on ductility and microstructure of shape memory Cu-Al-Ni alloys. Scripta Metallurgica et Materialia. 1991;25(11):2541-6.
- 131. Lojen, G., Anžel, I., Kneissl, A., Križman, A., Unterweger, E., Kosec, B., et al. Microstructure of rapidly solidified Cu–Al–Ni shape memory alloy ribbons. Journal of Materials Processing Technology. 2005;162-163:220-9.
- 132. Vajpai, S. K., Dube, R. K. and Sangal, S. Microstructure and properties of Cu– Al–Ni shape memory alloy strips prepared via hot densification rolling of argon atomized powder preforms. Materials Science and Engineering: A. 2011;529:378-87.
- 133. Tang, S. M., Chung, C. Y. and Liu, W. G. Preparation of CuAlNi-based shape memory alloys by mechanical alloying and powder metallurgy method. Journal of Materials Processing Technology. 1997;63(1):307-12.
- Rodríguez, P. P., Ibarra, A., Iza-Mendia, A., Recarte, V., Pérez-Landazábal, J.
 I., San Juan, J., et al. Influence of thermo-mechanical processing on the microstructure of Cu-based shape memory alloys produced by powder metallurgy. Materials Science and Engineering: A. 2004;378(1):263-8.
- 135. Kim, J. W., Roh, D. W., Lee, E. S. and Kim, Y. G. Effects on microstructure and tensile properties of a zirconium addition to a Cu-Al-Ni shape memory alloy. Metallurgical Transactions A. 1990;21(2):741-4.
- Zhang, X. and Liu, Q. Cu-Al-Ni-V high-temperature shape memory alloys. Intermetallics. 2018;92:108-12.
- 137. Sugimoto, K., Kamei, K., Matsumoto, H., Komatsu, S., Akamatsu, K. and Sugimoto, T. Grain-refinement and the Related Phenomena in Quaternary Cu-Al-Ni-Ti Shape Memory Alloys J Phys Colloques. 1982;43(C4):C4-761-C4-6.
- 138. Dalvand, P., Raygan, S., López, G. A., Meléndez, M. B. and Chernenko, V. A. Properties of rare earth added Cu-12wt%Al-3wt%Ni-0.6wt%Ti high temperature shape memory alloy. Materials Science and Engineering: A. 2019;754:370-81.

- 139. Yildiz, K., Kök, M. and Dağdelen, F. Cobalt addition effects on martensitic transformation and microstructural properties of high-temperature Cu–Al–Fe shape-memory alloys. Journal of Thermal Analysis and Calorimetry. 2015;120(2):1227-32.
- Gil, F. J. and Guilemany, J. M. Effect of cobalt addition on grain growth kinetics in Cu · Zn · Al shape memory alloys. Intermetallics. 1998;6(5):445-50.
- Hussain, S., Pandey, A. and Dasgupta, R. Designed polycrystalline ultra-high ductile boron doped Cu–Al–Ni based shape memory alloy. Materials Letters. 2019;240:157-60.
- 142. Zhang, X., Zhang, M., Cui, T., Li, J., Liu, Q. and Wang, H. The enhancement of the mechanical properties and the shape memory effect for the Cu-13.0Al-4.0Ni alloy by boron addition. Journal of Alloys and Compounds. 2019;776:326-33.
- 143. Birol, Y. Grain refinement of Al–Cu foundry alloys with B additions. International Journal of Cast Metals Research. 2012;25(2):117-20.
- 144. Lozovoi, A. Y. and Paxton, A. T. Boron in copper: A perfect misfit in the bulk and cohesion enhancer at a grain boundary. Physical Review B. 2008;77(16):165413.
- 145. Wang, T., Chen, Z., Fu, H., Xu, J., Fu, Y. and Li, T. Grain refining potency of Al–B master alloy on pure aluminum. Scripta Materialia. 2011;64(12):1121-4.
- 146. Sampath, V. Studies on the effect of grain refinement and thermal processing on shape memory characteristics of Cu–Al–Ni alloys. Smart Materials and Structures. 2005;14(5):S253-S60.
- 147. Sakamoto, H., Kijima, Y., Shimizu, K., rsquo and ichi. Fatigue and Fracture Characteristics of Polycrystalline Cu–Al–Ni Shape Memory Alloys. Transactions of the Japan Institute of Metals. 1982;23(10):585-94.
- 148. Saud, S. N., Abu Bakar, T. A., Hamzah, E., Ibrahim, M. K. and Bahador, A. Effect of Quarterly Element Addition of Cobalt on Phase Transformation Characteristics of Cu-Al-Ni Shape Memory Alloys. Metallurgical and Materials Transactions A. 2015;46(8):3528-42.
- 149. Saud, S. N., Hamzah, E., Abubakar, T., Zamri, M. and Tanemura, M. Influence of Ti additions on the martensitic phase transformation and mechanical

properties of Cu–Al–Ni shape memory alloys. Journal of Thermal Analysis and Calorimetry. 2014;118(1):111-22.

- 150. Campbell, F. C. Precipitation Hardening. Elements of Metallurgy and Engineering Alloys: ASM International; 2008. p. 135-48.
- 151. Gao, X. Y. and Huang, W. M. Transformation start stress in non-textured shape memory alloys. Smart Materials and Structures. 2002;11(2):256-68.
- 152. Shaw, J. A. and Kyriakides, S. Thermomechanical aspects of NiTi. Journal of the Mechanics and Physics of Solids. 1995;43(8):1243-81.
- 153. Ma, Y., Jiang, C., Li, Y., Xu, H., Wang, C. and Liu, X. Study of Ni50+xMn25Ga25-x (x=2-11) as high-temperature shape-memory alloys. Acta Materialia. 2007;55(5):1533-41.
- 154. Delorme, J. F. and Gobin, P. F. Internal Friction and Microdeformation Associated with the Martensitic Transformations in Metallic Solids. Met:Corros-Ind. 1973;573(185).
- Sugimoto, K., Kamei, K., Matsumoto, H., Komatsu, S., Akamatsu, K. and Sugimoto, T. Grain-Refinement and the Related Phenomena in Quatenary Cu-Al-Ni-Ti Shape Memory Alloys. J Phys Colloques. 1982;43(C4):C4-761-C4-6.
- 156. Ding, Y., Wang, Q., Yin, F., Cui, C. and Hao, G. Effect of combined addition of Cu51Zr14 inoculant and Ti element on the microstructure and damping behavior of a Cu-Al-Ni shape memory alloy. Materials Science and Engineering: A. 2019;743:606-10.
- 157. Canbay, C. A. and Karagoz, Z. Effects of Annealing Temperature on Thermomechanical Properties of Cu–Al–Ni Shape Memory Alloys. Int J Thermophys. 2013;34(7):1325-35.
- 158. Itsumi, Y., Miyamoto, Y., Takashima, T., Kamei, K. and Sugimoto, K. The Effects of Ageing on the Martensitic Transformation Temperature in Cu-Al-Ni-Mn-Ti Shape Memory Alloys. Materials Science Forum. 1990;56-58:469-74.
- Yang, G. S., Lee, J. K. and Jang, W. Y. Microstructural Evolution in CuAlNi Alloy with Ageing. Solid State Phenomena. 2007;124-126:1485-8.
- Bouabdallah, M., Baguenane-Benalia, G., Saadi, A., Cheniti, H., Gachon, J.-C. and Patoor, E. Precipitation sequence during ageing in β1 phase of Cu–Al–

Ni shape memory alloy. Journal of Thermal Analysis and Calorimetry. 2013;112(1):279-83.

- 161. Itsumi, Y., Miyamoto, Y., Takashima, T., Kamei, K. and Sugimoto, K. The Effects of Ageing on the Martensitic Transformation Temperature in Cu-Al-Ni-Mn-Ti Shape Memory Alloys. Materials Science Forum. 1991;56-58:469-74.
- 162. Sari, U., Kirindi, T., Ozcan, F. and Dikici, M. Effects of aging on the microstructure of a Cu-Al-Ni-Mn shape memory alloy. International Journal of Minerals, Metallurgy, and Materials. 2011;18(4):430.
- 163. Wei, Z. G., Peng, H. Y., Zou, W. H. and Yang, D. Z. Aging effects in a Cu-12Al-5Ni-2Mn-1Ti shape memory alloy. Metallurgical and Materials Transactions A. 1997;28(4):955-67.
- 164. Morawiec, H. and Gigla, M. Effect of ageing on shape recovery in Cu-Al-Ni alloy with Ti + B additions. Acta Metallurgica et Materialia. 1994;42(8):2683-6.
- 165. Ainul Haidar, M., Saud, S. N. and Hamzah, E. Microstructure, Mechanical Properties, and Shape Memory Effect of Annealed Cu-Al-Ni-xCo Shape Memory Alloys. Metallography, Microstructure, and Analysis. 2018;7(1):57-64.
- 166. Saud, S., Hamzah, E., Abubakar, T. and Reza Bekheshishi, H. Thermal Aging Behavior in Cu-Al-Ni-xCo Shape Memory Alloys2014.
- 167. Suresh, N. and Ramamurty, U. The effect of ageing on the damping properties of Cu-Al-Ni shape memory alloys. Smart Materials & Structures. 2005;14(5):N47-N51.
- Yoshida, I., Ono, T. and Asai, M. Internal friction of Ti–Ni alloys. Journal of Alloys and Compounds. 2000;310(1):339-43.
- 169. Gilstrap, R. and Allen, R. A colloidal nanoparticle form of indium tin oxide: system development and characterization. 2009.
- 170. Oishi, K. and Brown, L. C. Stress-induced martensite formation in Cu–Al–Ni alloys. Metallurgical Transactions. 1971;2(7):1971-7.
- Aydogdu, Y., Aydogdu, A. and Adiguzel, O. Self-accommodating martensite plate variants in shape memory CuAlNi alloys. Journal of Materials Processing Technology. 2002;123(3):498-500.

- Saburi, T., Nenno, S., Kato, S. and Takata, K. Configurations of martensite variants in Cu-Zn-Ga. Journal of The Less-Common Metals. 1976;50(2):223-36.
- 173. Saburi, T. and Wayman, C. M. Crystallographic similarities in shape memory martensites. Acta Metallurgica. 1979;27(6):979-95.
- 174. Sarı, U. and Aksoy, İ. Micro-structural analysis of self-accommodating martensites in Cu-11.92wt%Al-3.78wt%Ni shape memory alloy. Journal of Materials Processing Technology. 2008;195(1):72-6.
- 175. Kayali, N., özgen, S. and Adigüel, O. Ageing effects on ordering degree and morphology of 18R-type martensite in shape memory CuZnAl alloys. Materials Research Bulletin. 1997;32(5):569-78.
- 176. Qader, I. N., Kök, M. and Dağdelen, F. Effect of Heat Treatment on Thermodynamics Parameters, Crystal and Microstructure of (Cu-Al-Ni-Hf) Shape Memory Alloy. Physica B: Condensed Matter. 2018.
- 177. Morris, M. A. and Gunter, S. Effect of heat treatment and thermal cycling on transformation temperatures of ductile Cu-Al-Ni-Mn-B alloys. Scripta Metallurgica et Materialia. 1992;26(11):1663-8.
- 178. Xu, J. W. Effects of Gd addition on microstructure and shape memory effect of Cu–Zn–Al alloy. Journal of Alloys and Compounds. 2008;448(1):331-5.
- 179. Vedamanickam, S. Studies on the effect of grain refinement and thermal processing on shape memory characteristics of Cu–Al–Ni alloys2005. S253 p.
- Liu, D., Hashimoto, H. and Ko, T. Electron microscopy study of martensite in Cu-11.2 wt% Al-3 wt% Ni. Journal of Materials Science. 1997;32(6):1657-63.
- 181. Morawiec, H. and Gigla, M. J de Physique IV, Colloq C8. 1995;5:937-42.
- 182. Sarı, U. and Kırındı, T. Effects of deformation on microstructure and mechanical properties of a Cu–Al–Ni shape memory alloy. Materials Characterization. 2008;59(7):920-9.
- 183. Wei, Z. G., Peng, H. Y., Yang, D. Z., Chung, C. Y. and Lai, J. K. L. Reverse transformations in CuA1NiMnTi alloy at elevated temperatures. Acta Materialia. 1996;44(3):1189-99.
- 184. Sari, U. Influences of 2.5wt% Mn addition on the microstructure and mechanical properties of Cu-Al-Ni shape memory alloys. International Journal of Minerals, Metallurgy, and Materials. 2010;17(2):192-8.

- 185. Braga, F. d. O., Matlakhov, A. N., Matlakhova, L. A., Monteiro, S. N. and Araújo, C. J. d. Martensitic Transformation Under Compression of a Plasma Processed Polycrystalline Shape Memory CuAlNi Alloy. Materials Research. 2017;20:1579-92.
- 186. Zhang, X., Sui, J., Liu, Q. and Cai, W. Effects of Gd addition on the microstructure, mechanical properties and shape memory effect of polycrystalline Cu-Al-Ni shape memory alloy. Materials Letters. 2016;180:223-7.
- 187. Otsuka, K., Ohba, T., Tokonami, M. and Wayman, C. M. New description of long period stacking order structures of martensites in β-phase alloys. Scripta Metallurgica et Materialia. 1993;29(10):1359-64.
- 188. Fu, H., Song, S., Zhuo, L., Zhang, Z. and Xie, J. Enhanced mechanical properties of polycrystalline Cu–Al–Ni alloy through grain boundary orientation and composition control. Materials Science and Engineering: A. 2016;650:218-24.
- Delorme, J., F., Schmid, R., Robin, M. and Gobin, P. Frottement Interieur Et Microdeformation Dans Les Transformations Martensitique. J Phys Colloques. 1971;32(C2):C2-101-C2-11.
- Yin, H., Yan, Y., Huo, Y. and Sun, Q. Rate dependent damping of single crystal CuAlNi shape memory alloy. Materials Letters. 2013;109:287-90.
- 191. Emel'yanov, Y., Golyandin, S., Kustov, S., Nikanorov, S., Pugachev, G., Sapozhnikov, K., et al. Detection of shock-wave-induced internal stresses in Cu-Al-Ni shape memory alloy by means of acoustic technique. Scripta Materialia. 2000;43(12):1051-7.
- 192. Sure, G. N. and Brown, L. C. The fatigue properties of grain refined β-Cu Al Ni strain-memory alloys. Scripta Metallurgica. 1985;19(4):401-4.
- Adachi, K., Hamada, Y. and Tagawa, Y. Crystal structure of the X-phase in grain-refined Cu-Al-Ni-Ti shape memory alloys. Scripta Metallurgica. 1987;21(4):453-8.
- 194. Hurtado, I., Ratchev, P., Van Humbeeck, J. and Delaey, L. A fundamental study of the χ-phase precipitation in Cu-Al-Ni-Ti-(Mn) shape memory alloys. Acta Materialia. 1996;44(8):3299-306.

- 195. Wee, Y. C., Abubakar, T., Hamzah, E. and Saud, S. Study of X-phase formation on Cu-Al-Ni shape memory alloys with Ti addition. Journal of Mechanical Engineering and Sciences. 2017;11:27702-2779.
- 196. Adachi, K., Shoji, K. and Hamada, Y. Formation of X Phases and Origin of Grain Refinement Effect in Cu–Al–Ni Shape Memory Alloys Added with Titamium. ISIJ International. 1989;29(5):378-87.
- 197. Dar, R. D., Yan, H. and Chen, Y. Grain boundary engineering of Co-Ni-Al, Cu-Zn-Al, and Cu-Al-Ni shape memory alloys by intergranular precipitation of a ductile solid solution phase. Scripta Materialia. 2016;115:113-7.
- 198. Yang, J., Wang, Q. Z., Yin, F. X., Cui, C. X., Ji, P. G. and Li, B. Effects of grain refinement on the structure and properties of a CuAlMn shape memory alloy. Materials Science and Engineering: A. 2016;664:215-20.
- 199. Zhang, J., Perez, R. J. and Lavernia, E. J. Dislocation-induced damping in metal matrix composites. Journal of Materials Science. 1993;28(3):835-46.
- 200. Saud, S., Hamzah, E., Abubakar, T. and Bakhsheshi-Rad, H. R. Thermal aging behavior in Cu–Al–Ni–xCo shape memory alloys. J Therm Anal Calorim. 2015;119(2):1273-84.
- 201. Aydoğdu, Y., Kürüm, F., Kök, M., Yakinci, Z. D. and Aydoğdu, A. Thermal Properties, Microstructure and Microhardness of Cu–Al–Co Shape Memory Alloy System. Transactions of the Indian Institute of Metals. 2014;67(4):595-600.
- Bhattacharya, S., Bhuniya, A. and Banerjee, M. K. Influence of minor additions on characteristics of Cu–Al–Ni alloy. Materials Science and Technology. 1993;9(8):654-8.
- 203. Morris, M. A. Microstructural influence on ductility and shape memory effect of some modified Cu-Ni-Al alloys. Scripta Metallurgica et Materialia. 1991;25(6):1409-14.
- 204. Li, Y., Wang, J. and Jiang, C. Study of Ni–Mn–Ga–Cu as single-phase widehysteresis shape memory alloys. Materials Science and Engineering: A. 2011;528(22):6907-11.
- 205. Kubo, H., Miyake, A. and Shimizu, K. i. Electron Microscopy Obeservation and X-ray Microanalysis of Aged Martensites in a Cu-12. 9 mass % Al-2. 5 mass % Mn Shape Memory Alloy. Transactions of the Japan Institute of Metals. 1983;24(9):603-12.

- 206. Recarte, V., Pérez-Sáez, R. B., Nó, M. L. and Juan, J. S. Evolution of martensitic transformation in Cu–Al–Ni shape memory alloys during lowtemperature aging. Journal of Materials Research. 2011;14(7):2806-13.
- 207. Picornell, C., Pons, J. and Cesari, E. Stress-temperature relationship in Cu-Al-Ni single crystals in compression mode. Materials Science and Engineering: A. 2004;378(1):222-6.
- 208. Pang, Q. H., Li, W. J., Cai, M. Y., Qi, H., Zhang, C. C. and Wu, J. N. The investigation of strength and plasticity mechanism of low-temperature annealed ultrafine grained stainless steel. IOP Conference Series: Materials Science and Engineering. 2018;372:012038.
- 209. Emdadi, A., Nartey, M. A., Xu, Y.-g. and Golovin, I. S. Study of damping capacity of Fe–5.4Al–0.05Ti alloy. Journal of Alloys and Compounds. 2015;653:460-7.
- Van Humbeeck, J. and Liu, Y. Shape Memory Alloys as Damping Materials. Materials Science Forum. 2000;327-328:331-8.
- 211. Alaneme, K. K. and Umar, S. Mechanical behaviour and damping properties of Ni modified Cu–Zn–Al shape memory alloys. Journal of Science: Advanced Materials and Devices. 2018;3(3):371-9.
- 212. Seguí, C., Pons, J., Cesari, E., Muntasell, J. and Font, J. Characterization of a hot-rolled Cu-Al-Ni-Ti shape memory alloy. Materials Science and Engineering: A. 1999;273-275:625-9.
- Saud, S. N., Hamzah, E., Abubakar, T., Bakhsheshi-Rad, H. R. and Hosseinian, R. X-phase precipitation in aging of Cu–Al–Ni–xTi shape memory alloys and its influence on phase transition behavior. Journal of Thermal Analysis and Calorimetry. 2016;123(1):377-89.
- 214. Guilemany, J. M. and Fernández, J. Relationships between structure and hardness developed during the high temperature ageing of a smart Cu-based alloy. Journal of Materials Science. 1996;31(18):4981-4.
- 215. Saud, S. N., Hamzah, E., Abubakar, T., Ibrahim, M. K. and Bahador, A. Effect of a fourth alloying element on the microstructure and mechanical properties of Cu–Al–Ni shape memory alloys. Journal of Materials Research. 2015;30(14):2258-69.

- 216. Wang, Q., Han, F., Cui, C., Bu, S. and Bai, L. Effect of ageing on the reverse martensitic phase transformation behaviors of a CuAlMn shape memory alloy. Materials Letters. 2007;61(30):5185-7.
- 217. Arsenault, R. J. and Shi, N. Dislocation generation due to differences between the coefficients of thermal expansion. Materials Science and Engineering. 1986;81:175-87.
- 218. Dunand, D. C. and Mortensen, A. Reinforced silver chloride as a model material for the study of dislocations in metal matrix composites. Materials Science and Engineering: A. 1991;144(1):179-88.
- 219. Zhang, J., Perez, R. J., Wong, C. R. and Lavernia, E. J. Effects of secondary phases on the damping behaviour of metals, alloys and metal matrix composites. Materials Science and Engineering: R: Reports. 1994;13(8):325-89.
- 220. Kustov, S., Pons, J., Cesari, E. and Van Humbeeck, J. Chemical and mechanical stabilization of martensite. Acta Materialia. 2004;52(15):4547-59.
- 221. Yoshida, I., Monma, D., Iino, K., Otsuka, K., Asai, M. and Tsuzuki, H. Damping properties of Ti50Ni50-xCux alloys utilizing martensitic transformation. Journal of Alloys and Compounds. 2003;355(1):79-84.
- 222. Schoeck, G. Internal Friction Due to Precipitation. physica status solidi (b). 1969;32(2):651-8.

LIST OF PUBLICATIONS

Indexed Journal

- Y. C. Wee, T. A. Abubakar, E. Hamzah, Safaa N. Saud, (2015), Phase Transformation and Microstructure Behaviour of Cu-Al-Ni Shape Memory Alloys Incorporated with Co Addition, *Jurnal Teknologi*,74(10), 53-56. (Indexed by SCOPUS)
- Y.C. Wee, T. A. Abubakar, E. Hamzah, Safaa N. Saud, (2017), Study of Xphase formation on Cu-Al-Ni shape memory alloys with Ti addition, *Journal* of Mechanical Engineering and Sciences, 11(2), 2770-2779. https://doi.org/10.15282/jmes.11.2.2017.17.0251. (Indexed by SCOPUS)
- Sukiman, N. A., H. Ghandvar, T. A. Abubakar, Y.C. Wee, (2019), Microstructure Characterization and Tensile Properties of Al-15%Mg2SixYSZ Hybrid Composite, *Malaysian Journal of Microscopy*. (Indexed by SCOPUS)

Articles in conference proceedings

- Y. C. Wee, H. Ghandvar, T. A. Abubakar, E. Hamzah (2019), Influence of Cobalt Addition on Microstructure and Damping Properties of Cu-Al-Ni Shape Memory Alloys, 3rd International Conference on Advanced Materials Characterization Techniques (AMCT 2019). Material Science Forum.www.scientific.nte/MSF.101034. (Indexed by SCOPUS)
- H. Ghandvar, T.A. Abubakar, N. A. Sukiman , Y.C.Wee (2019), Role of ZrO₂ Addition on Microstructure and Tensile Characteristics of Commercial Al-20Mg2Si-2Cu Metal Matrix Composite, 4th International Conference on the Science and Engineering of Materials 2019 (ICoSEM 2019) (accepted)